

Crystallographic Studies of
Iodide-Containing Quasi-One-Dimensional Conductors

Thesis by
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Abstract

The crystal structures of three iodide-containing quasi-one-dimensional conductors, (tetrathiotetracene)₂(iodide)₃ (high disorder), tetrathiotetracene-iodide, and (hexamethylenetetraselenofulvalene)-(iodide)_x were solved by single crystal X-ray diffraction methods. These three iodides are single charge-carrier conductors and allow a comparison of such competing effects as disorder, interchain coupling, and overlap.

The crystal structure of metallic (tetrathiotetracene)₂(iodide)₃ (high disorder), TTT₂I₃ (h.d.), was solved at room temperature (~294° K), 164° K, 74° K, and at 19° K. At all four temperatures the lattice symmetry remained orthorhombic and the structures were successfully refined in the space group Cmca. During slow cooling the diffuse layer lines were also carefully monitored. In contrast to TTFCl_{0.67}, even with slow cooling the iodide chains do not three-dimensionally order, and there are no distortions in the TTT lattice to 19° K. A model of the iodide chains is presented which explains the positions and intensities of the diffuse layer lines and also explains why three-dimensional ordering at low temperatures is not observed.

The structure of semiconducting tetrathiotetracene-iodide (TTTI) was studied at room temperature. The structure consists of two ordered lattices which are incommensurate along \vec{c} , the stacking axis. The unit cell dimensions for Lattice 1 (triclinic, $C\bar{1}$) are $a = 13.028(2)$, $b = 16.445(2)$, $c = 3.643(1)$ Å and $\alpha = 90.81(1)^\circ$, $\beta = 96.11(1)^\circ$, and $\gamma = 91.11(1)^\circ$.

For Lattice 2, $c \approx 4.78 \text{ \AA}$. The positions of all of the layer lines, including the two "sixth" layer lines, which are observed on X-ray oscillation photographs of crystals of TTTI rotated about \vec{c} , can be explained by the presence of two lattices. The measured density $d_m \approx 2.09 \text{ g/cm}^3$ and refinement of the [001] projection (hk0 reflections) confirmed that the overall stoichiometry is TTTI (1:1). For a complete data set collected with copper K_α radiation, the refinement of Lattice 1 converged to $R = 0.102$. For the 1132 reflections with $F_o^2 > 3\sigma(F_o^2)$, $R = 0.081$. The overlap between adjacent TTT cations in the same stack in TTTI is significantly different from that observed in TTT_2I_3 (h.d.). There is also very little interchain coupling in TTTI.

Hexamethylenetetraselenofulvalene-iodide, HMTSF- I_x , is triclinic, $\overline{P1}$, with the unit cell parameters $a = 8.056(4)$, $b = 12.740(4)$, $c = 8.016(3) \text{ \AA}$ and $\alpha = 81.72(4)^\circ$, $\beta = 67.73(5)^\circ$, and $\gamma = 102.64(4)^\circ$. For a complete data set of 4213 reflections collected with monochromatized molybdenum K_α radiation to $2\theta = 60^\circ$ the structure refined to $R = 0.097$. For 2042 reflections with $F_o^2 > 3\sigma(F_o^2)$, $R = 0.051$. The hydrogen atoms were not located. There is disordered iodide and solvent at $1/2, 1/2, 1/2$. The HMTSF cations stack along \vec{a} . A new type of alternating overlap between adjacent HMTSF molecules was observed. The magnitude of the d.c. electrical conductivity at room temperature suggests that this phase of HMTSF- I_x is semiconducting.

These iodide-containing structures show three different types of iodide behavior in quasi-one-dimensional conductors. In TTT_2I_3 (h.d.) the slip-stacking and large interchain coupling favor formation of a

metallic state at high temperatures. At low temperatures the disordered iodide chains have a major effect on the transport properties by allowing states to exist in the semiconductor band gap. In TTTI the iodides are no longer disordered but still dominate the physical properties by causing a modulation of the TTT lattice. There is very little inter-chain coupling in TTTI. In HMTSF- I_x the iodide is probably of minor importance.

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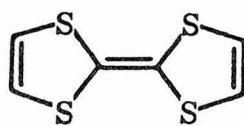
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Chapter 1

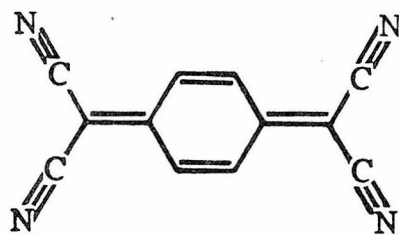
Introduction

Quasi-one-dimensional conductor is a label which has been given to materials that exhibit highly anisotropic, "one-dimensional", electrical conductivity. Conduction in the perpendicular directions in these materials is usually 10^3 to 10^5 times less than along the axis of high conductivity. Conductivity is not the only anisotropic property of these materials. They also have direction-dependent optical, magnetic and mechanical properties. An example of a quasi-one-dimensional material which has been studied in great detail is tetrathiafulvalene-tetracyanoquinodimethane (TTF-TCNQ). TTF and TCNQ are two planar organic molecules (see Figure 1) which together form a charge-transfer complex. TTF is an electron donor and TCNQ is an electron acceptor. In crystals of the complex the TTF molecules are stacked on top of each other and next to the TTF stacks are stacks of TCNQ molecules. The significance of this arrangement is that the TTF molecular orbitals are spatially close and it is energetically favorable for intermolecular orbital interaction and the formation of a "band". The TCNQ molecules also stack and the molecular orbitals also form a "band". Because there is charge transfer between the TTF and TCNQ molecules the bands become only partially filled and TTF-TCNQ exhibits metallic conductivity and metallic behavior parallel to the stacking direction. Conductivity parallel to the stacking direction is $\sim 500\text{--}700\ \Omega^{-1}\text{ cm}^{-1}$ at room temperature and $\sim 5\ \Omega^{-1}\text{ cm}^{-1}$ perpendicular to the stacking

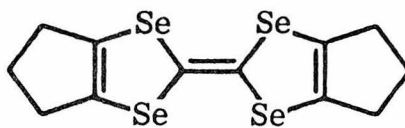
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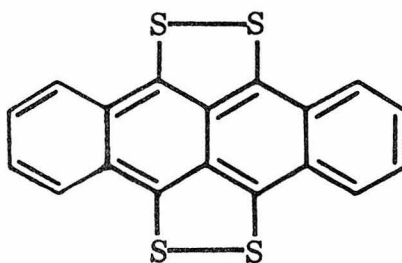
TTF



TCNQ



HMTSF



TTT

Figure 1

direction (1). Many quasi-one-dimensional conductors formed from organic donor-acceptor complexes with similar arrangements of separate cation and anion stacks are known (1). The best organic quasi-one-dimensional metal known is hexamethylenetetraselenofulvalene-tetracyanoquinodimethane (HMTSF-TCNQ) (2). It has a room temperature conductivity of $\sim 2000 \Omega^{-1} \text{ cm}^{-1}$ and, in contrast to TTF-TCNQ which undergoes a phase transition to an "insulating" state, HMTSF-TCNQ remains highly conducting to below 1° K (2). For comparison, copper, a good three-dimensional metal, has a conductivity of $5.88 \times 10^5 \Omega^{-1} \text{ cm}^{-1}$ at 295° K (3). Liquid mercury, a poor three-dimensional conductor, has a conductivity of $\sim 10000 \Omega^{-1} \text{ cm}^{-1}$ at 295° K (3).

The conductors formed from organic donor-acceptor charge-transfer complexes, such as TTF-TCNQ and HMTSF-TCNQ, are two-carrier systems. Electrons are conducted along the anion stacks and holes along the cation stacks. This makes the interpretation of their properties complex. The properties of single charge-carrier systems, in principle, are easier to interpret. Studies of single charge-carrier systems help toward a better understanding of the effect of disorder, impurities, intermolecular overlap, and interchain coupling. The TTF halides and pseudo-halides (SCN^- , SeCN^-) have been extensively studied for these reasons (for example 4,5,6,7,8).

The interpretation of solid-state physical properties ultimately depends on a knowledge of the underlying crystal and molecular structure. Although X-ray diffraction always emphasizes crystalline perfection, it is a convenient method for determining the actual internal structure

of a crystal. X-ray diffraction studies of the iodides of HMTSF and tetrathiotetracene (TTT) (Figure 1), which are single charge-carrier quasi-one-dimensional conductors, were undertaken to determine their stoichiometry, the role of disorder, the effect of variations in intermolecular overlap and interchain coupling, and in the case of the temperature-dependent structural studies of TTT_2I_3 the nature of its phase transitions.

All of the crystallographic data reduction and structural computations necessary for this study were done with the CRYM crystallographic computing system and an IBM computer 370/3032.

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Chapter 2
Temperature-Dependent Studies of
Highly-Disordered (Tetrathiotetracene)₂-(Iodide)₃

Introduction

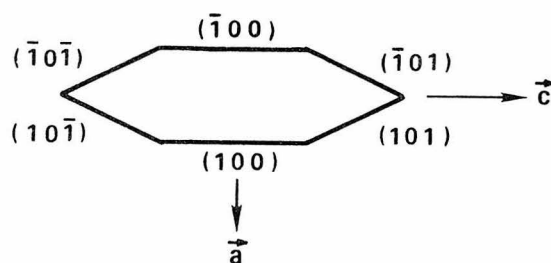
Tetrathiotetracene* (TTT) is a planar organic molecule which as long ago as 1965 was known to form organic donor-acceptor charge-transfer complexes (1,2). These early charge-transfer complexes had resistivity values of 2-15 Ω cm. Although Marschalk and coworkers were the first to report the syntheses of ion-radical salts of TTT (3,4), Perez-Albuerne and coworkers reported in 1971 that the halide and thiocyanate salts of TTT formed highly conducting powders (5). The iodide salt, of poorly defined stoichiometry, had a powder resistivity of 0.71 Ω cm. Eventually single crystals of TTT_2I_3 were grown and temperature-dependent conductivity measurements indicated that TTT_2I_3 was metallic (6,7). Additional electrical, magnetic, and optical studies of crystals of TTT_2I_3 grown at the Jet Propulsion Laboratory (JPL), Pasadena, have shown that for temperatures of $100^\circ < T < 300^\circ$ K TTT_2I_3 is metallic (8). The d.c. conductivity at room temperature is about $1000 \Omega^{-1} \text{ cm}^{-1}$ parallel to the crystal needle axis. At about 100° K abrupt changes in the conductivity, ESR linewidth, and thermoelectric power (TEP) suggest that a phase transition has occurred.

*naphthaceno[5,6-cd:11,12-c'd']bis[1,2]dithiole.

Below 20-30° K the electrical and magnetic properties indicate that TTT_2I_3 has undergone an additional phase transition to a nonmetallic state (8).

The basic structure of TTT_2I_3 is known at room temperature (6,9). However, temperature-dependent structural information would clarify the nature of the phase transitions at ~100° K and ~20°-30° K and would help clarify the role of disorder. For these reasons, an extensive temperature-dependent X-ray diffraction study of TTT_2I_3 was undertaken.

Single crystals of TTT_2I_3 are grown from hot nitrobenzene solutions of TTT and iodine (I_2) by slow-cooling (V. Hadek and R. B. Somoano, JPL). TTT_2I_3 crystals are long, dark, well-formed, six-sided orthorhombic needles which give golden reflections off the faces. The needle axis, the axis of high conductivity, is \vec{b} . For a C-centered TTT lattice, the crystallographic axes and the indices of the faces are oriented as shown below. (\vec{b} is perpendicular to the page.) The a and c axes



have almost the same lengths but can be distinguished because the 800 and $00\bar{1}2$ are weak reflections whereas the $\bar{1}200$ and 008 are both strong reflections. Two general features of X-ray oscillation photographs of crystals of TTT_2I_3 rotated about \vec{b} are the presence of layer lines with three-dimensional Bragg diffraction spots and the presence of

several incommensurate diffuse layer lines. (See, for example, Figure 15.) The lattice giving rise to the three-dimensional diffraction spots is referred to as the TTT lattice. However, this is not meant to imply that only the TTT molecules contribute to the diffraction pattern. By just slightly varying the crystallization conditions, crystals of TTT_2I_3 with varying amounts of disorder can be obtained. The variations in disorder, as observed by X-ray diffraction, have been correlated with observed variations in conductivity and thermoelectric power measurements (10, Appendix A). The variations in the amount of disorder can be determined, with oscillation photographs or diffractometer scans, by the variation in beading or diffuseness of the strongest diffuse layer line. At room temperature the highly-disordered material, TTT_2I_3 (h.d.) has very little beading on the strongest diffuse layer line. Crystals of TTT_2I_3 (h.d.) were obtained from JPL for the temperature-dependent structural studies reported here.

2.1 Room Temperature Structure

Experimental

A crystal of TTT_2I_3 (h.d.) with the dimensions $0.06 \times 0.123 \times 0.60$ mm was used for room temperature data collection. Unit cell parameters for the TTT lattice were obtained from a symmetry-constrained (orthorhombic) least-squares fit of the angular settings (2θ , ϕ , χ) of ten reflections carefully centered with a beam splitter (Table I). The room temperature data were obtained with nickel-filtered copper K_α radiation on an automatic, General Electric, quarter-circle diffractometer. Intensities were collected with 1° per minute 2θ - θ scans. Backgrounds were collected for 20 seconds on each side of the scans. The integrated intensities were corrected for Lorentz and polarization effects. Weights were based on counting statistics and corrected for proportionality error (11). No more than two observations contributed to a weight. Absorption corrections based on a rectangular solid ($\mu = 274.108 \text{ cm}^{-1}$) (12) were applied to the data (13). The final, complete data set for the TTT lattice consisted of 919 reflections of which 713 reflections had $F_o^2 \geq 3\sigma(F_o^2)$. No violations of the systematic absences hkl , $h+k$ odd; $h0l$, h and l odd; and $hk0$, h odd were observed suggesting the space group Cmca . This agrees with what Smith and Luss reported (9). Preliminary atomic coordinates were determined from a three-dimensional Patterson map. Successive improvements to the model were based on three-dimensional Fourier and difference Fourier maps. The hydrogens

Table I

Crystal Data for TTT_2I_3 (h.d.) $\sim 294^\circ \text{K}$

$$a = 18.347(3) \text{ \AA}$$

$$b = 4.962(1) \text{ \AA} \quad (\lambda = 1.5418 \text{ \AA})$$

$$c = 18.454(3) \text{ \AA}$$

$$\alpha = \beta = \gamma = 90.0^\circ$$

$$V = 1680.0(9)$$

$$Z = 2$$

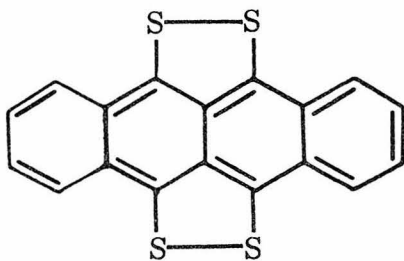
$$d_c = 2.147$$

Orthorhombic, Space group Cmca

Linear absorption coefficient (CuK_α), $\mu = 274.108 \text{ cm}^{-1}$

Crystal dimensions $0.123 \text{ mm} \times 0.60 \text{ mm} \times 0.06 \text{ mm}$

$\text{TTT} \equiv$



were included in the refinement after they were observed on a difference Fourier map. After trial and error the electron density in the iodide channels was determined to be best fit by using two iodine atoms with variable populations and highly anisotropic temperature factors. The iodine atoms in the model are actually correlated because they both represent a single broad maximum of electron density and, therefore, only their populations and temperature factors were allowed to vary during refinement. The best coordinates for the iodine atoms were determined from difference Fourier sections. With the exception of the iodine atomic coordinates, all of the parameters were refined by the full-matrix least-squares method. Structure factors were calculated using the real part ($f_0 + \Delta f'$) of sulfur and iodine atomic scattering factors which had been corrected for anomalous dispersion (14). The refinement converged to $R = 0.061$, and the goodness-of-fit = 3.96 for the entire data set (15). For 713 reflections with $F_o^2 \geq 3\sigma(F_o^2)$, $R = 0.054$. A final three-dimensional difference Fourier map showed residual density ranging from $-0.7\text{e}/\text{\AA}^3$ to $+0.6\text{e}/\text{\AA}^3$.

Discussion

Listed in Table II are the atomic coordinates and thermal parameters for the room-temperature structure. Shown in Figure 1 is an [010] projection of the structure. The TTT molecule has $2/m$ symmetry and only one-quarter of the molecule is crystallographically unique. The iodine atoms are in channels which parallel \vec{b} . As schematically shown

Table II

Atomic Coordinates and Thermal Parameters for TTT_2I_3 (h.d.) at $\sim 294^\circ \text{K}$ (Copper K_α)

	$\underline{\text{X}}$	$\underline{\text{Y}}$	$\underline{\text{Z}}$	<u>Populations</u>				
I1	.25	.323	.25	.8426(795)				
I2	.25	.198	.25	.5684(792)				
C1	.03875(26)	0	0					
C2	.07609(20)	.17861(87)	.04433(21)					
C3	.03965(21)	.36659(84)	.08902(19)					
C4	.07602(25)	.55283(100)	.13418(22)					
C5	.03919(29)	.73078(106)	.17562(21)					
S	.17080(6)	.15216(28)	.03878(7)					
	$\underline{\text{U}}_{11}$	$\underline{\text{U}}_{22}$	$\underline{\text{U}}_{33}$	$\underline{\text{U}}_{12}$	$\underline{\text{U}}_{13}$	$\underline{\text{U}}_{23}$		
I1	.0489(17)	.3425(301)	.0575(17)	0	-.0120(14)	0		
I2	.0568(27)	.3680(419)	.0585(23)	0	.0289(29)	0		
C1	.0337(25)	.0270(31)	.0268(21)	0	0	.0022(21)		
C2	.0330(16)	.0305(26)	.0323(17)	-.0007(17)	.0007(14)	.0012(17)		
C3	.0442(20)	.0280(22)	.0286(15)	-.0016(20)	-.0038(15)	.0028(16)		
C4	.0517(21)	.0341(28)	.0321(17)	-.0043(22)	-.0082(17)	-.0020(18)		
C5	.0774(30)	.0344(27)	.0293(17)	-.0068(26)	-.0074(21)	-.0021(18)		
S	.0345(5)	.0456(7)	.0594(7)	-.0035(6)	-.0030(5)	-.0085(6)		
	$\underline{\text{X}}$	$\underline{\text{Y}}$	$\underline{\text{Z}}$	$\underline{\text{B}}$				
H1	.1308(32)	.5516(116)	.1279(29)	5.11(1.29)				
H2	.0674(29)	.8876(106)	.2040(29)	4.68(1.25)				

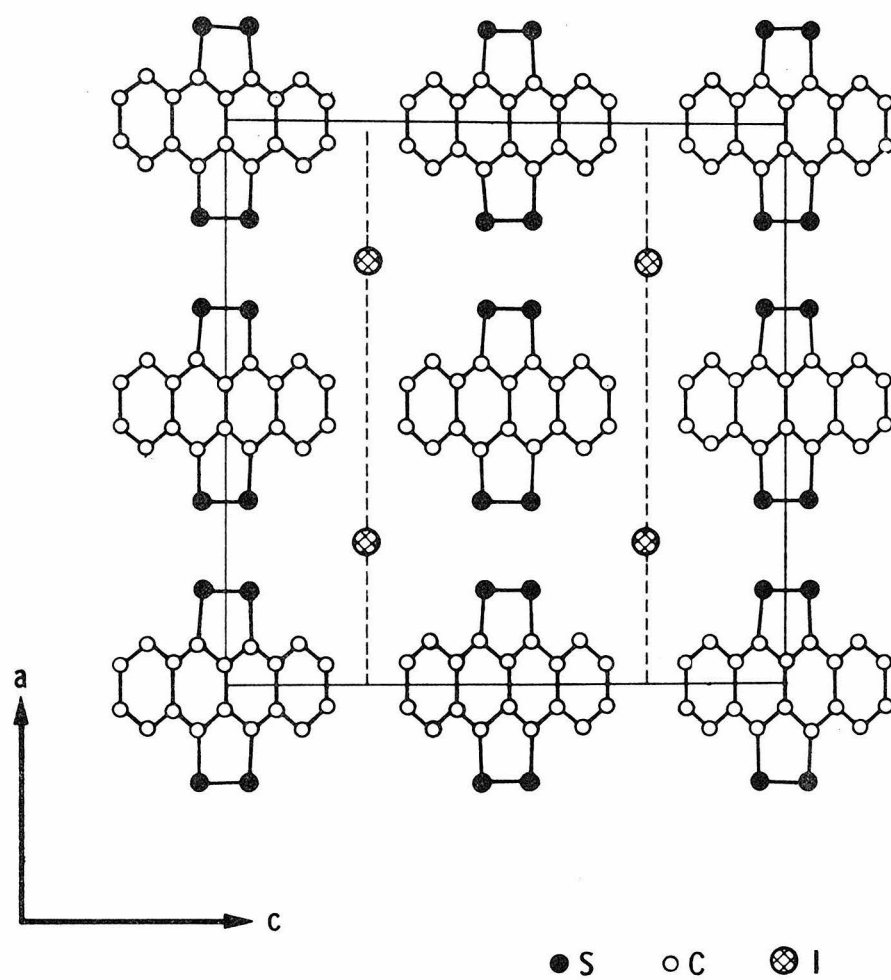
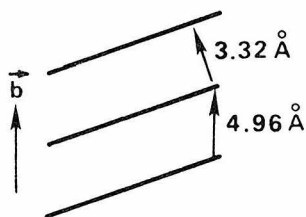
PROJECTION DOWN b-AXIS OF TTT_2I_3 

Figure 1

below, each TTT molecule is slipped with respect to neighboring molecules, but the entire stack is parallel \vec{b} .



There are uniform TTT-TTT spacings along the stack because a TTT to TTT translation along \vec{b} defines the lattice spacing.

As can be seen in Tables III, XI and Figure 2 (16) the TTT molecule is the same as reported by Smith and Luss (Kodak) (9) and very similar to neutral TTT (17). In general there are no features in this room temperature structure which might suggest that there is any significant difference in the TTT framework between the TTT_2I_3 crystals from JPL and those from Kodak. From the description given by Buravov, *et al.*, (Chernogolovka) (6) their TTT_2I_3 crystals also have the same framework. Mihály, Jánosy, and Grüner (Budapest) did not report any structural data but apparently believe that their TTT_2I_3 crystals have the same structure as the Kodak and Chernogolovka crystals (18). However, the physical properties of TTT_2I_3 crystals are dependent on which laboratory grew the crystals. This suggests that variations in the properties of TTT_2I_3 may arise from variations in the iodide channels.

Shown in Figure 3 is the electron density of the iodide channels based on the refinement of the TTT lattice. Shown in Figure 4 are the corresponding difference Fourier sections which indicate that a

Table III

TTT Geometry in TTT_2I_3 (h.d.) $\sim 294^\circ \text{ K}$ (CuK_α)

<u>Bond Lengths, Å[*]</u>		<u>Interior Bond Angles[*]</u>	
C1-C1	1.422(10)	C1-C1-C2	119.6(8) ⁰
C3-C3	1.455(8)	C1-C2-S	114.1(3)
C5-C5	1.438(10)	C2-S-S	95.5(2)
C1-C2	1.387(7)	C2-C1-C2	120.8(8)
C2-C3	1.413(6)	C1-C2-C3	122.2(4)
C3-C4	1.412(6)	C2-C3-C3	118.2(7)
C4-C5	1.349(7)	C3-C3-C4	118.2(7)
S-C2	1.746(4)	C3-C4-C5	121.7(4)
S-S	2.080(2)	C4-C5-C5	120.1(8)
H1-C4	1.01(6)	H1-C4-C3	113(3)
H2-C5	1.07(5)	H1-C4-C5	125(3)
		H2-C5-C4	121(3)
		H2-C5-C5	119(3)

Direction cosines of the plane normal 0.0, -0.6700, 0.7424 (best plane calculation of inner carbon rings).

TTT tilt angle to \vec{b} 47.93° .

Interplanar spacing 3.324 Å ($b = 4.962 \text{ Å}$).

*The deviation in the last decimal place is given in parentheses (assuming isotropic atoms).

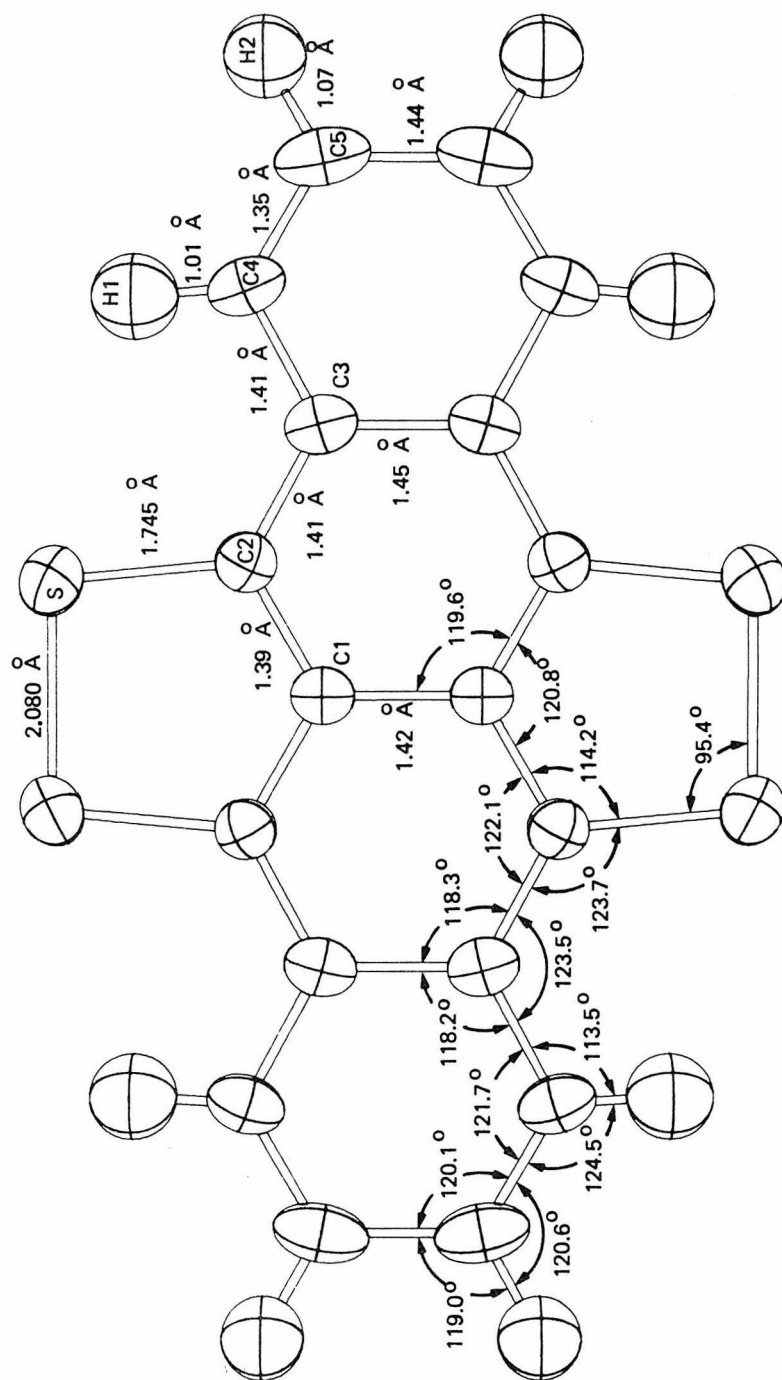
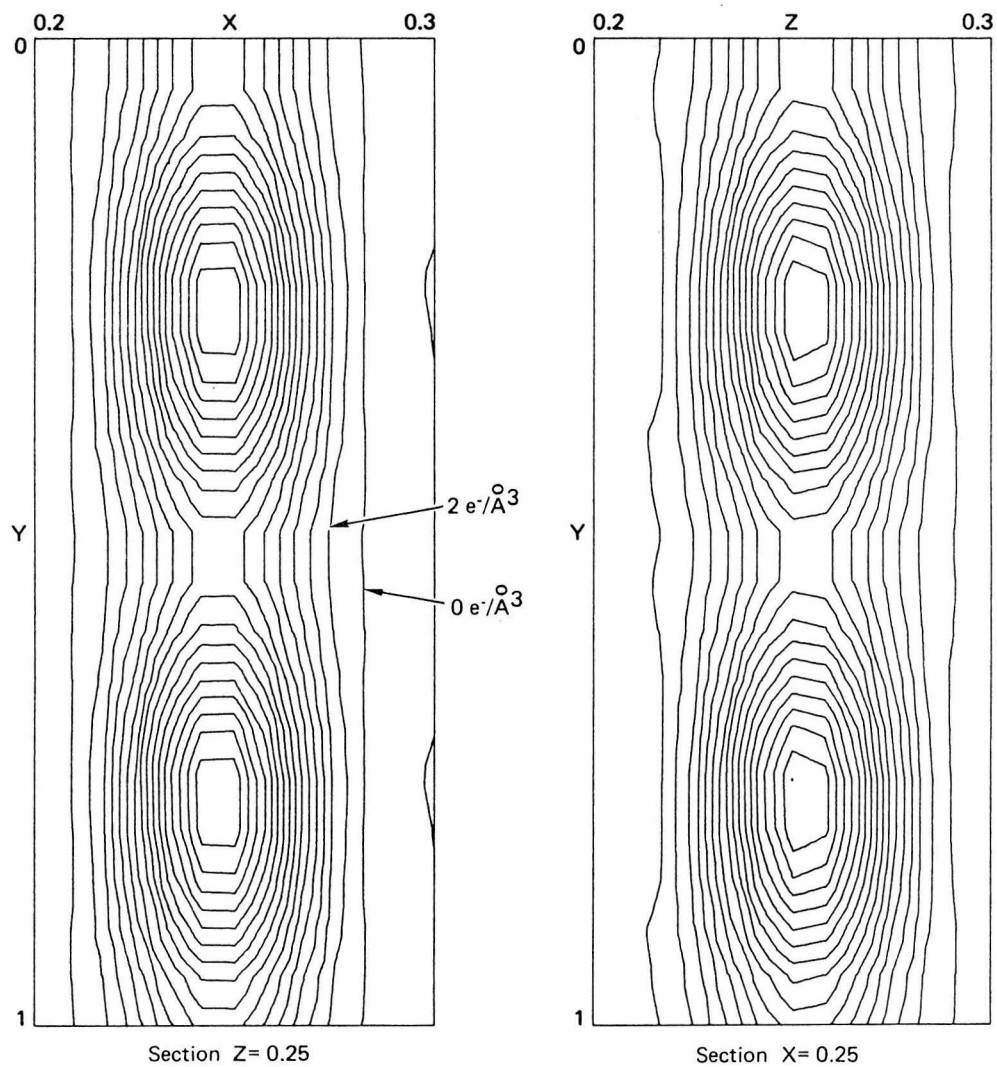


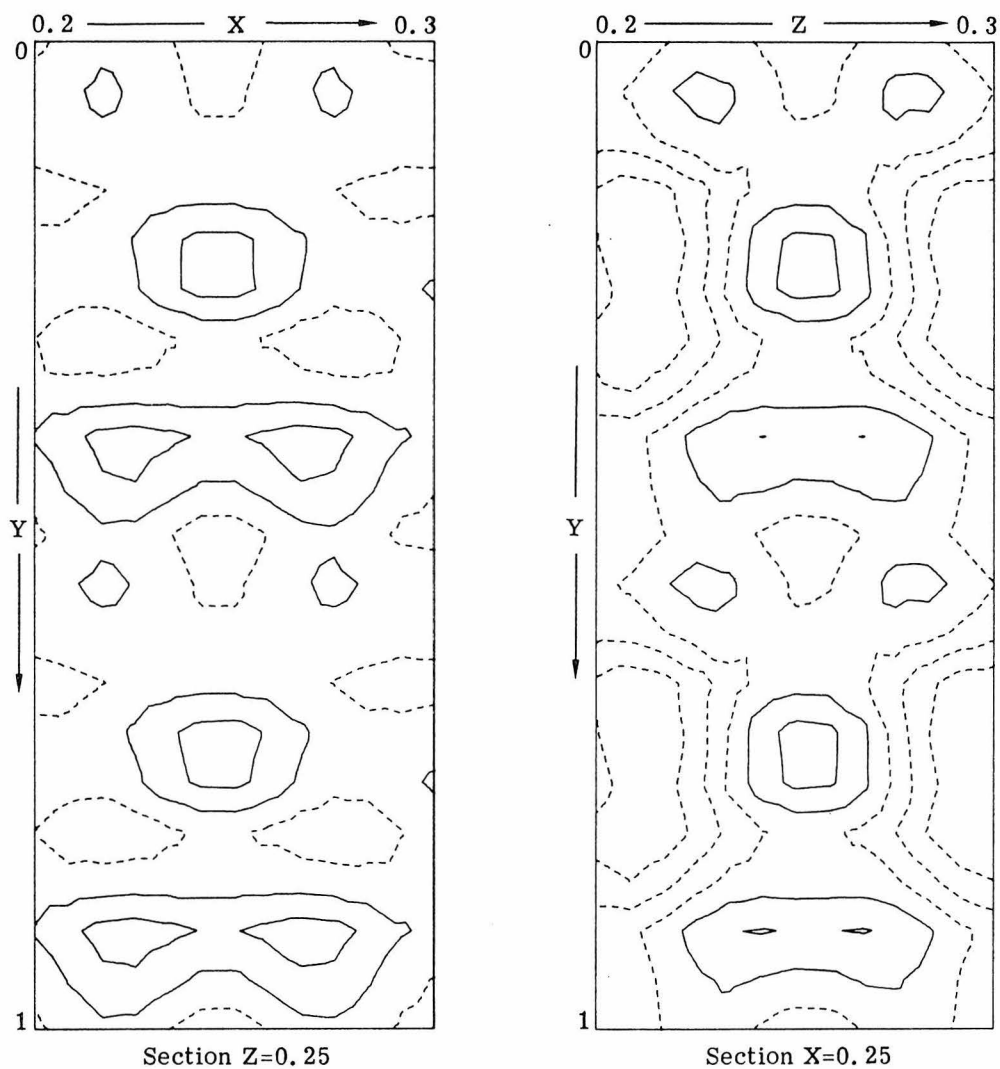
Figure 2

Tetrathiotetracene cation in high disorder TTT_2I_3 at room temperature
 Thermal ellipsoids are drawn at the 50% probability level



TTT₂I₃ (h.d.) - Room Temperature
 Electron Density Along Iodine Channel
 Contoured at 2 electrons/Å³ Intervals

Figure 3



TTT_2I_3 (h. d.) - Room temperature
 Difference Fourier Sections through Iodine Channel
 Contoured at 0.2 electrons / \AA^3 Intervals
 ----- negative contours
 ————— zero and positive contours

Figure 4

reasonable fit has been obtained for the observed and calculated electron density. Figure 3 shows that there are not discrete iodide sites in a spacing of $b = 4.96 \text{ \AA}$. The two broad maxima are related by symmetry and are only $4.96 \text{ \AA}/2 = 2.48 \text{ \AA}$ apart. This distance is too short for any chemically reasonable iodide species. That there is also significant electron density at the minima, suggests that Figure 3 represents an average of, perhaps, several iodide configurations present in different channels throughout the structure. This could be caused by slippage of the iodide chains parallel to \vec{b} . The minimum and maximum are $12.9 \text{ e}^-/\text{\AA}^3$ and $32.9 \text{ e}^-/\text{\AA}^3$ respectively. The refined population, the only iodine parameter from Table II of any significance, gives a stoichiometry of $\text{TTTI}_{1.41 \pm 0.08}$. The Kodak crystals exhibit minima and maxima of $16 \text{ e}^-/\text{\AA}^3$ and $37 \text{ e}^-/\text{\AA}^3$ for a refinement based on a stoichiometry of $\text{TTTI}_{1.5}$. (The iodine population was not refined.) The difference in the maximum and minimum electron densities between TTT_2I_3 (h.d.) and the Kodak crystals could be due to different size grids in the Fourier maps. The similarities between the structural and photographic data reported for Kodak crystals (9) and the data obtained for TTT_2I_3 (h.d.) suggest that the physical properties of crystals from Kodak and JPL ought to be similar, as, in fact, they are (10). It is unfortunate that there is no detailed structural information available for the Chernogolovka or Budapest crystals for comparison with the Kodak and JPL crystals. Detailed information about the Chernogolovka crystals would be especially interesting because those crystals have different properties from those of JPL or Kodak.

2.2 Low-Temperature Structures

Experimental

For the low-temperature studies a new, locally designed and built, goniometer was used. The goniometer is used on a modified, full-circle Syntex PT automated diffractometer. The outer shell of the goniometer is a sphere which has kapton-sealed entrance and exit slits for the X-rays. The volume enclosed by the sphere is evacuated. Inside the crystal is enclosed by two additional, cooled, inner shells. The innermost copper shell surrounding the crystal is attached directly, mechanically and thermally, to the extended tip of the cryocooler (extender). The cryocooler is a Cryogenics Technology Inc. closed-cycle helium gas refrigerator rated to cool to $\sim 10^{\circ}$ – 14° K. The crystal is cooled by a copper heat-transfer cable which thermally links the top of the phi-shaft to the cold, inner shell (19).

The crystal of TTT_2I_3 (h.d.) used for data collection, the fifth one of six studied, was a six-sided needle 0.333 mm long and had a cross-sectional area of 0.00686 mm^2 . The crystal was mounted on a glass fiber with Armstrong low-temperature epoxy resin. The glass fiber was mounted onto a copper pin which was inserted directly into the copper block at the top of the phi-shaft. Crystal alignment was accomplished by bending the copper pin. At room temperature the crystal was set high, above the center of the diffractometer, to allow for shrinkage at low temperatures. Temperatures were continuously

monitored at the extender (extended tip of the cryocooler) and just underneath the copper pin at the top of the phi-shaft. The temperature of the innermost copper shell could also be measured. At temperatures between 18.5° and 25.5° K the diode measuring the phi-shaft temperature gave correct, accurate readings. Between 26.5° K and liquid N₂ (77.4° K) the diode gave low readings, the largest correction being +4.7° at 35.5° K. No calibration was done at higher temperatures. Unless otherwise stated, the temperatures reported are the observed uncorrected temperatures at the phi-shaft. The rate at which the crystal was cooled could be controlled by how often the temperature at the extender was lowered. From room temperature (~294° K) to ~164° K the fifth crystal was cooled at 10° per 30-60 minutes in 2° steps. After every 10° the crystal was equilibrated for one to twelve hours while peak intensities and scans of several reflections were obtained.

At 164° K a data set was collected with monochromatized molybdenum K_α radiation. The unit cell parameters listed in Table IV were obtained from a constrained least-squares fit of the 2θ values for thirty computer-centered reflections (20). The symmetry during the least-squares fit was constrained to be orthorhombic. Intensities of the TTT lattice reflections were collected with 1° per minute 2θ-θ scans. Total time of background counting was 25% of the amount of time spent scanning a reflection. Backgrounds were collected on each side of a reflection. The observed range of temperatures at the

Table IV

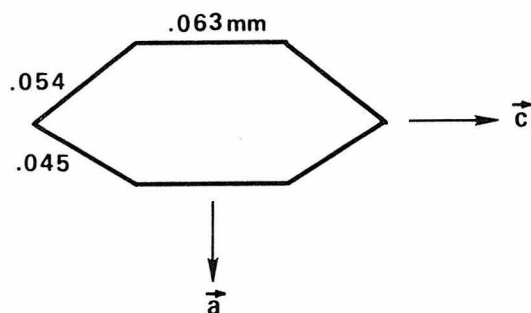
Crystal Data for TTT_2I_3 (h.d.)

Orthorhombic, Space group Cmca

$Z = 2$

Linear absorption coefficient (MoK_α), $\mu = 33.5225 \text{ cm}^{-1}$.

Crystal dimensions: length 0.333 mm
cross-sectional area 0.00686 mm^2



<u>$\sim 164^\circ \text{ K}$</u>	<u>$\sim 74^\circ \text{ K}$</u>	<u>$\sim 19^\circ \text{ K}$</u>
$a = 18.285(3) \text{ \AA}$	$18.246(4) \text{ \AA}$	$18.225(3) \text{ \AA}$
$b = 4.957(1)$	$4.947(1)$	$4.950(1)$
$c = 18.351(3)$	$18.284(5)$	$18.242(6)$
$V = 1663(1) \text{ \AA}^3$	$1650(1) \text{ \AA}^3$	$1646(1) \text{ \AA}^3$

(MoK_α , $\lambda = 0.71069 \text{ \AA}$)

phi-shaft during data collection was 163.7° and 167° K. The temperature at the extender varied by $\pm 0.1^\circ$.

After 366 hours of data collection cooling was resumed. The rate of cooling was 10°/hour in 1° steps with peak scans every ten degrees. Below about 90° K the crystal was cooled at 5°/hour in 0.5° steps with peak scans every ten degrees to ~74° K. At 74° K a data set was again collected with monochromatized molybdenum K_α radiation. The unit cell parameters listed in Table IV were obtained from a symmetry-constrained least-squares fit of the 2θ values of twenty-nine computer-centered reflections (20). Intensities were collected as at 164° with 1° per minute 2θ - θ scans. The total time for background counting was 20% of the amount of time for each scan at low angle and 30% at high 2θ values. The background counting time was changed during the data collection if the scan width was also changed. The percentage was chosen so that the background on each side of a reflection was collected for 25-45 seconds. A shell of data for a primitive cell was collected to check that the centering condition had not changed. No violations were observed. During the data collection the average temperature at the phi-shaft was about 73°-74° K with a total range of about three degrees. The innermost cold shell surrounding the crystal was at 62.7° K.

After 377 hours of data collection, cooling was resumed. The temperature was lowered at 0.1° per minute with equilibration and peak scans every ten degrees until the lowest temperature, 20° K, was reached. During the first fifty-five hours of data collection the phi-shaft temperature gradually dropped from ~20° K to 18.9° K at which

point the heater was turned on to prevent any further temperature drop. During the remainder of the data collection the temperature at the phi-shift remained between 18.6° and 19.6° K. Intensities were collected as before with 1° per minute 2θ - θ scans. Total time for background counting was 25% of the scan time per reflection for low-angle reflections and 35% at high angles. Part of the data was collected for a primitive cell to check for violations of the lattice centering. In addition to the complete data set obtained for the octant $\bar{h}k\ell$, part of octants $\bar{h}k\ell$ and $hk\ell$ were also collected to check if the lattice symmetry had changed. No violations of the orthorhombic lattice symmetry or of the C-centering were observed. Towards the end of the data collection a slight decay of about 2% in the intensities of the check reflections was observed. The unit cell parameters listed in Table IV were obtained from a symmetry-constrained least-squares fit of twenty-two computer-centered reflections (20).

The integrated intensities for all three sets of data of the TTT lattice were corrected for Lorentz and polarization effects and for absorption ($\mu = 33.5225 \text{ cm}^{-1}$) (12,13). No correction was made for the slight decay observed towards the end of the data set collected at 19° K. The weights used in refinement were based on counting statistics. No more than two observations contributed to a weight. The F_o for reflections with more than one observation is a weighted average of the F^2 values (21). No reflections were observed at 164° K, 74° K, or 19° K which would violate the space group Cmca.

Initial atomic coordinates for the structure at 164° K were obtained from a three-dimensional Patterson map. Successive iterations of structure factor calculations followed by Fourier synthesis were used until all the atomic coordinates, except for the hydrogen atoms, were known. The real part ($f_0 + \Delta f'$) of the atomic scattering factor corrected for anomalous dispersion was used for iodine and sulfur for the structures at all three temperatures (14). As at room temperature two iodine atoms were used to fit the electron density in the channels at $1/4, y, 1/4$. Their positions were determined from difference Fourier sections. All of the structural parameters, except the iodine coordinates, were refined by the full-matrix least-squares method. After the structure was partially refined, the hydrogen atoms were observed on a three-dimensional difference Fourier map. The hydrogen parameters were included in the final least-square cycles. The refinement converged to $R = 0.082$, with a goodness-of-fit = 1.42 for the complete data set of 1338 reflections of which 1160 reflections had F_0^2 greater than zero (15). For the 731 reflections with $F_0^2 \geq 3\sigma(F_0^2)$, $R = 0.042$.

The carbon and sulfur atomic coordinates from the 164° K structure were used as input to the initial structure factor calculations for the 74° K structure. The iodine atomic coordinates were determined from Fourier and difference Fourier sections. After preliminary refinement the hydrogen atoms were located on a low-angle difference Fourier map. Full-matrix least-squares refinement was used to refine all of the structural parameters except for the iodine coordinates. The structure

in space group Cmca converged to $R = 0.081$, with a goodness-of-fit = 1.42 for a complete data set of 1670 reflections of which 1506 had F_o^2 greater than zero. For the 982 reflections with $F_o^2 \geq 3\sigma(F_o^2)$, $R = 0.042$.

Because there were no observed violations of the systematic absences for the space group Cmca or any indications for a change in lattice symmetry, initial structure factor calculations for the structure at 19° K were based on the sulfur and inner-ring carbon atom coordinates at 74° K. The atomic coordinates were gradually added through successive iterations of structure factor-Fourier calculations. The data set at 19° K had $h0l$ reflections to $2\theta = 108^\circ$ and hkl , $k > 0$ reflections to $2\theta = 70^\circ$, so a low-angle cut-off of the data was used during the early phase of structure determination. The electron density in the channel could still be fit reasonably well with two iodine atoms representing the broad maximum. For the final full matrix least-squares refinement the complete data set was used. For the space group Cmca the refinement converged to $R = 0.073$, with the goodness-of-fit = 1.35 for the complete data set of 1984 reflections, of which 1792 reflections had F_o^2 greater than zero. For the 1194 reflections which had $F_o^2 \geq 3\sigma(F_o^2)$, $R = 0.039$.

Listed in Tables V, VI, and VII are the atomic coordinates and thermal parameters. The bond lengths and bond angles for the refined structures at 164° K, 74° K, and 19° K are listed in Tables VIII, IX, and X. Complete lists of the observed and calculated structure factors are in Appendix C.

Table V

Atomic Coordinates and Thermal Parameters for TTT_2I_3 (h.d.) at $\sim 164^\circ \text{K}$
(MoK_α)

	<u>X</u>	<u>Y</u>	<u>Z</u>	<u>Populations</u>		
I1	.25	.325	.25	1.1335(183)		
I2	.25	.182	.25	.4655(180)		
C1	.03904(21)	0	0			
C2	.07679(15)	.18186(58)	.04458(16)			
C3	.03948(16)	.36770(63)	.08940(15)			
C4	.07619(20)	.55735(67)	.13443(17)			
C5	.03876(19)	.73805(75)	.17622(17)			
S	.17154(5)	.15337(18)	.03883(5)			

	<u>U₁₁</u>	<u>U₂₂</u>	<u>U₃₃</u>	<u>U₁₂</u>	<u>U₁₃</u>	<u>U₂₃</u>
I1	.0240(7)	.3519(52)	.0319(7)	0	-.0050(5)	0
I2	.0332(27)	.2022(149)	.0367(25)	0	.0239(16)	0
C1	.0136(19)	.0136(21)	.0148(17)	0	0	.0025(20)
C2	.0134(13)	.0143(15)	.0190(12)	-.0024(15)	-.0011(11)	.0045(15)
C3	.0209(15)	.0142(14)	.0151(12)	-.0017(16)	-.0007(11)	.0034(15)
C4	.0249(17)	.0188(17)	.0169(13)	-.0037(18)	-.0056(13)	.0036(14)
C5	.0431(22)	.0196(16)	.0134(12)	-.0033(18)	-.0055(14)	-.0009(15)
S	.0151(4)	.0235(4)	.0325(4)	-.0020(5)	-.0021(4)	-.0051(4)

	<u>X</u>	<u>Y</u>	<u>Z</u>	<u>B</u>
H1	.1298(17)	.5594(61)	.1333(16)	1.52(0.71)
H2	.0667(18)	.8703(66)	.2064(18)	2.11(0.77)

Table VI

Atomic Coordinates and Thermal Parameters for TTT_2I_3 (h.d.) at $\sim 74^\circ \text{K}$
(MoK_α)

	<u>X</u>	<u>Y</u>	<u>Z</u>	<u>Populations</u>		
I1	.25	.328	.25	1.1230(138)		
I2	.25	.189	.25	.3757(135)		
C1	.03913(18)	0	0			
C2	.07685(12)	.18273(53)	.04472(13)			
C3	.03972(13)	.36936(55)	.08954(12)			
C4	.07657(15)	.56009(58)	.13460(13)			
C5	.03894(15)	.74221(65)	.17640(12)			
S	.17211(4)	.15388(15)	.03901(3)			
	<u>U₁₁</u>	<u>U₂₂</u>	<u>U₃₃</u>	<u>U₁₂</u>	<u>U₁₃</u>	<u>U₂₃</u>
I1	.0125(5)	.3318(39)	.0162(4)	0	-.0032(4)	0
I2	.0148(17)	.1879(105)	.0159(12)	0	.0128(9)	0
C1	.0086(14)	.0107(17)	.0095(11)	0	0	.0015(14)
C2	.0091(10)	.0096(12)	.0122(8)	-.0009(12)	-.0012(8)	.0022(10)
C3	.0120(10)	.0093(12)	.0104(8)	.0000(12)	-.0015(8)	.0027(10)
C4	.0152(12)	.0130(14)	.0111(8)	-.0011(14)	-.0018(9)	.0014(10)
C5	.0247(14)	.0106(12)	.0105(8)	-.0018(14)	-.0029(9)	.0006(11)
S	.0092(3)	.0131(3)	.0200(3)	-.0011(4)	-.0011(2)	-.0029(3)
	<u>X</u>	<u>Y</u>	<u>Z</u>	<u>B</u>		
H2	.0671(15)	.8734(59)	.2085(14)	.73(0.58)		
H1	.1297(15)	.5595(57)	.1326(14)	.72(0.59)		

Table VII

Atomic Coordinates and Thermal Parameters for TTT_2I_3 (h.d.) at $\sim 19^\circ \text{K}$
(MoK_α)

	<u>X</u>	<u>Y</u>	<u>Z</u>	<u>Populations</u>		
I1	.25	.329	.25	1.014(211)		
I2	.25	.201	.25	.4916(211)		
C1	.03937(15)	0	0			
C2	.07700(10)	.18251(44)	.04485(12)			
C3	.03965(11)	.37076(48)	.08976(11)			
C4	.07696(12)	.56103(51)	.13477(12)			
C5	.03897(13)	.74358(57)	.17650(11)			
S	.17235(3)	.15438(13)	.03909(3)			

	<u>U₁₁</u>	<u>U₂₂</u>	<u>U₃₃</u>	<u>U₁₂</u>	<u>U₁₃</u>	<u>U₂₃</u>
I1	.0045(3)	.3132(63)	.0072(4)	0	-.0013(3)	0
I2	.0115(10)	.2624(132)	.0081(9)	0	.0054(6)	0
C1	.0068(9)	.0085(13)	.0066(9)	0	0	.0003(12)
C2	.0073(7)	.0066(9)	.0089(7)	.0002(9)	-.0006(6)	.0020(8)
C3	.0078(6)	.0074(9)	.0076(6)	.0014(9)	-.0001(6)	.0018(8)
C4	.0110(8)	.0107(10)	.0084(7)	-.0010(10)	-.0026(6)	.0005(8)
C5	.0174(9)	.0099(9)	.0064(7)	.0002(11)	-.0020(7)	.0006(9)
S	.0066(2)	.0093(2)	.0149(2)	-.0005(3)	-.0012(2)	-.0022(3)

	<u>X</u>	<u>Y</u>	<u>Z</u>	<u>B</u>
H1	.1281(17)	.5641(61)	.1322(17)	1.58(0.68)
H2	.0672(17)	.8788(65)	.2088(18)	1.75(0.70)

Table VIII

TTT Geometry in TTT_2I_3 (h.d.) $\sim 164^\circ$ K (MoK_α)

Bond Lengths, Å*		Interior Bond Angles*	
C1-C1	1.428(8)	C1-C1-C2	119.5(6)°
C3-C3	1.444(6)	C1-C2-S	113.8(2)
C5-C5	1.417(7)	C2-S-S	95.8(2)
C1-C2	1.399(5)	C2-C1-C2	120.9(6)
C2-C3	1.411(4)	C1-C2-C3	121.5(3)
C3-C4	1.420(4)	C2-C3-C3	118.9(6)
C4-C5	1.363(5)	C3-C3-C4	118.2(6)
S-C2	1.742(3)	C3-C4-C5	121.7(3)
S-S	2.084(2)	C4-C5-C5	120.1(6)
H1-C4	0.98(3)	H1-C4-C3	118(2)
H2-C5	0.999(33)	H1-C4-C5	120(2)
		H2-C5-C4	119(2)
		H2-C5-C5	121(2)

Direction cosines of the plane normal 0.0, -0.6690, 0.7433 (Best plane calculation of inner carbon rings).

TTT tilt angle to \vec{b} 48.02° .

Interplanar spacing 3.316 Å .

*The standard deviations in the last decimal place are given in parentheses (assuming isotropic atoms).

Table IX

TTT Geometry in TTT_2I_3 (h.d.) $\sim 74^\circ$ K (MoK_α)

Bond Lengths, Å*		Internal Bond Angles*	
C1-C1	1.428(7)	C1-C1-C2	119.5(6)°
C3-C3	1.449(5)	C1-C2-S	113.7(2)
C5-C5	1.421(6)	C2-S-S	95.8(1)
C1-C2	1.400(4)	C2-C1-C2	121.1(5)
C2-C3	1.408(3)	C1-C2-C3	121.8(2)
C3-C4	1.422(4)	C2-C3-C3	118.8(4)
C4-C5	1.367(4)	C3-C4-C5	121.6(2)
S-C2	1.747(5)	C4-C5-C5	120.2(5)
S-S	2.087(1)	H1-C4-C3	117(2)
H1-C4	0.97(3)	H1-C4-C5	122(2)
H2-C5	1.01(3)	H2-C5-C4	119(2)
		H2-C5-C5	120(2)

Direction cosines of the plane normal 0.0, -0.6673, 0.7448 (best plane calculation of inner carbon rings).

TTT tilt angle to \vec{b} 48.14° .

Interplanar spacing 3.301 Å .

*The deviations in the last decimal place are given in parentheses (assuming isotropic atoms).

Table X

TTT Geometry in TTT_2I_3 (h.d.) $\sim 19^\circ \text{ K}$ (MoK_α)

Bond Lengths, Å*		Internal Bond Angles*	
C1-C1	1.435(6)	C1-C1-C2	119.4(5)°
C3-C3	1.445(4)	C1-C2-S	113.7(2)
C5-C5	1.420(5)	C2-S-S	95.7(1)
C1-C2	1.399(3)	C2-C1-C2	121.3(4)
C2-C3	1.415(3)	C1-C2-C3	121.9(2)
C3-C4	1.423(3)	C2-C3-C3	118.7(4)
C4-C5	1.369(3)	C3-C4-C5	121.1(2)
S-C2	1.747(2)	C4-C5-C5	120.4(4)
S-S	2.090(1)	H1-C4-C3	117(2)
H1-C4	0.93(3)	H1-C4-C5	121(2)
H2-C5	1.03(3)	H2-C5-C4	120(2)
		H2-C5-C5	120(2)

Direction cosines of the plane normal 0.0, -0.6657, 0.7462 (best plane calculation of inner carbon rings).

TTT tilt angle to \vec{b} 48.26° .

Interplanar spacing 3.295 Å .

*The deviations in the last decimal place are given in parentheses (assuming isotropic atoms).

Discussion

ORTEP drawings (16) of the TTT molecule at 164° K, 74° K, and 19° K are shown in Figures 5, 6, and 7. There is nothing unusual with the geometry of the TTT molecules. Table XI lists, for comparison, the bond lengths of neutral TTT (17), the bond lengths of TTT in Kodak's TTT_2I_3 , and the bond lengths as a function of temperature for TTT_2I_3 (h.d.). The geometry of TTT in all of the structures is very similar. The S-S bond seems to lengthen with decreasing temperature. The orientation of adjacent TTT molecules in the same stack remains about the same at 19° K as at room temperature (Figure 8). The angle of the TTT tilt to the b axis increases slightly with decreasing temperature. The interplanar distance between TTT molecules decreases as the temperature is lowered (Tables VIII, IX, X).

A comparison of Figure 3 with Figure 9 indicates that, except for a loss of thermal motion perpendicular to the channel and a general increase in coherently scattering electron density, the channel structure remains the same at 164° K as at room temperature (~294° K). At 294° K the maximum and minimum electron density in the channel are 12.9 $\text{e}/\text{\AA}^3$ and 32.9 $\text{e}/\text{\AA}^3$. At 164° K the minimum and maximum are 26.9 $\text{e}/\text{\AA}^3$ and 58.4 $\text{e}/\text{\AA}^3$. Although the amount of coherently scattering electron density in the channel continues to increase with decreasing temperature, the channel structure at 19° K is essentially the same as at room temperature (Figures 11 and 13). The position of the broad maximum does shift a little (Table XII). Figures 10, 12, and 14 are the

Table XI

Comparison of TTT Molecular Geometry

	TTT ^a	TTT ₂ I ₃ ^b	TTT ₂ I ₃ (h.d.) ^c	164° K	TTT ₂ I ₃ (h.d.) ^d 74° K	19° K
C1-C1	1.42(1)	1.429(8)	1.422(10)	1.428(8)	1.428(7)	1.435(6)
C1-C2	1.41(1)	1.390(4)	1.387(6)	1.399(5)	1.400(4)	1.398(4)
C2-C3	1.37(1)	1.414(4)	1.413(6)	1.411(4)	1.408(3)	1.415(3)
C3-C3	1.46(1)	1.444(6)	1.455(8)	1.444(6)	1.449(5)	1.445(4)
C3-C4	1.45(1)	1.410(5)	1.412(6)	1.420(4)	1.422(4)	1.423(3)
C4-C5	1.37(1)	1.358(5)	1.350(7)	1.363(5)	1.367(4)	1.369(3)
C5-C5	1.42(1)	1.404(8)	1.438(10)	1.417(7)	1.421(6)	1.420(4)
C2-S	1.78(1)	1.738(3)	1.746(4)	1.742(3)	1.747(3)	1.747(2)
S-S	2.100(3)	2.078(2)	2.080(2)	2.084(2)	2.086(1)	2.090(1)
inter- planar spacing ^e		3.32(2)	3.324(2)	3.316(2)	3.301(2)	3.295(2)

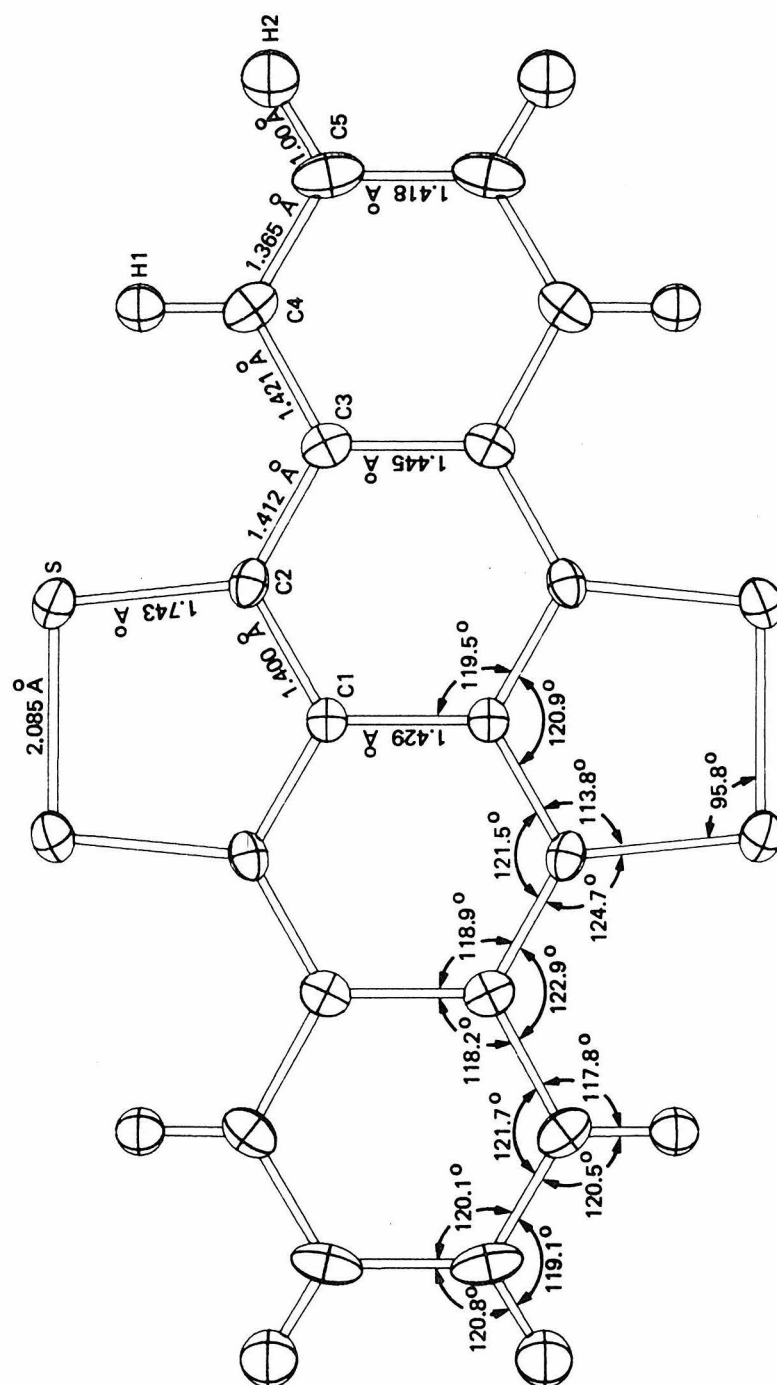
^aDideberg and Toussaint (1974) (17), CuK_α, room temperature. Neutral TTT crystallizes in space group P1̄; the bond lengths in this table are an average of the values reported by Dideberg and Toussaint.

^bSmith and Luss (1977) (9), MoK_α, room temperature.

^cThis study, CuK_α, room temperature.

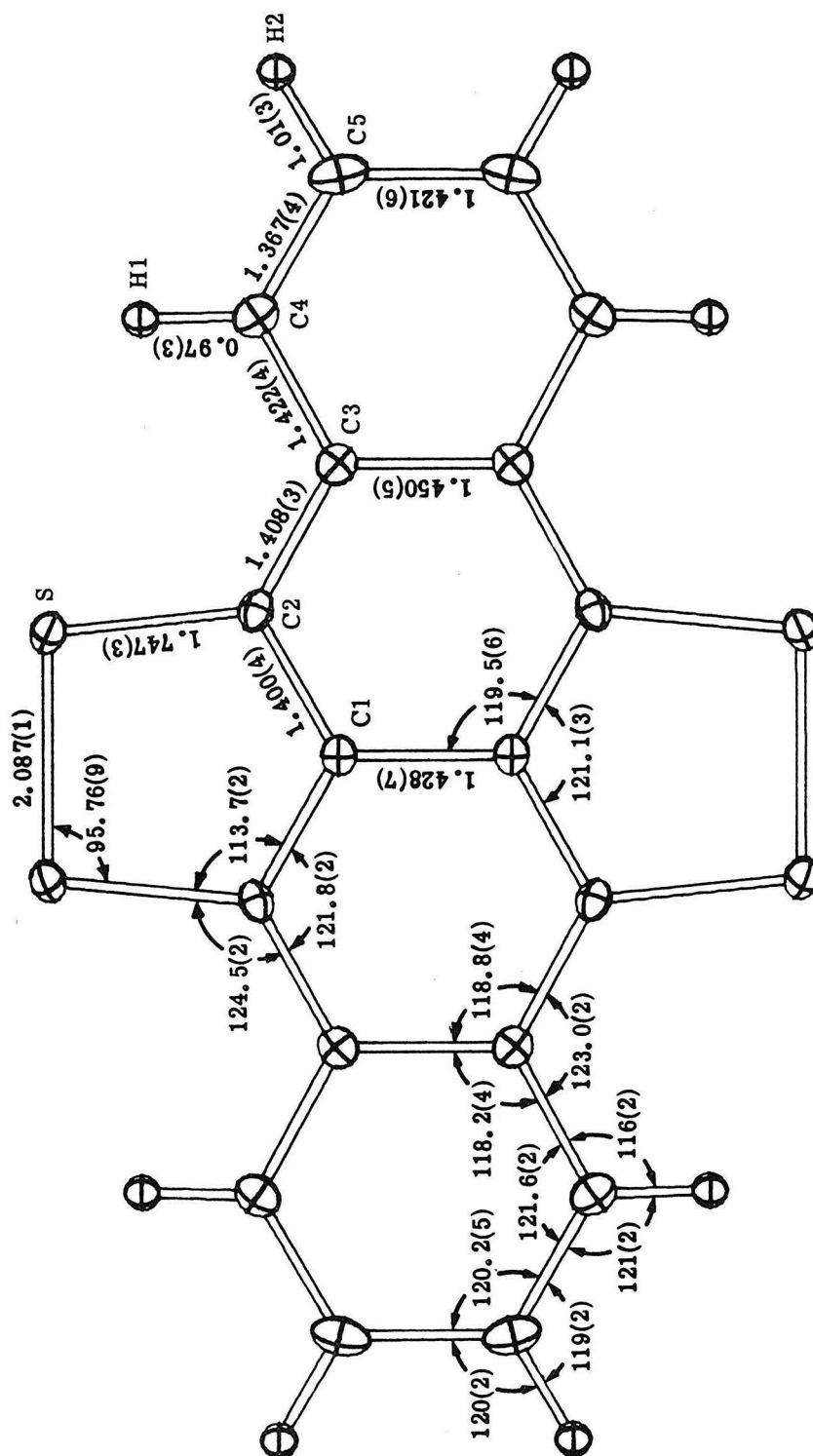
^dThis study, MoK_α, sigmas are for isotropic atoms and neglecting unit cell errors.

^eThe interplanar spacing reported for this study are based on the best least-squares plane of just the inner TTT rings (C₁ C₂ C₃).



Tetrathiotetracene cation in high disorder TTT_2I_3 at 164 K.
Thermal ellipsoids are drawn at the 50% probability level

Figure 5



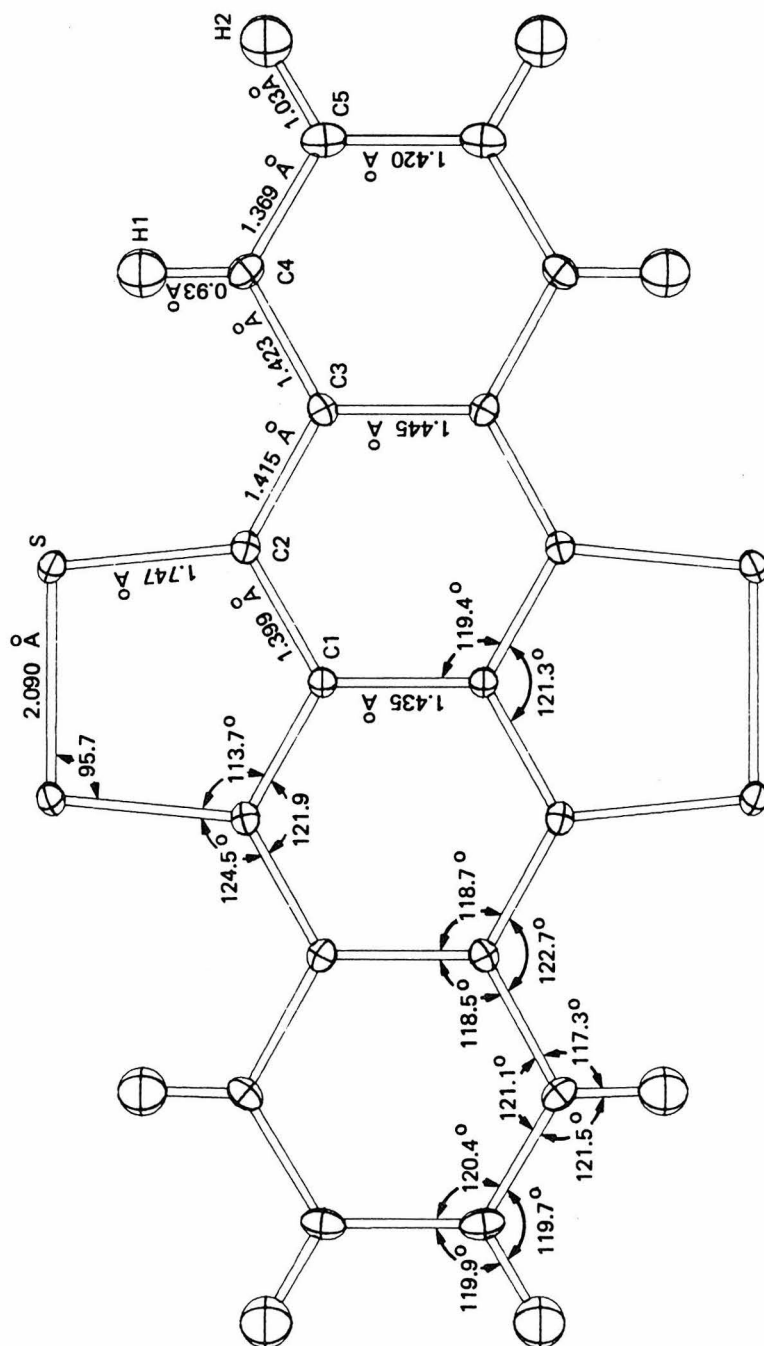
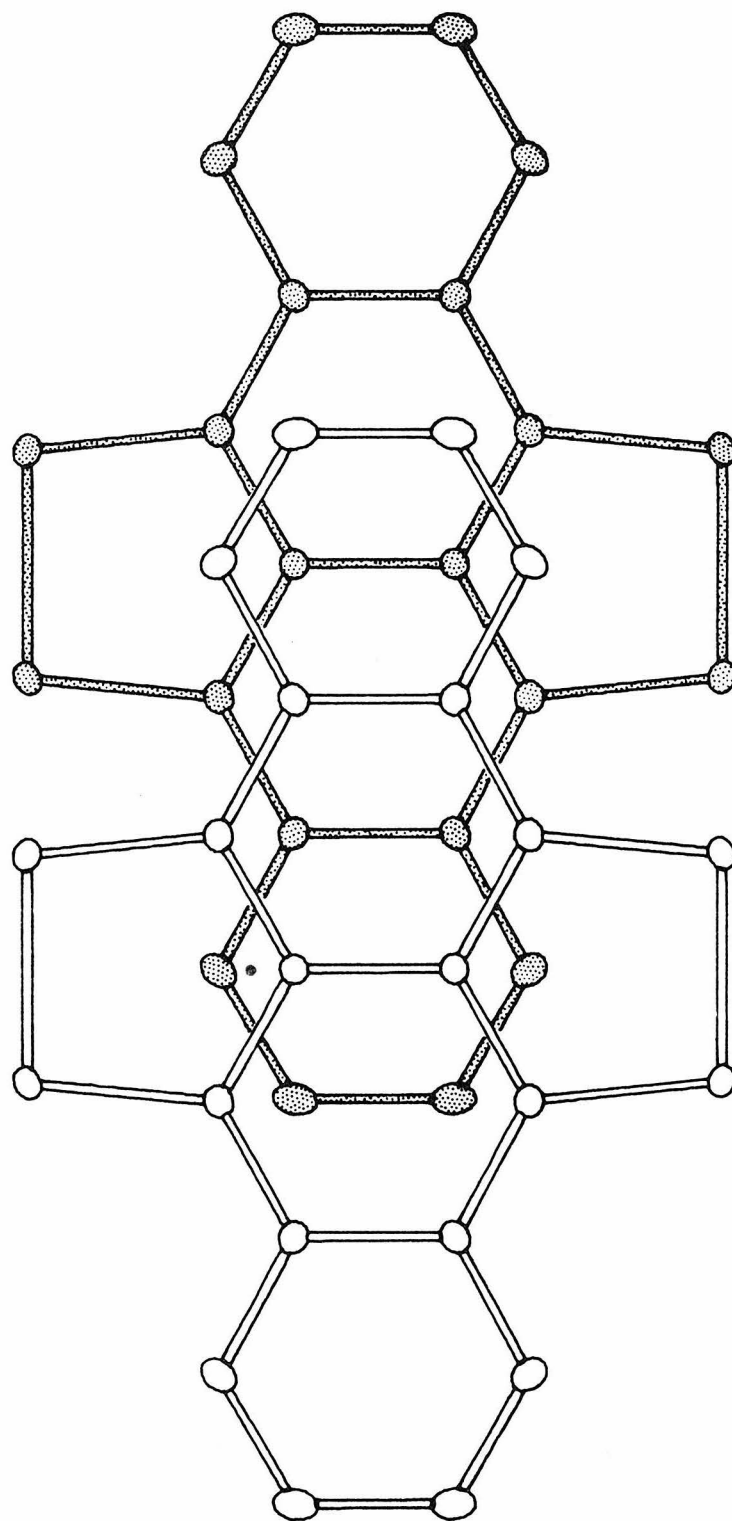


Figure 7

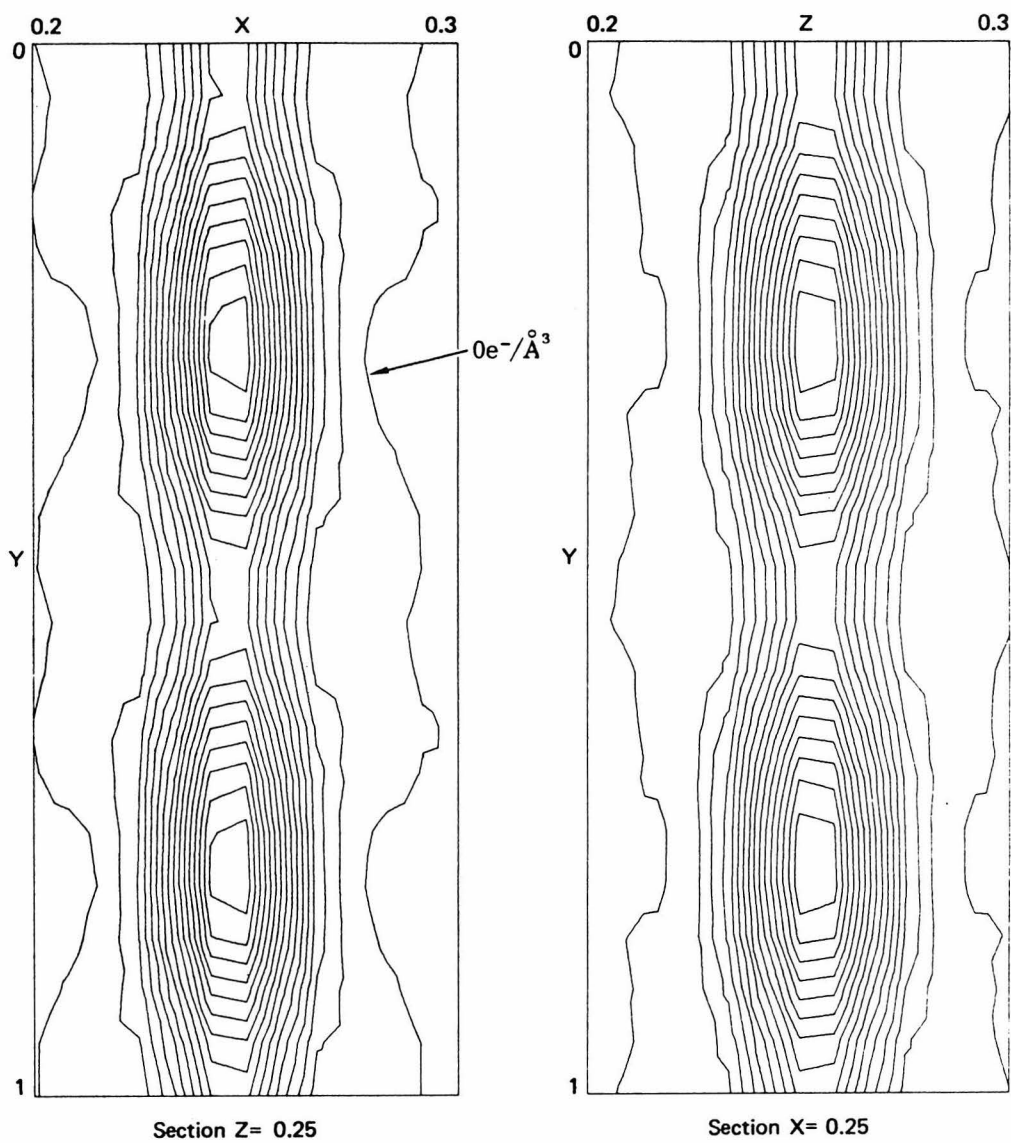
Tetrathiotetracene cation in high disorder $\text{TTT}_{2/3}$ at 190 K.
 Thermal ellipsoids are drawn at the 50% probability level



Neighboring tetrathiotetracene molecules in the same cation stack

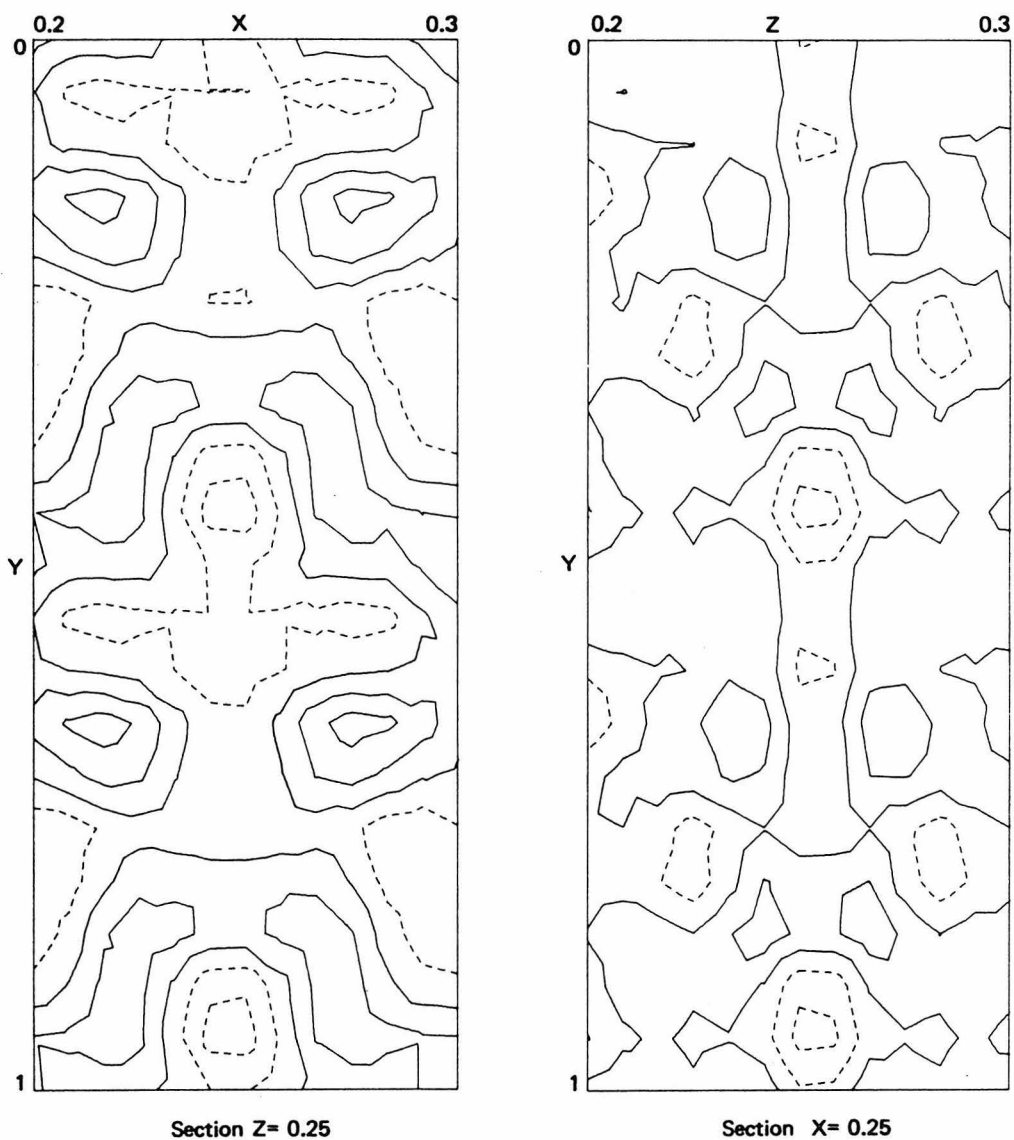
TTT_2I_3 (h. d.) $\phi \sim 19^\circ\text{K}$

Figure 8



TTT_2I_3 (h.d.) at 164°K
 Electron Density along Iodine Channel
 Contoured at $3.5 \text{ electrons}/\text{\AA}^3$ intervals beginning at zero

Figure 9



TTT_2I_3 (h.d.) at 164°K

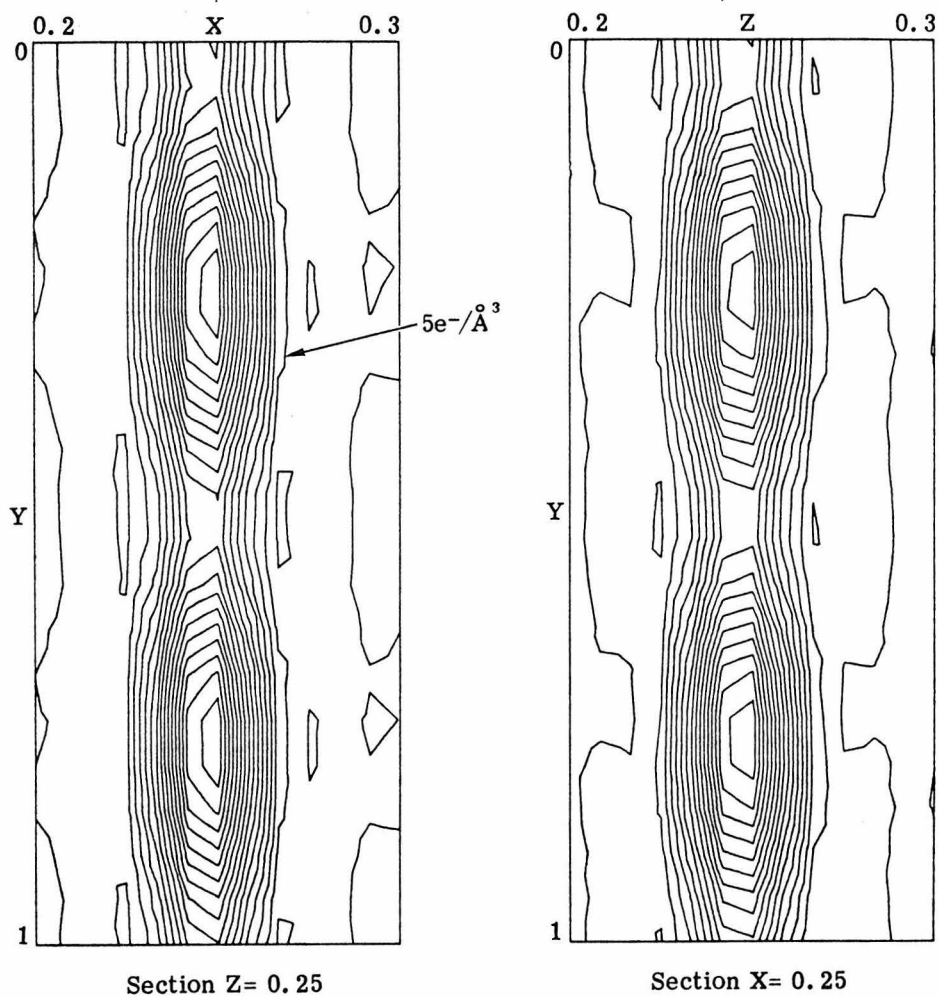
Difference Fourier Sections through Iodine channel

Contoured at 0.5 electrons / Å³ interval

----- negative contours

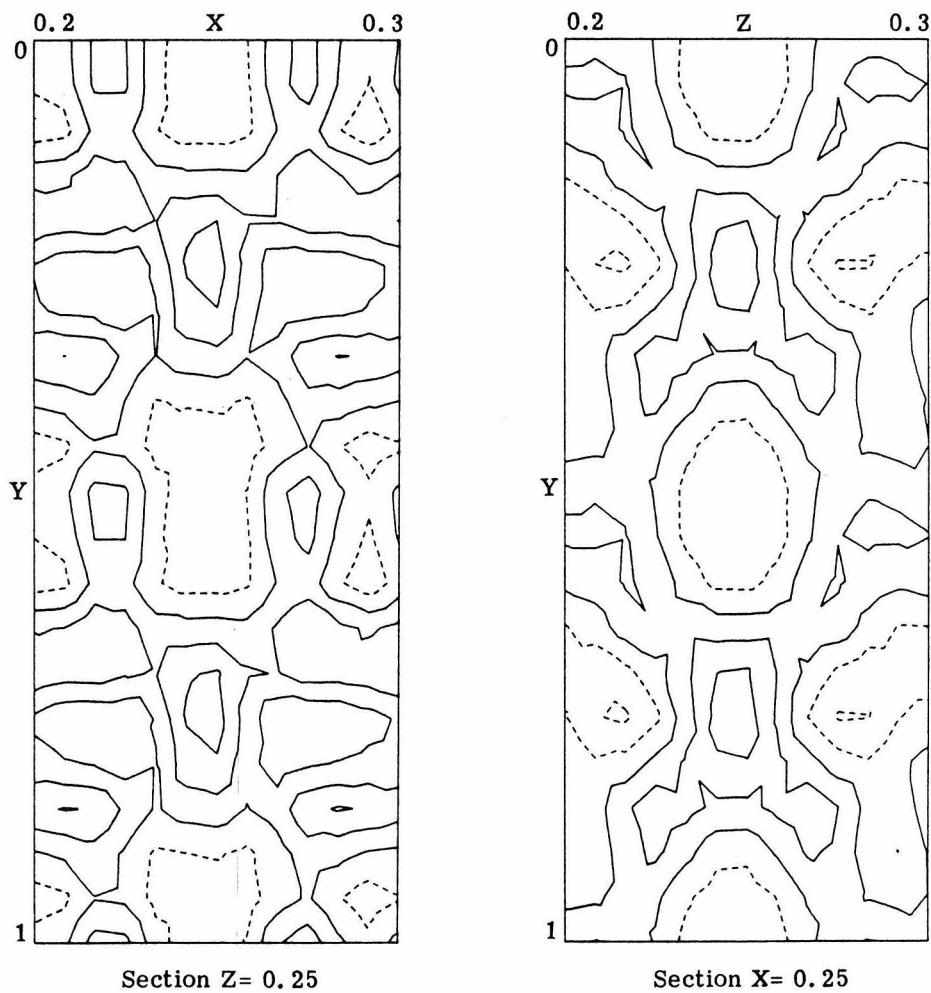
———— zero and positive contours

Figure 10



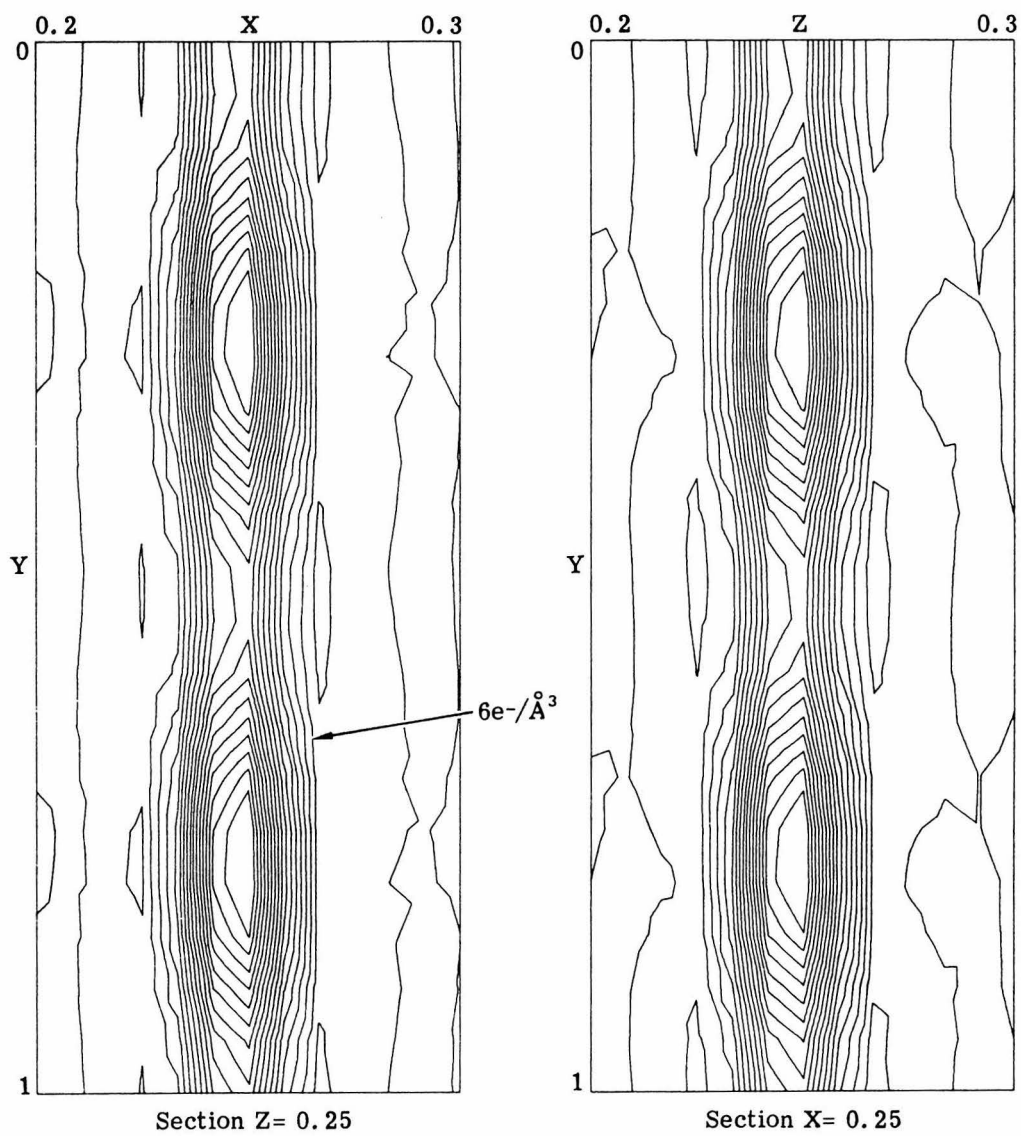
TTT₂I₃ (h.d.) at 74°K
 Electron Density along Iodine Channel
 Contoured at 5 electrons/Å³ intervals beginning at zero

Figure 11



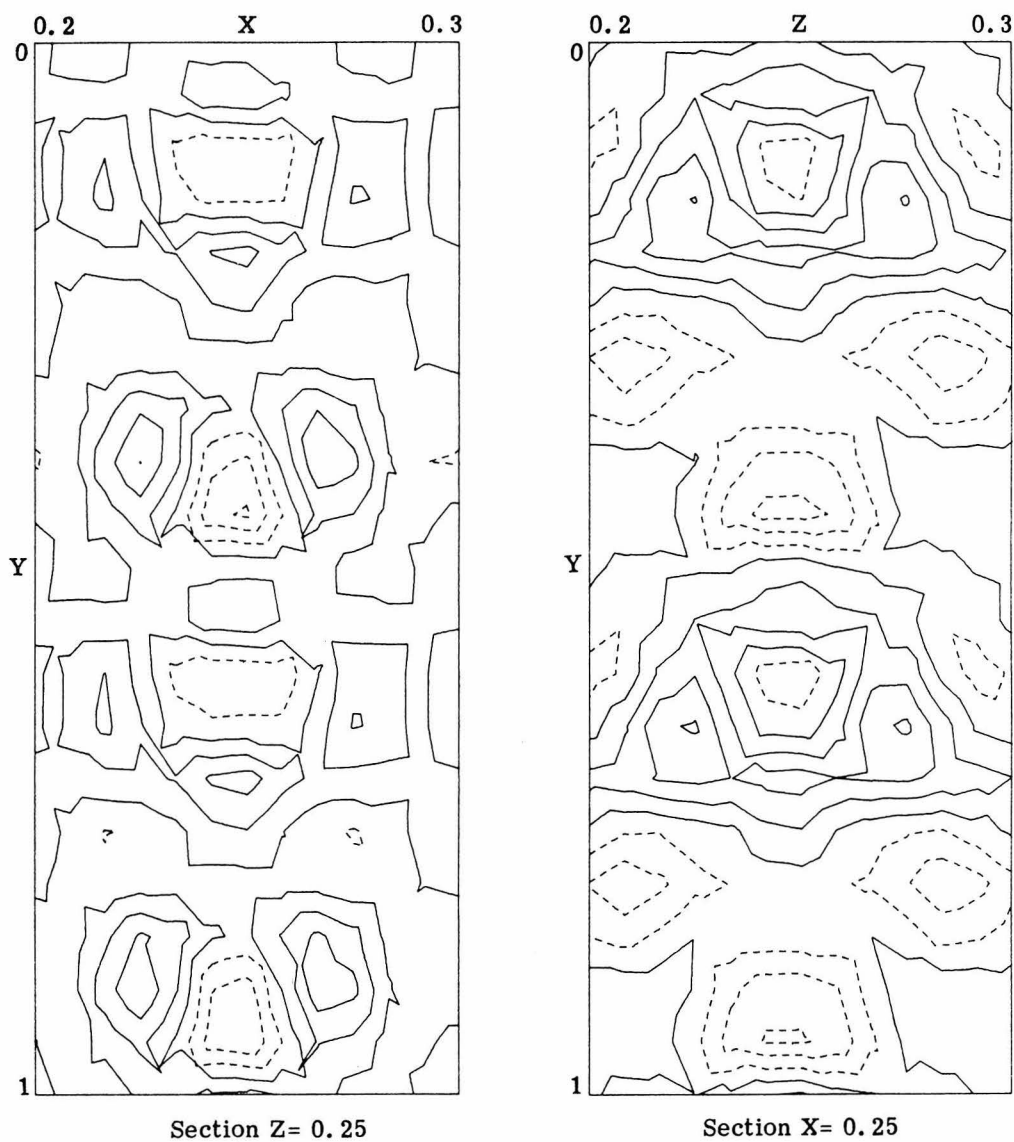
TTT_2I_3 (h.d.) at 74°K
 Difference Fourier Sections through Iodine Channel
 Contoured at 0.5 electrons/ \AA^3 intervals
 ----- negative contours
 ————— zero and positive contours

Figure 12



TTT₂I₃ (h. d.) at 19°K
 Electron Density along Iodine Channel
 Contoured at 6 electrons / \AA^3 intervals beginning at zero

Figure 13



TTT₂I₃ (h. d.) at 19°K
 Difference Fourier Sections through Iodine Channel

Contoured at 0.5 electrons/Å³ intervals

----- negative contours

———— zero and positive contours

Figure 14

Table XIIMaximum and Minimum Electron Density at $1/4, y, 1/4$ in TTT_2I_3 (h.d.).

	<u>$\sim 294^\circ \text{ K}$</u>	<u>164° K</u>	<u>74° K</u>	<u>19° K</u>	
Maximum	32.9	58.4	83.5	112.4	$\text{e}/\text{\AA}^3$
Position*	0.28	0.29	0.28	0.34	
Minimum	12.9	26.9	35.3	55.0	$\text{e}/\text{\AA}^3$
Position*	0.03	0.03	0.04	0.04	

*y, in fractional coordinates.

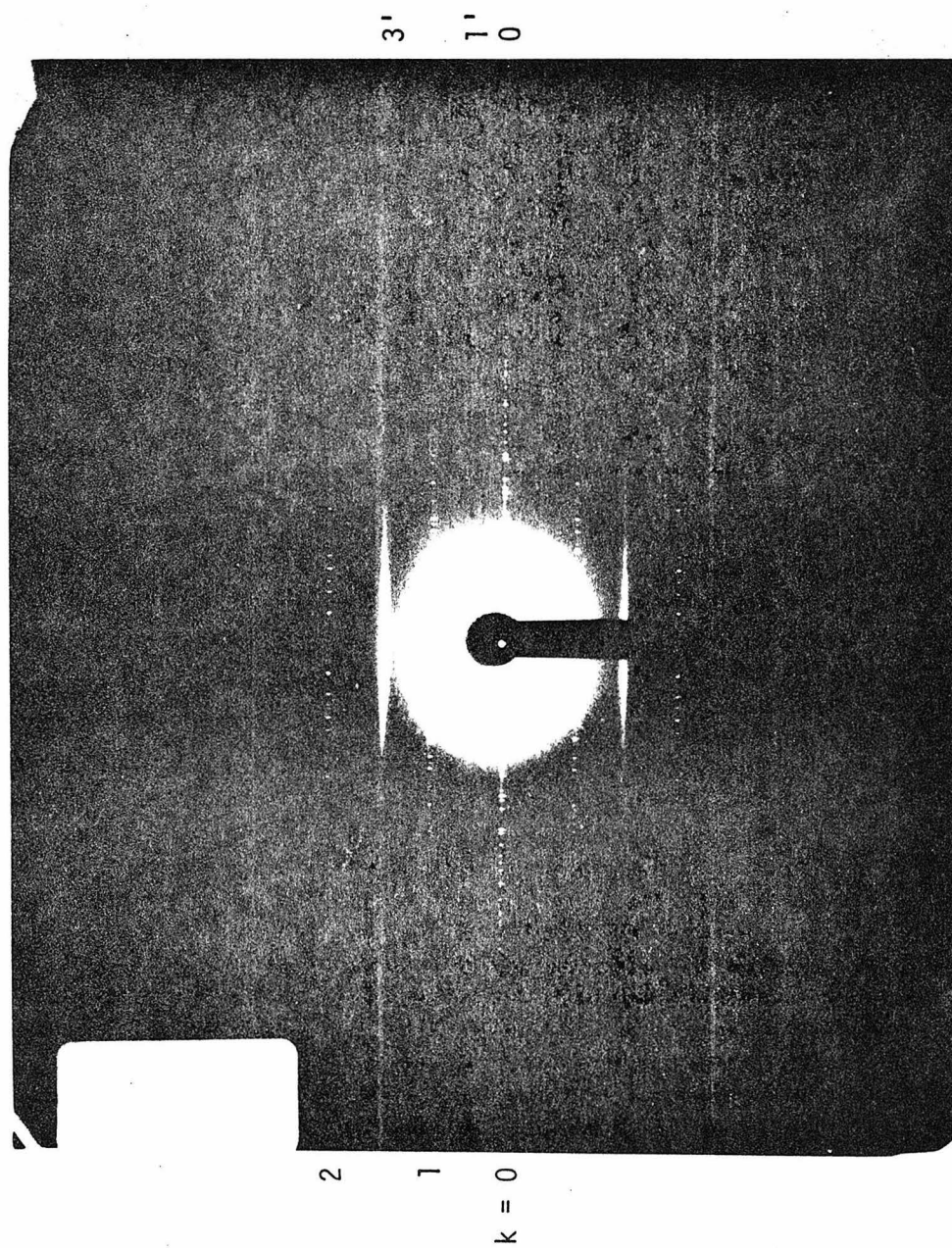
difference Fourier sections at 164° K, 74° K, and 19° K. The difference Fourier sections show that using two iodine atoms to represent the disordered iodide channel is not a perfect representation, especially at 19° K, but it is adequate. No additional information would be obtained by including more fractional iodine atoms to better represent the broad maximum. The root-mean-square (rms) displacements perpendicular to the channel at 19° K are $\sim 0.1 \text{ \AA}$. Small displacements off $1/4, y, 1/4$ are possible, but the rms displacements are too small to suggest that any large kinks occur in the iodide chain.

The similarity of the iodide channels at 19° K, 74° K, and 164° K suggests that the "TTT" lattice sees only average, disordered iodide channels which never order. The TTT lattice does not appear to distort at low temperatures.

Temperature Dependence of the Diffuse Layer Lines

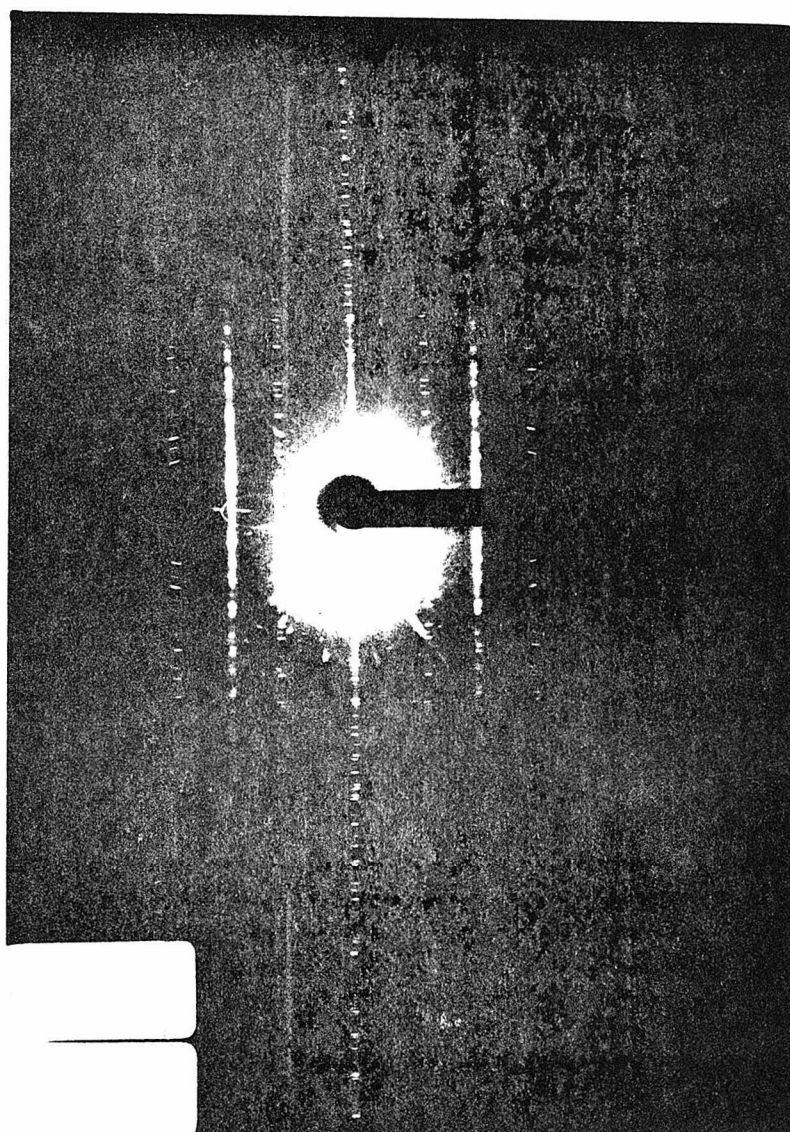
Although there is apparently no information in the TTT lattice (three-dimensional Bragg reflections) to suggest any phase transitions, the TTT lattice does not contain all of the structural information. The phase transitions could be related to changes that might only affect the diffuse layer lines. Figure 15 is a room-temperature X-ray oscillation photograph of a single crystal of TTT_2I_3 (h.d.) rotated about \vec{b} and taken with nickel-filtered copper K_α radiation. The layer lines for the TTT lattice ($b = 4.962 \text{ \AA}$) are labelled on the left side of the photograph. Because the diffuse layer lines are incommensurate

with the TTT lattice, they do not really represent a supercell. However, for convenience the diffuse layer lines have been labelled as shown on the right side of the photograph. At room temperature there is very slight beading along the intense, diffuse, $k = 3'$ layer line, but no beading on the other diffuse layer lines. The diffuse layer lines can be explained by the presence of one-dimensional chains of iodine atoms which are not correlated perpendicular to the chain direction. The beading on the $k = 3'$ layer indicates that there is some very weak, short-range order between the chains. Lowering the temperature of the crystal could cause any of three effects to occur: (i) the chains could three-dimensionally order amongst themselves into a lattice incommensurate with the TTT lattice, (ii) the chains could order with respect to the TTT lattice causing the formation, probably, of a supercell, or (iii) the chains could remain disordered and uncorrelated. Figure 16 is an oscillation photograph about \vec{b} of a TTT_2I_3 (h.d.) crystal at $120 \pm 10^\circ \text{ K}$. The crystal had been rapidly cooled in a stream of cold N_2 gas. There is a definite increase of beading on the $k = 3'$ layer, but no apparent beading on the other diffuse layers. A Weissenberg photograph of the $k = 3'$ layer at $\sim 120^\circ \text{ K}$ (Figure 17) shows that the increased beading is not due to additional ordering but to a loss of thermal motion. The size and shape of the diffraction 'spots' (blobs really) at 120° K are about the same as at room temperature indicating, still, only short-range chain-chain ordering. An X-ray photograph at $\sim 120^\circ \text{ K}$ obviously will not give information about a phase transition at $\sim 100^\circ \text{ K}$.



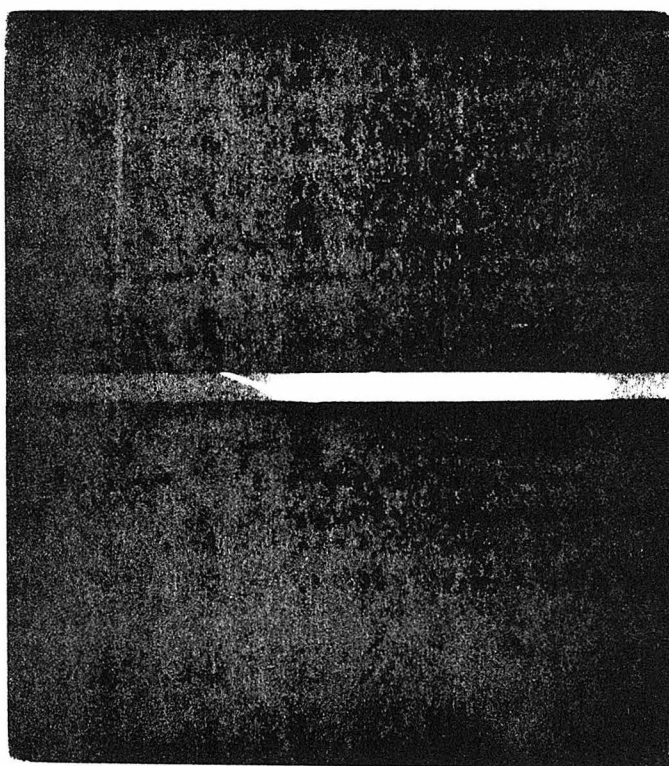
A Room Temperature X-Ray Oscillation Photograph of TTT_2I_3 (h.d.) ($\text{CuK}\alpha$)

Figure 15



An X-Ray Oscillation Photograph of TTT_2I_3 (h.d.) at $\sim 120 \pm 10^\circ \text{ K}$ ($\text{CuK}\alpha$)

Figure 16



A Weissenberg Photograph ($k = 3'$)
of TTT_2I_3 (h.d.) at $\sim 120 \pm 10^\circ \text{ K}$ (CuK_α)

Figure 17

During the slow-cooling of the fifth TTT_2I_3 (h.d.) crystal (see the previous Experimental Section) the $k = 3'$ layer was carefully monitored. In addition, a sixth crystal of TTT_2I_3 (h.d.) was put on the low-temperature diffractometer to monitor just the diffuse layer lines during slow-cooling. The sixth crystal was cooled in the following manner. From room temperature ($\sim 294^\circ \text{K}$) to 164°K the crystal was cooled between 2° per five minutes and 5° per forty-five minutes, with equilibration and peak scans every ten degrees. The phi-shaft was actually cooling at $0.1^\circ/10$ secs at the higher temperatures and $0.1^\circ/25$ secs at temperatures near 164°K . At 164°K omega scans, based on an orientation matrix for the 'spots' on the $k = 3'$ layer, were collected for $k = 1', 3', 5', 6', 7',$ and $9'$. From 164°K to 76°K the temperature was turned down by 2° every five minutes, although the actual cooling rate at the phi-shaft was $\sim 0.1^\circ/35$ secs. Every ten degrees the crystal was allowed to equilibrate and the peak scans were collected. At 76°K omega scans, based on an orientation matrix for 'spots' with $k = 3'$, were again collected for reflections with $k = 1', 3', 5', 6', 7'$ and $9'$. From 76°K to 24°K , the lowest temperature attained with the sixth crystal, the temperature was turned down at a rate of 0.2° per five minutes. The phi-shaft was actually cooling at $\sim 0.1^\circ/45$ secs. At 24°K omega scans of reflections with $k = 1', 3', 5', 6', 7',$ and $9'$ were collected with an orientation matrix for 'spots' with $k = 3'$.

There is no evidence from the slow-cooling experiments with the fifth and sixth TTT_2I_3 (h.d.) crystals studied at low temperatures that

the beading on the diffuse layer lines ever collapses to sharp reflections corresponding to long-range, three-dimensional ordering. The iodide chains remained disordered with only short-range chain-chain order from 294° to 19° K. Patterson maps calculated from intensities of only diffuse layer 'spots' at 164°, 76°, and 27° K confirm that the diffuse layer lines are caused solely by scattering from the iodide chains.

Shown in Figures 18, 19, and 20 are scans at various temperatures of reflections on the diffuse layer lines.* Although the 33'4 certainly increases in intensity with decreasing temperature (Figure 18), the full-width at half-height (FWHH) remains about the same. At 136° K the FWHH is about $1.38^\circ \omega$ ($\pm \sim .1^\circ \omega$), at 96° K $\text{FWHH} \approx 1.65^\circ \omega$, at 48° K $\text{FWHH} \approx 1.26^\circ \omega$, and at 20° K $\text{FWHH} \approx 1.38^\circ \omega$. Figure 19 shows the effect of temperature on a different reflection, the 13'0. At room temperature with molybdenum K_α radiation the 13'0 cannot really be resolved with diffractometer scans, although it can be seen on a 135 hour, room temperature Weissenberg photograph taken with copper K_α radiation. With decreasing temperature the 13'0 gains in intensity, but the peak shape continues to indicate only short-range coherent scattering. The intensity gain with slow cooling is completely reversible. The room temperature scan of the 13'0 in Figure 19 was actually run after the crystal had been cooled to 24° K and warmed

*The h and l indices in Figures 18, 19, 20, 21, and 22 and references to them in the text refer to an A-centered TTT lattice. For comparison with C-centered TTT indices h and l should be interchanged. In Figures 18 - 22 the index k is really k'.

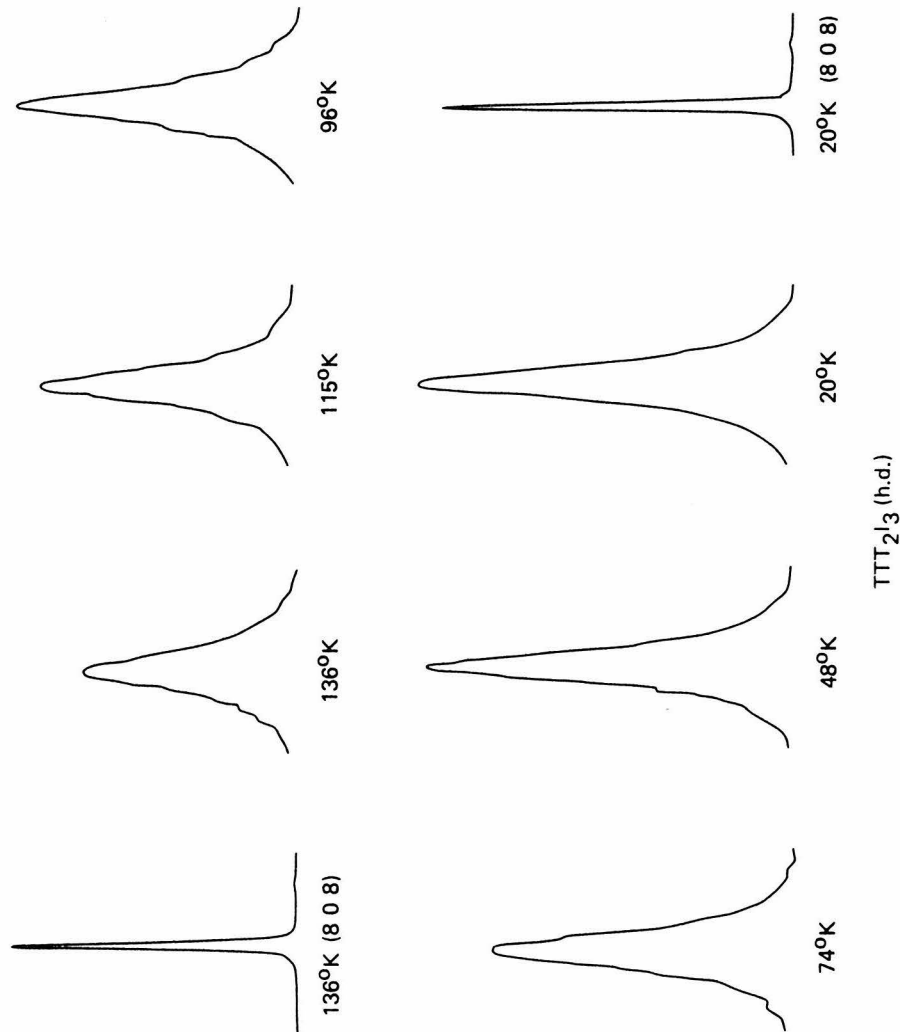
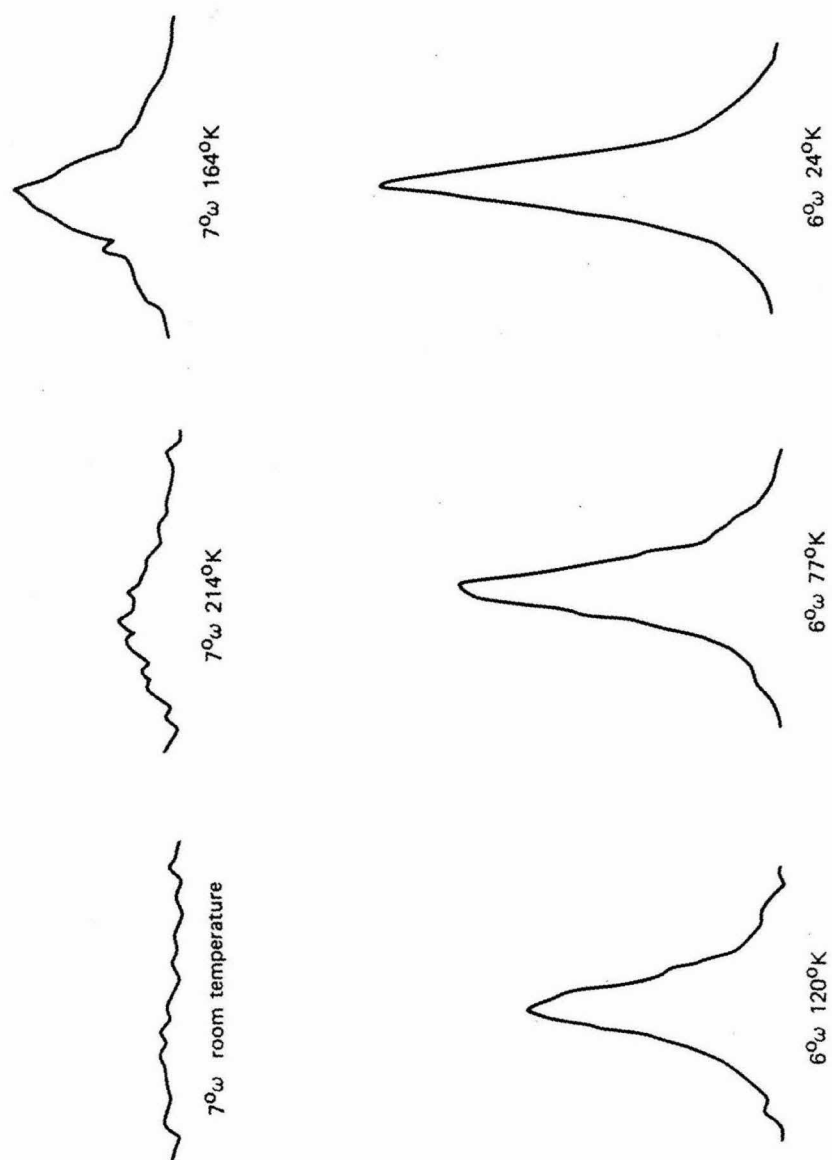


Figure 18

A Selected "Diffuse" Reflection, 3 3 4, as a Function of Temperature at the ϕ -shaft
(6.2° omega scans)

The intensity scale for 3 3 4 is expanded by ten times that of the 8 0 8



TTT_2I_3 (h.d.) omega scans of 1 3 0 at various ϕ shaft temperatures

Figure 19

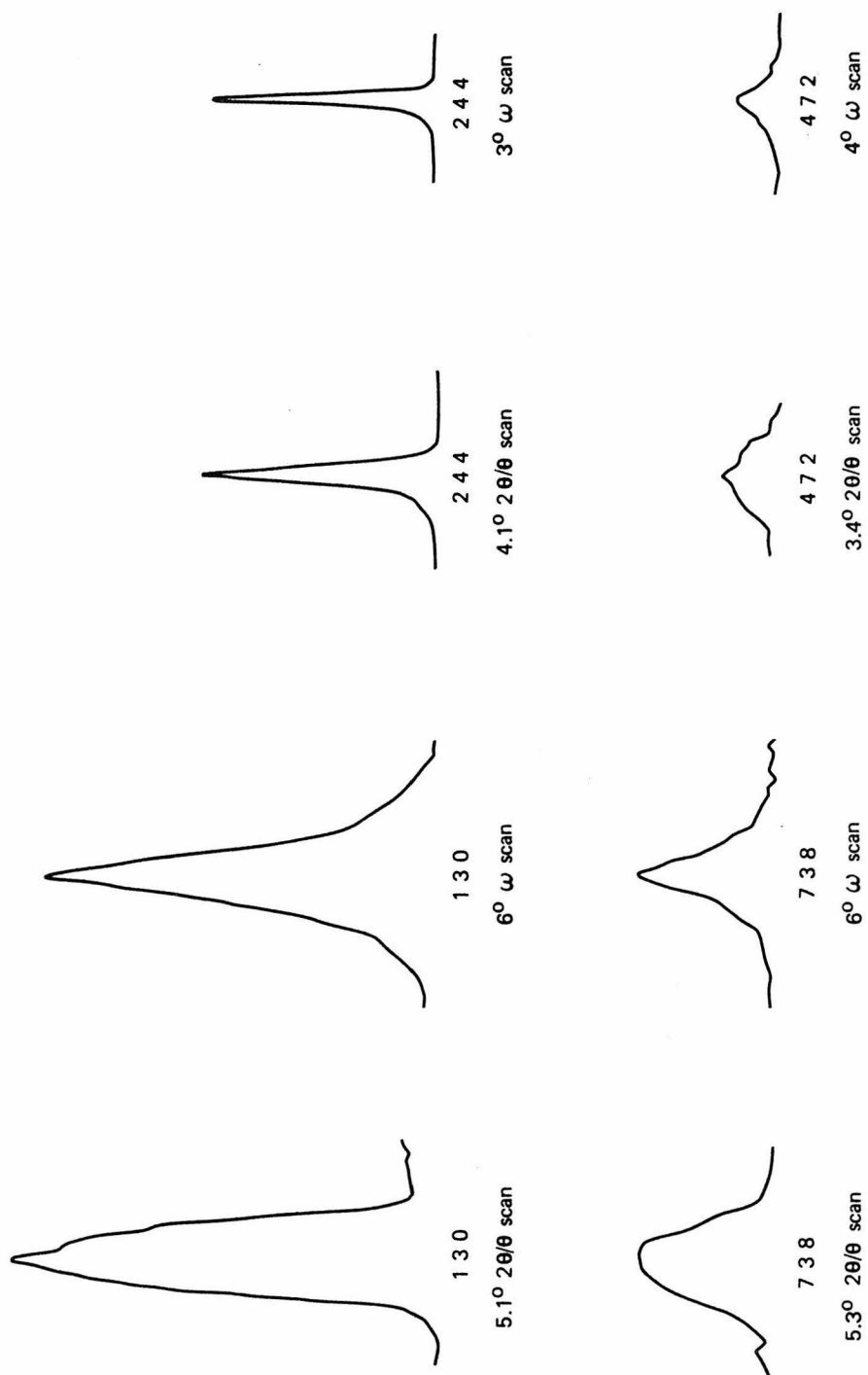


Figure 20

back to room temperature. Figure 20 shows scans of three reflections on the diffuse layer lines at 24° K and, for comparison, a TTT lattice reflection, the "2 4 4". The "2 4 4" is really 422 for the C-centered (Cmca) TTT lattice. The intensity scale for the diffuse reflections is expanded by five times that of the TTT lattice reflection. The full-widths at half-height for the omega scans are: 13'0, FWHH $\sim 1.22^\circ \omega$, 73'8, FWHH $\sim 1.22^\circ \omega$, 47'2, FWHH $\sim 0.9^\circ \omega$, "2 4 4", FWHH $\sim 0.24^\circ \pm .05^\circ \omega$, and for 808 (not shown in Figure 20), FWHH $\sim 0.24^\circ \pm 0.05^\circ \omega$. For 13'0, 73'8, and 47'2 the estimated error in measuring the FWHH is about $\pm 0.10^\circ \omega$. Figures 18-20 show that from 294° K to 19° K the iodide chains, even with very slow cooling, do not achieve long-range coherency.

Figure 21 is a plot of the integrated intensities from omega scans of several reflections on the diffuse layers and one h0l reflection as a function of temperature.* Figure 22 is a similar plot of the integrated intensities from omega scans of several TTT lattice reflections. Figures 21 and 22 are really a summary of the effect of slow-cooling on TTT_2I_3 (h.d.). Except for a possible anomaly at 85-90° K and a general leveling-off of intensities below about 40-50° K, with very slow cooling there is no evidence for a structural phase transition in TTT_2I_3 (h.d.) from $\sim 294^\circ$ K to 19° K. The TTT lattice remains orthorhombic with Cmca symmetry and does not form a supercell.

*All of the scans in Figures 21 and 22 were run with a large, 5 mm collector aperture. A normal 1 mm collimator was used on the monochromator for the primary X-ray beam.

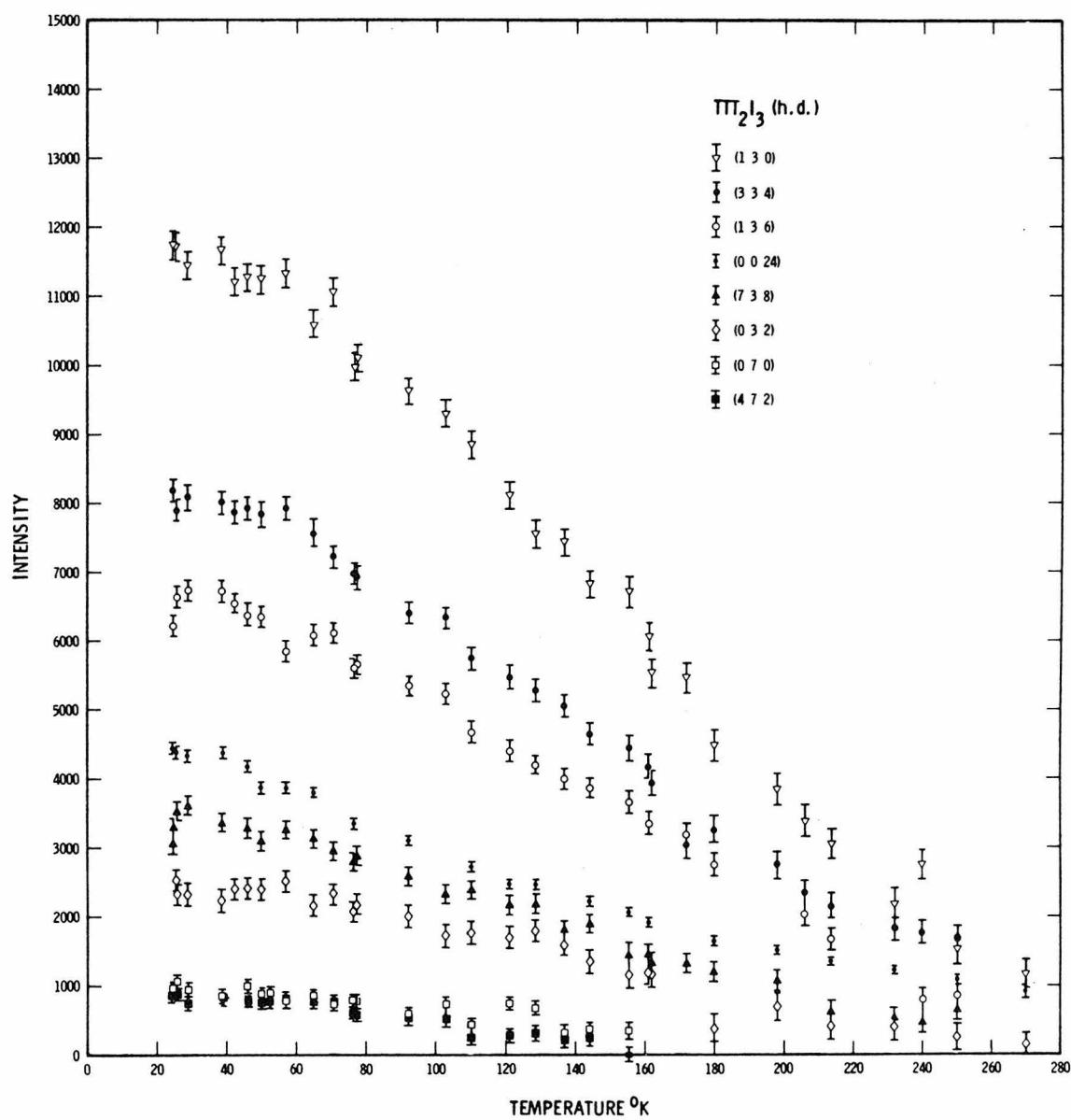


Figure 21

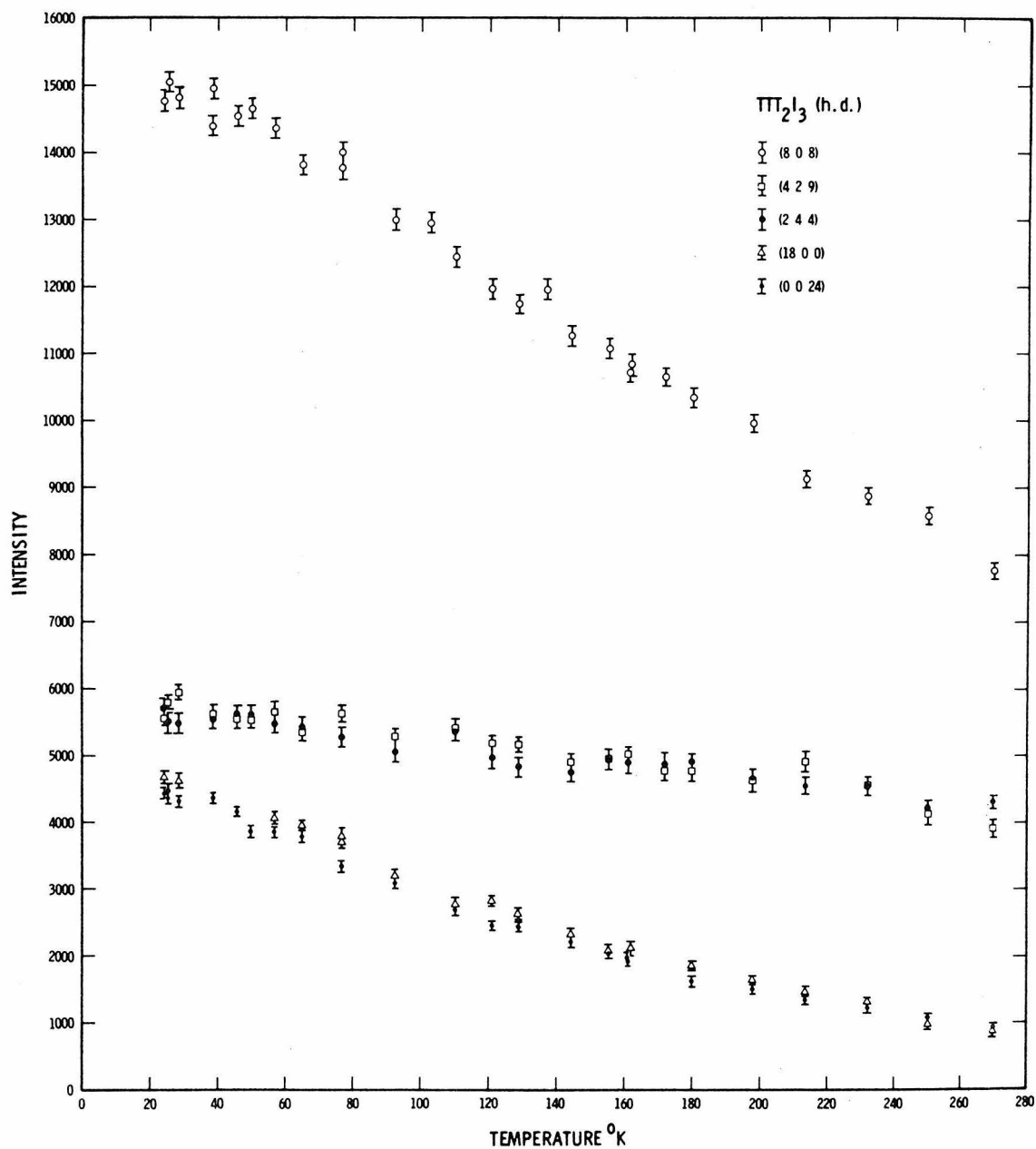


Figure 22

Reflections with $k = 3'$ and $k = 7'$ are incommensurate with the stacking axis of the TTT molecules (Table XIII). The diffuse layers with $k = 1'$, $5'$, $6'$, and $9'$ apparently remain diffuse. No reflections from these layers were observed. The intensities of the reflections on the two diffuse layer lines, $k = 3'$ and $k = 7'$, do increase. But, reflections on these layers remain very broad and weak even at 19° K indicating no long-range three-dimensional ordering.

Table XIII

TTT₂I₃ (h.d.)

Reflections ^b	Number of Reflections	Unit Cell Parameters ^a			Temperature ^c
		<u>a</u>	<u>b</u>	<u>c</u>	
hkl (Cmca)	22	18.259(6)	4.933(4)	18.341(8)	164° K
h3'l	5	18.266(9)	9.614(17)	18.342(13)	164° K
h0l	7				
hkl (Cmca)	32	18.208(6)	4.924(2)	18.265(8)	76° K
h3'l	8	18.212(28)	9.569(19)	18.252(43)	76° K
h3'l	8	18.215(7)	9.564(11)	18.268(8)	76° K
h0l	9				
hkl (Cmca)	36	18.192(6)	4.923(3)	18.235(8)	24° K
h3'l	10	18.158(31)	9.601(21)	18.230(52)	24° K
h3'l	10	18.193(6)	9.584(13)	18.238(8)	24° K
h0l	14				
h7'l	4	18.195(6)	9.732(5)	18.238(8)	24° K
h0l	14				

^aValues in this table are from the sixth crystal used for low temperature studies and should not be compared to values from the fifth crystal (Table I). The sixth crystal was not as well centered. Cell edge lengths are from a symmetry-constrained (orthorhombic) least-squares fit of 2 θ values.

^b h and l are the same as for the TTT lattice, Cmca.
 k' is the diffuse layer index (Figure 15).

^cphi-shaft temperature.

2.3 Discussion of the Diffuse Layer Lines

There are many causes for the appearance of diffuse scattering in the reciprocal space of seemingly ordered, three-dimensional crystals (22, 23). The causes can be broadly classified as from either thermal (dynamic) effects or structural (static) effects. The temperature-dependence studies of TTT_2I_3 (h.d.) suggest that the major cause of the diffuse layer lines is structural disorder. The anionic species in the channels of TTT_2I_3 are commonly assumed to be I_3^- , and that random shifts along \vec{b} of the I_3^- units in the channels will cause the diffuse layer lines observed on the X-ray diffraction photographs (9, 24, 25). This is a fairly reasonable explanation. However, the room-temperature structure of low disorder TTT_2I_3 (JPL crystal) suggests that the disorder be best accounted for by assuming, in addition to chain slippage, the presence of small amounts of I^- , I_2 , and possibly, other polyiodide species (26). Chemically this is quite reasonable. Iodine is known to form a variety of polyiodide species with strong inter-molecular interactions (27, 28, 29). A report by Beyeler in 1976 of the successful modeling of the X-ray diffraction pattern of the ionic conductor $\text{K}_{1.54}\text{Mg}_{0.77}\text{Ti}_{7.23}\text{O}_{16}$ (30) suggested that a similar approach might help to better understand the diffraction pattern for TTT_2I_3 (h.d.). In $\text{K}_{1.54}\text{Mg}_{0.77}\text{Ti}_{7.23}\text{O}_{16}$ the potassium cations are in channels. Beyeler modelled the potassium ions by assuming a random collection of cells of varying length and large displacements of the ions. He gave no details about how he calculated his final interference function.

The first model of the iodide chains in TTT_2I_3 (h.d.) assumes a collection of randomly occurring I_3^- , I_2 , I^- spacings. Guinier (31) has shown how to calculate the interference function for such a model. For three spacings the distribution of these spacings is a sum of three delta functions $h(x) = p_1\delta(x-d_1) + p_2\delta(x-d_2) + p_3\delta(x-d_3)$. The probability of each spacing, d_j , occurring is p_j , and $p_3 = 1-p_1-p_2$. The crystal in real space is built from the convolution of $h(x)$ with itself an infinite number of times and this function is $z(x)$. The Fourier transform of $z(x)$ gives the interference pattern of the crystal in one-dimensional reciprocal space. The transform of $h(x)$ is

$$H(s) = \int h(x)\exp(2\pi isx)dx = \sum_j p_j \exp(2\pi isd_j) . \quad \text{Equation 1}$$

Therefore,

$$\begin{aligned} \text{transf } z(x) = Z(s) &= 1 + H(s) + H^2(s) + H^3(s) + \dots \\ &+ H^*(s) + [H^*(s)]^2 + [H^*(s)]^3 + \dots \end{aligned}$$

If $h(x)$ had been represented by a single delta function, ie. a lattice with a single repeat spacing a , then $Z(s)$ would be the interference function (or diffraction pattern) for a one-dimensional grating and $Z(s) = 1 + 2 \sum_{n=1}^{\infty} \cos(2\pi na)$. Since $\sum_j p_j = 1$, $|H(s)|$ (equation 1) is always less than one when $N > 1$ (except at $s = 0$) and so $Z(s)$ is a geometric progression which can be expressed as

$$Z(s) = 2 \operatorname{Re} \left[\frac{1}{1-H(s)} \right] - 1 .$$

This is the same result that Wright obtained (32). Hendricks and Teller (33) and Méring (34) also calculated this same result but expressed $Z(s)$ in the form

$$Z(s) = \frac{1-R^2}{1+R^2-2R\cos\phi} \quad \text{Equation 2}$$

where $\operatorname{Re}^{i\phi} = \sum_j p_j \exp(2\pi i s d_j) \equiv H(s)$ (35). On the basis of known bond lengths and van der Waal's radii (29,36), an I_3^- spacing would be $\sim 10.1 \text{ \AA}$, an I_2^- spacing $\sim 7 \text{ \AA}$, and an I^- spacing $\sim 4.3 \text{ \AA}$ for non-interacting species. Shown in Figure 23, as a function of $|\vec{s}|$ (35), are a series of computer-generated plots of Equation 2 in which the probabilities and spacings have been varied. The parameters for each plot are listed in Table XIV. These plots demonstrate that the positions and the intensities of the diffraction lines are sensitive to changes in the probabilities and spacings. Figure 24 is a 135 hour X-ray oscillation photograph of TTT_2I_3 (h.d.) rotated about \vec{B} . The photograph was taken in a cylindrical camera with monochromatized molybdenum K_α radiation. The film was wrapped in 0.005 inch aluminum and after development the back emulsion of the Kodak No-Screen X-ray film was removed. This oscillation photograph exhibits a weak $k = 1'$ diffuse layer line, a weak $k = 2'$ diffuse layer line next to but incommensurate with the TTT lattice line, a very strong $k = 3'$ line with possibly two components, a weak $k = 4'$ line again next to, but incommensurate with the TTT

Table XIV

Parameters for the Plots Shown in Figure 23

	I_3^- probability, spacing (Å)	I_2 probability, spacing (Å)	I_1^- probability, spacing (Å)
<u>A</u>	0.80 , 10.1	0.10 , 7.0	0.10 , 4.3
<u>B</u>	0.50 , 10.1	0.30 , 7.0	0.20 , 4.3
<u>C</u>	0.70 , 9.8	0.25 , 6.5	0.05 , 4.3
<u>D</u>	0.70 , 10.0	0.25 , 6.7	0.05 , 4.2
<u>E</u>	0.70 , 10.1	0.25 , 7.0	0.05 , 4.3

The function plotted is $Z(s) = (1-R^2)/(1 + R^2 - 2R\cos\phi)$ for $\text{Rexp}(i\phi) = \sum_{j=1}^3 p_j \exp(2\pi i s d_j)$.

p_j is the probability of spacing d_j occurring and $\sum_j p_j = 1$.

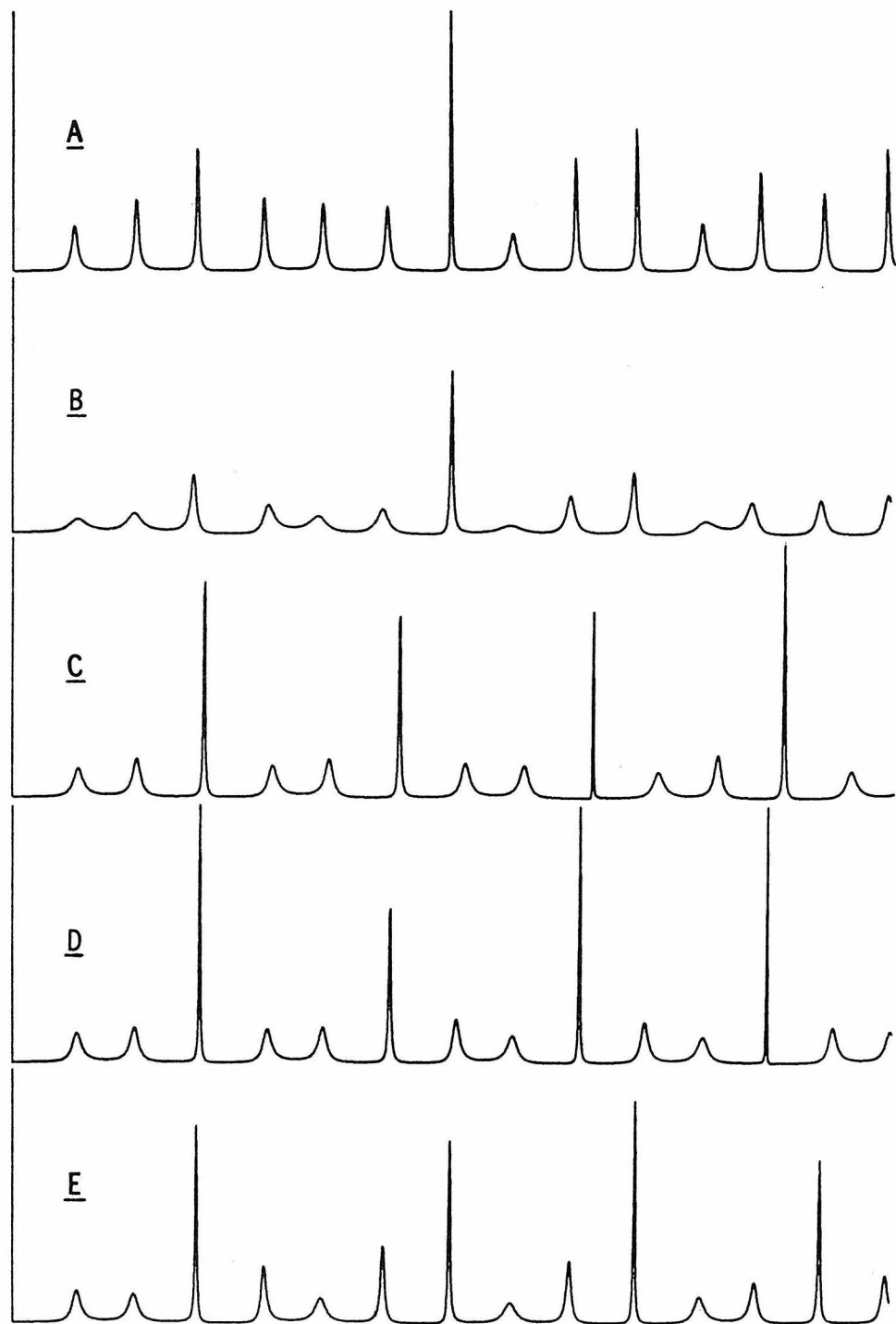
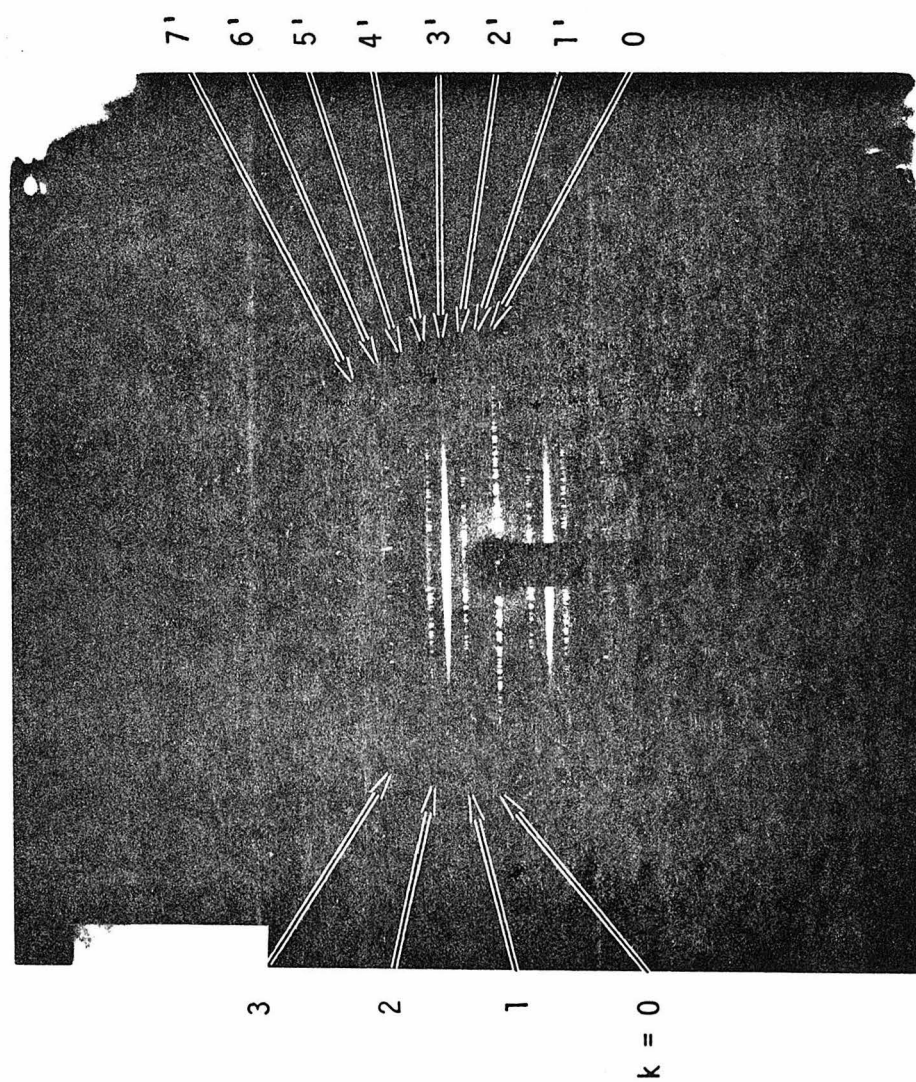


Figure 23

Interference function, $Z(s)$ (Equation 2).



A. Room Temperature X-Ray Oscillation Photograph of TTT_2I_3 (h.d.)
(135 hrs. monochromatized MoK_α)

Figure 24

lattice line, a $k = 5'$ diffuse layer line, a moderately strong $k = 6'$ line clearly displaced from the TTT line, a moderately strong $k = 7'$ line, and two or three broad, very weak diffuse lines at higher angles. Although none of the plots in Figure 23 are an exact replication of the oscillation photograph, the similarities are encouraging.

Further improvements in the model ought to include the effect of slight atomic displacements (ie. "thermal" parameters or "temperature factors"), the effect of different spacings containing different amounts of scattering matter (ie. structure factors), and the effect, perhaps, of slight variations in the spacings and nearest-neighbor correlations. A model with different spacings for different iodide species is very similar to stratification in clays. Several people have worked out the expected diffraction patterns for clay with interstitial material. One of the more recent and fairly complete treatments of this problem has been published by Wright (32). His intensity expression is used here to model the diffraction pattern of the iodide chains in TTT_2I_3 (h.d.). This is a one-dimensional model and represents, essentially, the diffraction pattern expected for one infinitely long chain.

Wright has assumed in the derivation of his model that there is a basic layer with some spacing which represents the majority of the structure. But, the basic layers do not always come in perfect succession; occasionally there are layers of other material, with different spacings, in between the basic layers. Although this model allows for the

possibility of an I_3^- following an I_3^- in the chain, I_2 or I^- species would always be bordered on either side by I_3^- . For this model of an infinite chain the intensity would be given by the expression (32)

$$|A(s)|^2 = 2[\beta\beta^*/(1-\alpha)]_{\text{real}} - |F_b|^2 + p_1|F_1|^2 + p_2|F_2|^2 \quad \text{Equation 3}$$

where

$$\beta = F_b + p_1 F_1 e^{i\phi_1/2} + p_2 F_2 e^{i\phi_2/2}$$

$$\beta^* = F_b^* + p_1 F_1^* e^{i\phi_1/2} + p_2 F_2^* e^{i\phi_2/2}$$

$$\text{and } \alpha = \sum_k p_k e^{i\phi_k}.$$

p_k are probabilities and $\sum_k p_k = 1$. F_b is the structure factor for the basic layer, in this case I_3^- . F_1 and F_2 are structure factors for I_2 and I^- . When an I_3^- follows an I_3^- the structure factor for the nonexistent interstitial material is zero, but there is a term in α for the probability of an I_3^- following an I_3^- . ϕ_k is the phase shift due to the spacing of an I_3^- plus the spacing for the k species. Included in the structure factors are terms of the form $\exp(-B \sin^2 \theta / \lambda^2)$ to account for thermal motion of the atoms (37). Preliminary calculations of Equation 3 showed that I_3^- following I_3^- is a prominent feature of the iodine chains. Shown in Figure 25a are computer-generated plots of Equation 3.

The parameters for each plot are listed in Table XVa. The effect of a slightly asymmetric I_3^- is negligible. The probabilities of each species primarily determine the intensities of the various lines, although the spacings do affect the intensities. The plots show basically the same diffraction pattern as in Figure 24. The plots indicate that 70-80% of the I_3^- will have I_3^- as nearest neighbors, but there is a small percentage of I_2 and I^- present.

The oscillation photographs have broad diffraction lines, especially at high angles, which the plots in Figure 25a do not have. Two effects could cause the observed broadening. The mosaic blocks in the crystal could be very small or the spacings of the iodide species could be variable. Because the TTT lattice in TTT_2I_3 (h.d.) gives rise to very normal, sharp reflections, there is no reason to assume unusually small mosaic blocks. However, the spacings of the iodide species could, realistically, be quite variable. I_3^- , I_2 , and I^- are known to form "polyiodide" species with a wide range of inter- and intramolecular I-I distances (27,28,29,38,39). For the case when the variations in the spacings are much smaller than the average spacings and the variations have a normal distribution about the average spacings, Wright has derived the following modifications to Equation 3. For spacings d_k (d_k is the basic plus interlayer spacing) with a deviation of Δ_k ($\Delta_k \ll d_k$), then

$$\alpha = \sum_k p_k \exp(2\pi i s d_k) \exp(-2\pi^2 s^2 \Delta_k^2)$$

Table XVa

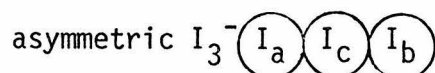
Parameters for the Plots of the Wright Model

Figure 25a - No deviations in the spacings

Constants in all five plots

$$D_{II}(I_2) = 2.70 \text{ \AA}$$

$$\text{spacings: } I_3^- + I_2 = 16.7 \text{ \AA} \quad I_3^- + I^- = 14.3 \text{ \AA} \quad I_3^- + 0 = 9.9 \text{ \AA} *$$

A

$$I_a - I_c = 2.83 \text{ \AA}$$

$$I_b - I_c = 3.04 \text{ \AA}$$

$$B(I_a) = 8.0$$

$$B(I_b) = 10.0$$

$$B(I_c) = 2.0$$

$$B(I_2) = 4.0$$

$$B(I^-) = 3.0$$

$$\text{probabilities: } I_3^- + I_2, 0.15 \quad I_3^- + I^-, 0.05 \quad I_3^- + 0, 0.80$$

B

$$\text{symmetric } I_3^- \quad I_a - I_c = I_b - I_c = 2.92 \text{ \AA}$$

$$B(I_a) = B(I_b) = 8.0$$

$$B(I_2) = 4.0$$

$$B(I^-) = 3.0$$

probabilities: same as in ACasymmetric I_3^- (same bond lengths as in A)

$$B(I_a) = 3.0$$

$$B(I_b) = 4.0$$

$$B(I_c) = 2.0$$

$$B(I_2) = 4.0$$

$$B(I^-) = 6.0$$

$$\text{probabilities: } I_3^- + I_2, 0.15 \quad I_3^- + I^-, 0.15 \quad I_3^- + 0, 0.70$$

Dsymmetric I_3^- (same bond lengths as in B)

$$B(I_a) = B(I_b) = 3.0$$

$$B(I_c) = 2.0$$

$$B(I_2) = 4.0$$


$$B(I^-) = 6.0$$

probabilities: same as in C* I_3^- are nearest neighbors

Table XVb

Parameters for the Plots of the Wright Model (32)

Figure 25b - With deviations in the spacings

Constants in all four plots; asymmetric I_3^-  $I_a - I_c = 2.87 \text{ \AA}$
 $I_b - I_c = 2.97 \text{ \AA}$
 $B(I_a) = B(I_b) = 2.0$ $B(I_c) = 1.0$ $B(I_2) = 3.0$ $B(I^-) = 3.0$ $D_{II}(I_2) = 2.70 \text{ \AA}$

E

Spacings:	$I_3^- + I_2 = 16.7 \text{ \AA}$	$I_3^- + I^- = 14.0$	$I_3^- + 0 = 9.6^*$
Deviations:	0.5 \AA	0.3	0.3
Probabilities:	0.18	0.04	0.78

F

Spacings:	$I_3^- + I_2 = 16.7 \text{ \AA}$	$I_3^- + I^- = 14.3$	$I_3^- + 0 = 10.0^*$
Deviations:	0.5 \AA	0.3	0.3
Probabilities:	0.18	0.04	0.78

G

Spacings:	$I_3^- + I_2 = 17.0 \text{ \AA}$	$I_3^- + I^- = 14.3$	$I_3^- + 0 = 9.6^*$
Deviations:	0.5 \AA	0.3	0.3
Probabilities:	0.18	0.04	0.78

H

Spacings:	$I_3^- + I_2 = 16.7 \text{ \AA}$	$I_3^- + I^-$	$I_3^- + 0 = 9.7^*$
Deviations:	0.5 \AA		0.3
Probabilities:	0.20	0.0	0.80

* I_3^- are nearest neighbors.

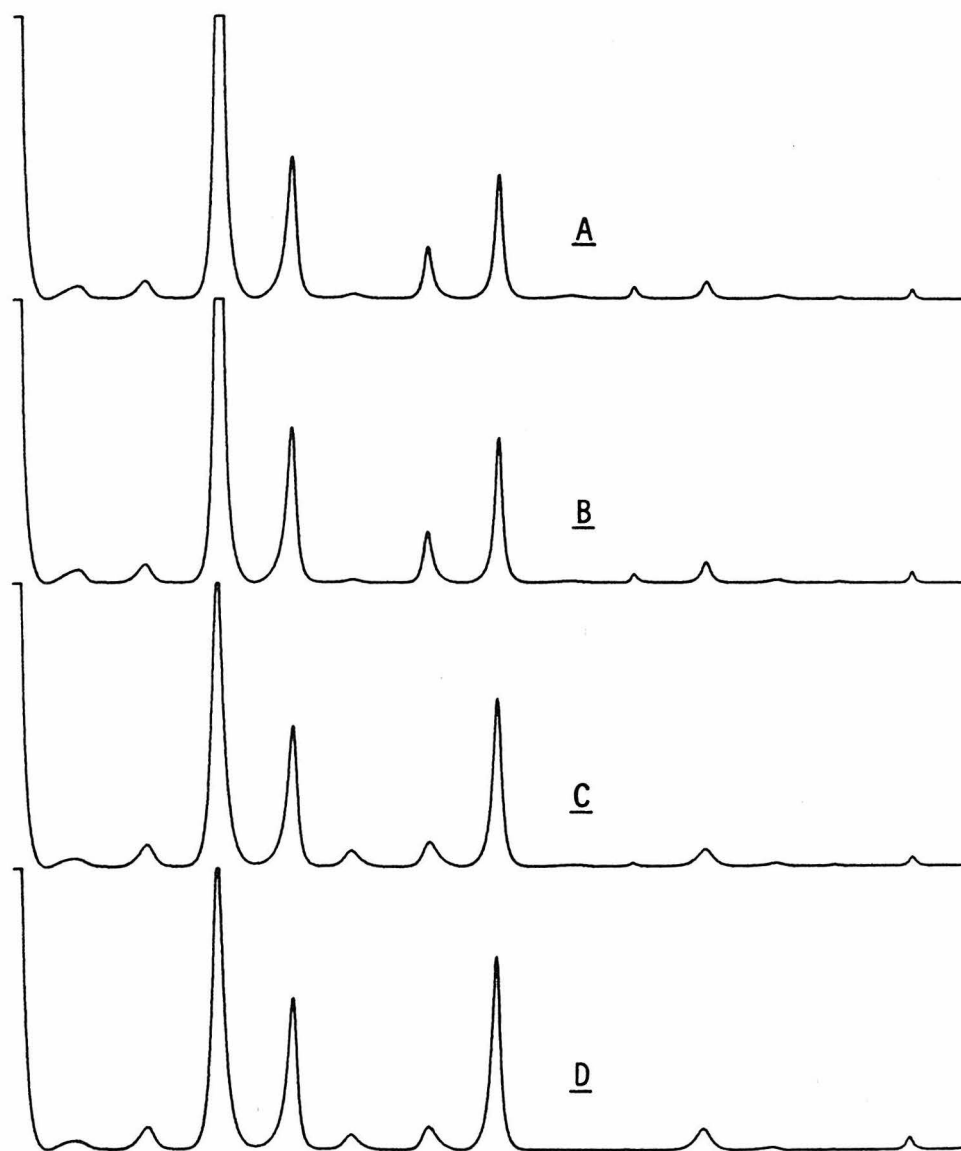


Figure 25a

Plot of the Wright Model (see Table XVa)

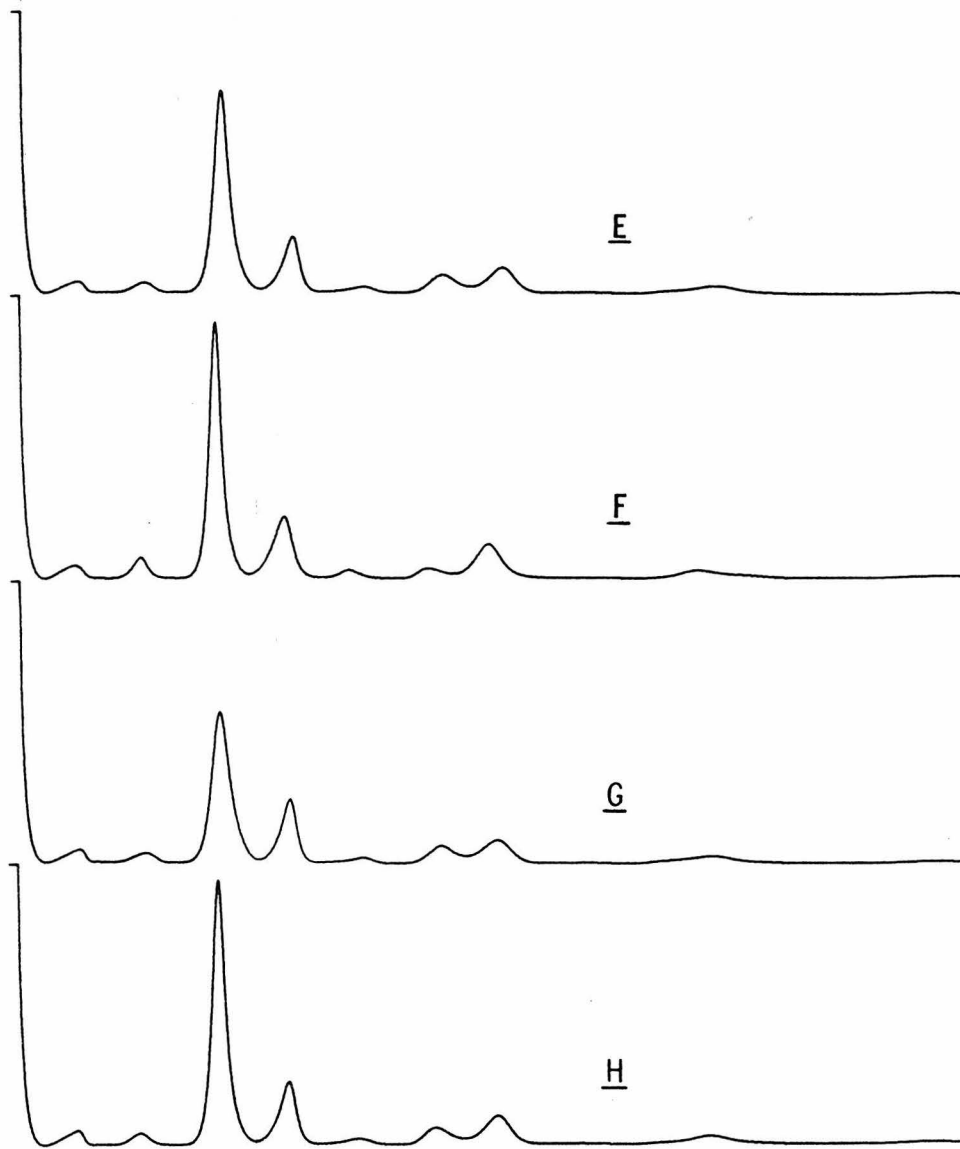


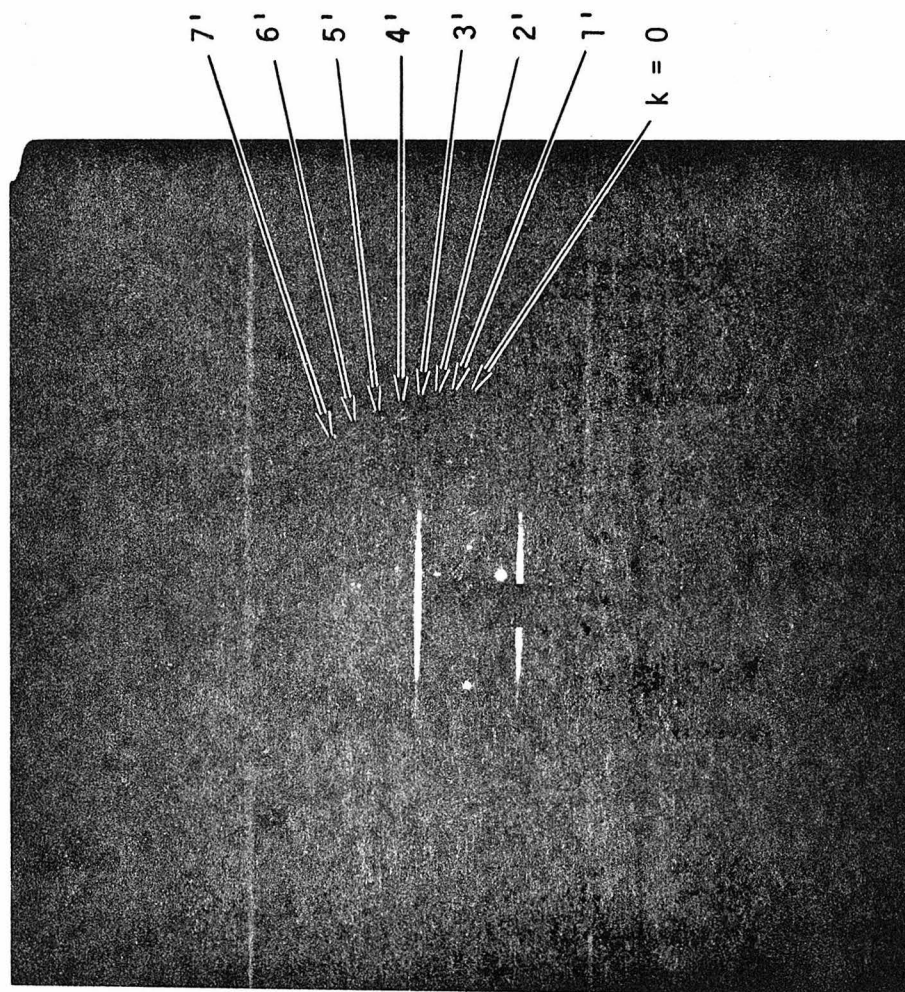
Figure 25b

Plot of Wright Model (see Table XVb)

$$\beta \approx F_b + \sum_k p_k F_k \exp(i\pi s d_k) \exp(-\pi^2 s^2 \Delta_k^2 / 2)$$

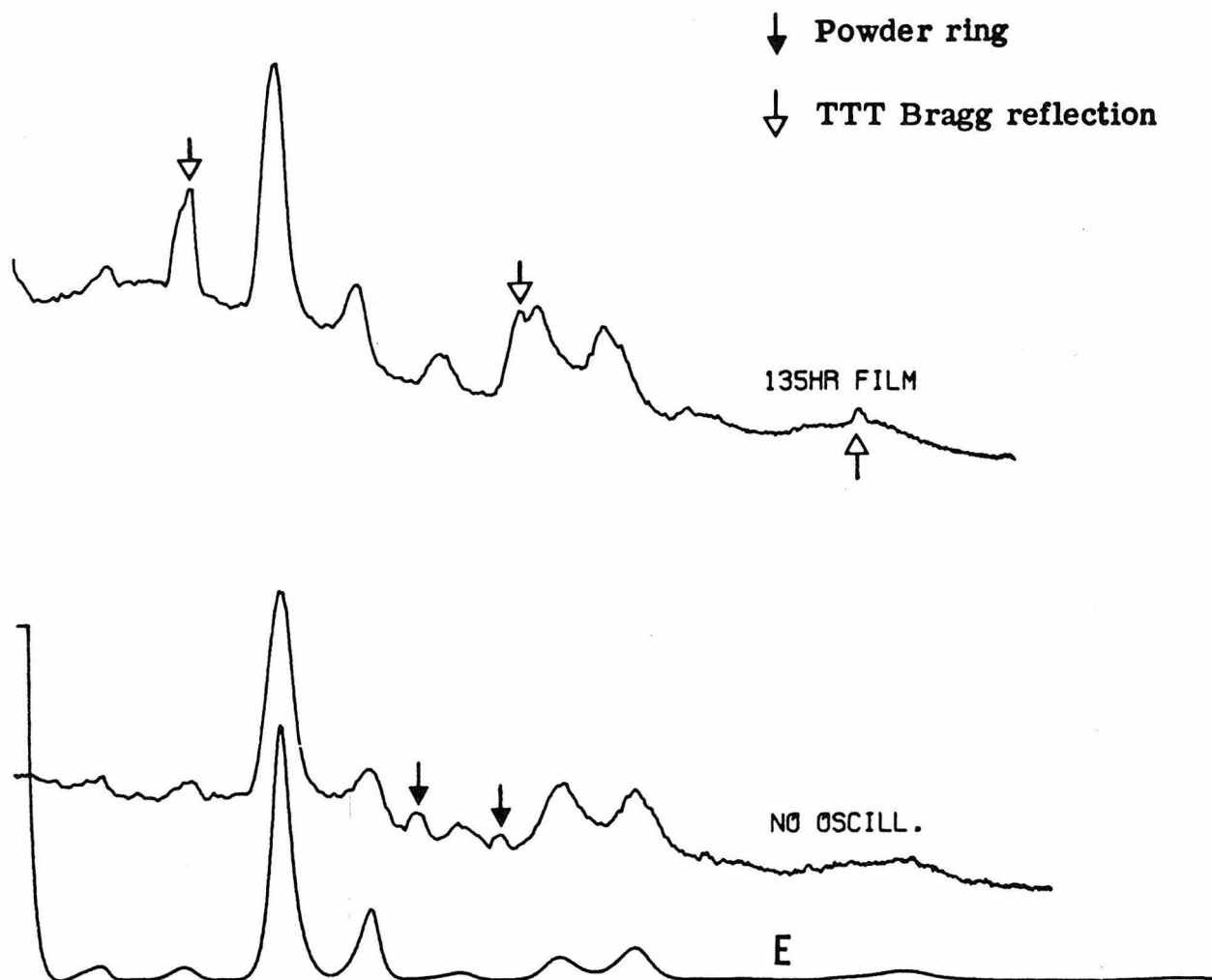
$$\text{and } \phi_k \equiv 2\pi s d_k \left(|\vec{s}| \equiv s = \left(\frac{2 \sin \theta}{\lambda} \right) \right).$$

This treatment assumes that the deviations, Δ_k , will not cause a change in the structure factors. For the iodide chain this is reasonable, because the Δ_k are being used to represent the deviations in inter-molecular spacings caused by variable amounts of interaction between iodide species. Shown in Figure 25b are computer-generated plots of Equation 3 but small deviations for the spacings have been included. Table XVb lists the parameters for the plots shown in Figure 25b. Not only do small deviations in the spacings cause broadening of the peaks at high angles, but the deviations seem to act as a "smoothing" function. With small spacing deviations Equation 3 is fairly insensitive to small changes in the other parameters. (Compare the changes in the parameters listed in Table XVb with the plots of Figure 25b,) Figure 26 is an X-ray photograph of TTT_2I_3 (h.d.) taken as the photograph in Figure 24 was with monochromatized molybdenum K_α radiation. The Kodak No-Screen X-ray film was wrapped in 0.005 inch aluminum and after developing the back emulsion was removed. This is a 72.6 hour exposure of a stationary crystal with a cylindrical camera. (The powder rings are not from the TTT_2I_3 (h.d.) crystal.) Optical film scans were made of the photographs in Figures 24 and 26 and of three other oscillation photographs taken under the same conditions but with different exposures (40). Figure 27 shows two of the film scans plotted as a function of $|\vec{s}|$. The



A Stationary X-Ray Diffraction Photograph of TTT_2I_3 (h.d.)
 (monochromatized MoK_α , 72.6 hrs, $\sim 294^\circ \text{K}$)

Figure 26



Comparison of Optical Film Scans with the Wright Model (Curve E, Figure 25b).

Figure 27

optical density plotted for the 135 hour exposure is an average of ten 0.1 mm scans of a 1 mm strip parallel to b . The optical density plotted for the 72.6 hour exposure, stationary photograph is an average of five 0.1 mm scans of a 0.5 mm strip parallel to \vec{b} . Also in Figure 27 is plot \underline{E} of Figure 25 b. Plot \underline{E} (the parameters are listed in Table XVb) shows all of the major features seen with the optical film scans. Although the fit may not be perfect, this one-dimensional model of the structure of the iodide chains can explain all of the major features of the TTT_2I_3 (h.d.) diffuse diffraction pattern. Although not seen with the film scans, the diffuse layers $k = 3'$ and $k = 7'$ actually consist of two components. This can be explained by the presence of two average spacings for I_3^- in the chains, for example I_3^- spacings of 10.0 Å and 9.6 Å.

Although the model is not perfect, in general it shows excellent agreement with the observed diffraction pattern. The agreement is good enough to say that, besides I_3^- , there is some finite amount of I_2 and probably a small amount of I^- present in the channels. This is a different conclusion from that of Isett and Perez-Albuerne (41). They interpret their Raman data as indicating the presence of less than 2% I_2 in Kodak crystals. The structural report by Smith and Luss and their descriptions of their X-ray photographs suggest, however, that the Kodak crystals are similar to the JPL crystals structurally. The presence of small amounts of I_2 and I^- in the channels would certainly explain why the iodide chains never three-dimensionally order. Although

at higher temperatures there is undoubtedly dynamic disorder, there is also, built into the structure, static disorder. The occasional mistake of an I_2 or I^- in the channel would prevent any long-range chain-chain ordering even at low temperatures.

From the information reported by Kaminskii, et al., (42) for Chernogolovka crystals and the diffuse X-ray scattering experiments done by Megtert, Pouget, Comès and Fourme on Chernogolovka crystals at low temperatures (43) the model presented here may not be a valid description of the Chernogolovka crystals. Some caution should be used in assuming that this model is completely valid for any TTT_2I_3 other than for TTT_2I_3 (h.d.) crystals grown at JPL since it is apparent that crystallization conditions can strongly effect the iodide composition of the resulting crystals.

2.4 Conclusions

The results of this study on TTT_2I_3 (h.d.) crystals grown at JPL are as follows:

(i) The room-temperature structure of the TTT lattice in TTT_2I_3 (h.d.) is the same as reported by Smith and Luss (9).

(ii) There is primarily I_3^- present in the channels, but I_2 and I^- are also occasionally present causing "mistakes" in the chains and preventing long-range chain-chain coupling.

(iii) With very slow cooling the TTT lattice in TTT_2I_3 (h.d.) apparently shows no structural phase transition to 19° K.

(iv) With very slow cooling the iodide chains never order which is consistent with (ii).

However, as mentioned in the Experimental Section for the low-temperature structures, all of the temperature-dependent data was collected on two crystals of TTT_2I_3 (h.d.), the fifth and sixth crystals studied. It might be of interest to mention what happened to the first four crystals.

The first crystal was cooled at a rate which would have cooled the crystal from 294° to 20° K in about 15-24 hours. The crystal shattered at 80-90°. The only piece of the crystal which could be found after warming to 294° K was the end of the crystal which was still firmly attached to the glass fiber.

The second crystal was cooled about twice as slowly, but cracked at $\sim 90^\circ$ K during the cooling. X-ray photographs of the crystal after warming to room temperature confirmed that the crystal really had split and remained split at room temperature.

The third crystal was coated with G. E. varnish after it was mounted on the glass fiber. It was cooled at a much slower rate. From 294° to $\sim 90^\circ$ K the temperature was lowered at $\sim 1^\circ$ per 5-10 minutes. From $\sim 90^\circ$ to 27° K the temperature was lowered at $\sim 0.1^\circ$ per 5 minutes. The crystal survived to the lowest temperature attained, $\sim 27^\circ$ K. During the data collection at 27° K the check reflections exhibited substantial decay.

Because a complete, reliable data set was not obtained from the third crystal, a fourth crystal, also coated with G. E. varnish, was cooled. This crystal was cooled rapidly from $\sim 294^\circ$ K to 27° K in about four hours. After 70 hours of data collection the check reflection intensities became very erratic, suggesting the structure was unstable and was changing with time.

The most reliable data collected from the third and fourth crystals were merged to form an almost complete data set. Solution of the structure using this data set suggested that the end ring of the TTT molecule was substantially bent away from planarity with the rest of the molecule. At room temperature the tilt of the end ring away from planarity is $\sim 1^\circ$ ($\pm 0.2^\circ$). This same tilt was observed at 164° K, 74° K, and 19° K with the good data from the fifth crystal. However, with the combined data set from the third and fourth crystals, the

end ring appeared to be tilted $\sim 3^\circ$ ($\pm 0.2^\circ$) away from the planarity of the rest of the TTT molecule. It is not known whether this is the result of the fast cooling or of a systematic error in the data caused by merging data sets from crystals cooled at different rates. The $k = 3'$ reflections observed with the fourth crystal also showed differences from the fifth and the sixth crystals. For the fifth and sixth crystals, reflections with h odd, $k = 3'$ (h and l here refer to the C-centered TTT lattice) were always absent. There were very, very weak intensities for reflections with h odd, $k = 3'$ collected from the fourth crystal.

Although the data from the third and fourth crystals are somewhat unreliable and the third and fourth crystals were the only ones coated with G. E. varnish, the possibility still exists that cooling rate may affect the structure. Point-by-point sampling of reciprocal space with a diffractometer is extremely inefficient and time-consuming. It is possible that there was additional information such as satellites in regions of reciprocal space which were not scanned. The loss of the first two crystals may suggest the existence of a structural phase transition. Crystals of TTT_2I_3 (h.d.) are often reported to break on cooling (8). Measurement of the transport properties on crystals cooled at very slow rates have not been done.

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- (20) Syntex P $\bar{1}$ computer-centering program. Reflections were centered with reduced tolerances at one-half or one-quarter of the default speed depending on their intensities.
- (21) For two observations of the same reflection $F_{ave} = [(F_1^2 w_1^2 + F_2^2 w_2^2) / (w_1^2 + w_2^2)]^{1/2}$.
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Chapter 3

Structure of (Tetrathiotetracene)-(Iodide)

Slight variations in the crystallization conditions of the tetrathiotetracene (TTT)-iodine system can lead to variations in the amount of disorder in TTT_2I_3 and, also, to other phases (1,2,3). When semiconducting crystals with a physical morphology very similar to metallic TTT_2I_3 were obtained at JPL, it was of interest to determine why the material was semiconducting (Appendix B) rather than metallic. To help elucidate why the material was semiconducting, a crystal structure determination was undertaken.

Experimental

The crystals of this phase were usually very thin, fragile needles with golden reflections. The density was measured by flotation to be 2.09 g/cm^3 . Initial X-ray oscillation photographs, taken with the crystals rotated about their needle axes, showed an ordered structure with a plethora of layer lines which could not be indexed on the basis of a single lattice. At least ten crystals were examined by X-ray photographs and all exhibited an identical diffraction pattern. The diffraction pattern can be indexed by assuming two interacting lattices. (This is discussed in more detail later.) It was decided that the lattice with the repeat spacing of about 3.64 \AA along the needle axis would be studied first.

Initial X-ray photographs of this lattice suggested a monoclinic cell. However, careful centering of fifteen reflections with a beam-splitter on a General Electric quarter-circle diffractometer and comparison of the integrated intensities of "equivalent" reflections confirmed that the unit cell chosen from the photographs actually has C-centered triclinic symmetry. All possible $hk0$ reflections to $2\theta = 155.0^\circ$ were collected with 1° per minute 2θ - θ scans on an automatic General Electric quarter-circle diffractometer using nickel-filtered copper K_α radiation. Backgrounds were collected for thirty seconds on each side of a scan. Equivalent Friedel-related reflections were combined by taking a weighted average of the F_o^2 (4). Weights were based on counting statistics and no more than two observations contributed to a weight. The structure in projection was solved from the two-dimensional, UVO Patterson projection. Partial refinement of the projection indicated the stoichiometry as TTTI (1:1) in agreement with the measured density. The crystal had dimensions of 0.83 mm x .008 mm x .517 mm (Table I). Absorption corrections (5) ($\mu = 212.875 \text{ cm}^{-1}$) were applied to the data set and full-matrix least-squares refinement continued until the refinement converged.

A complete three-dimensional data set for $c = 3.643 \text{ \AA}$ was collected, as before, with 1° per minute 2θ - θ scans using nickel-filtered copper radiation on a General Electric quarter-circle diffractometer. Thirty second backgrounds were collected on both sides of a scan. Equivalent Friedel-related reflections were combined to yield a data set of 1620 reflections of which 1512 had $F_o^2 > 0$.

Table ICrystal Data TTTI (Lattice 1, $c = 3.643 \text{ \AA}$)

$a = 13.028(2) \text{ \AA}$	$\alpha = 90.81(1)^\circ$	$(\lambda = 1.5418 \text{ \AA})$
$b = 16.445(2)$	$\beta = 96.11(1)$	
$c = 3.643(1)$	$\gamma = 91.11(1)$	

$V = 776 \text{ \AA}^3$

$Z = 2$

$F.W. = 479.4$

$d_c = 2.053$

$d_m = 2.09(3)$

Triclinic, Space group $C\bar{1}$ Linear absorption coefficient (CuK_α), $\mu = 212.875 \text{ cm}^{-1}$.Crystal dimensions $0.083 \text{ mm} \times 0.008 \text{ mm} \times 0.517 \text{ mm}$.

The same crystal used to collect the hk0 data was used for the complete three-dimensional data collection. Absorption corrections for the stoichiometry TTTI ($\mu = 212.875 \text{ cm}^{-1}$) were applied to the complete data set. The z coordinates of most of the atoms were obtained from the three-dimensional Patterson map. Iodine and sulfur atomic scattering factors were corrected for anomalous dispersion, and the real parts ($f_0 + \Delta f'$) were used in the structure factor calculations (6). After the positions of all the atoms (except hydrogens) were determined, the structure was refined by the full-matrix least-squares method. The weights used in the least-squares refinement were based on counting statistics. On a low-angle difference Fourier map of the approximately refined structure, very large, weak blobs of electron density were seen in positions that were geometrically correct for hydrogen atoms. The hydrogen atoms were included in the final least-square cycles, but were not refined. During the final least-square cycle all shifts were less than one-third of the corresponding standard deviation. Final atom coordinates and temperature factors are listed in Table II. The full matrix least-squares refinement converged to $R = 0.102$, with the goodness-of-fit = 4.7 for the complete data set (7). For 1132 reflections with $F_o^2 \geq 3\sigma(F_o^2)$, $R = 0.081$.

Discussion

Shown in Figure 1 and listed in Table III are bond lengths and bond angles, and their estimated standard deviations (assuming isotropic

Table IIa

Refined Atomic Coordinates for TTTI (Lattice 1, $c = 3.643 \text{ \AA}$)

	<u>X</u>	<u>Y</u>	<u>Z</u>
I	0	0	0
S	.25896(17)	.06298(14)	.94596(81)
S'	.25951(17)	-.05920(14)	1.08012(81)
C1	.44396(60)	.00078(53)	1.01076(291)
C2	.39217(63)	.07140(52)	.91656(261)
C2'	.39269(63)	-.07028(53)	1.09972(252)
C3	.44302(67)	.14283(51)	.81213(253)
C3'	.55460(65)	.14147(51)	.80716(266)
C4	.39062(71)	.21558(55)	.71329(284)
C4'	.60321(74)	.21412(56)	.70358(271)
C5	.44277(86)	.28337(56)	.61511(318)
C5'	.55050(82)	.28219(57)	.61810(298)
H1	.32	.205	.56
H2	.402	.326	.54
H1'	.67	.222	.76
H2'	.59	.329	.61

Table IIb

Refined Temperature Factors for TTTI (Lattice 1, $c = 3.643 \text{ \AA}$)

	<u>U₁₁</u>	<u>U₂₂</u>	<u>U₃₃</u>	<u>U₁₂</u>	<u>U₁₃</u>	<u>U₂₃</u>
I	.0494(6)	.0349(5)	.4702(43)	.0020(4)	.0932(13)	-.0075(11)
S	.0465(12)	.0559(12)	.0460(18)	.0180(10)	.0121(12)	-.0006(12)
S'	.0489(12)	.0568(12)	.0449(18)	.0116(10)	.0143(12)	-.0016(13)
C1	.0400(42)	.0523(44)	.0380(59)	.0150(36)	.0061(43)	-.0078(44)
C2	.0476(44)	.0508(43)	.0239(54)	.0145(37)	.0015(41)	-.0072(42)
C2'	.0475(44)	.0528(44)	.0193(51)	.0076(37)	-.0027(40)	-.0082(42)
C3	.0562(49)	.0473(42)	.0158(48)	.0142(38)	-.0033(40)	-.0046(40)
C3'	.0520(47)	.0474(41)	.0253(55)	.0070(37)	.0063(43)	-.0038(42)
C4	.0630(57)	.0540(49)	.0321(59)	.0166(44)	.0018(48)	-.0072(47)
C4'	.0673(56)	.0559(38)	.0194(52)	.0044(44)	.0091(44)	-.0015(44)
C5	.0826(70)	.0461(45)	.0477(76)	.0190(47)	.0090(59)	-.0030(51)
C5'	.0782(67)	.0524(48)	.0314(62)	.0067(47)	.0001(52)	-.0073(48)

B

H1	8.0
H2	8.0
H1'	8.0
H2'	8.0

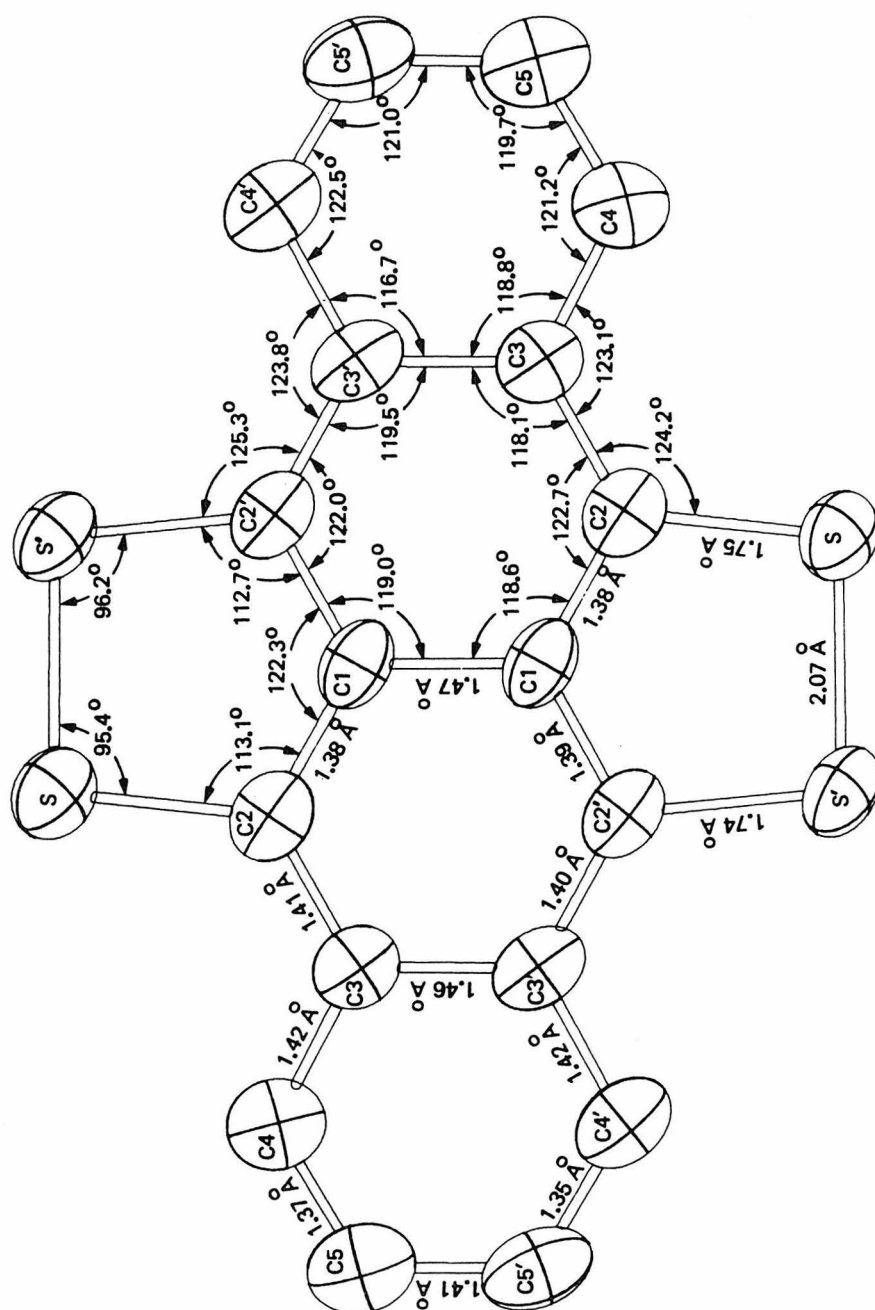


Figure 1

Tetrathiotetracene cation in TTTI

(Based on the refinement of Lattice 1, $c=3.64 \text{ \AA}$)

Thermal ellipsoids are drawn at the 50% probability level

Table III

TTT Geometry in TTTI (Lattice 1, $c = 3.643 \text{ \AA}$)

Bond Lengths, \AA^*		Bond Angles*	
C1-C1	1.471(18)	C1-C1-C2	118.5(1.6) $^\circ$
C3-C3'	1.456(12)	C1-C1-C2'	119.0(1.6)
C5-C5'	1.403(15)	C2'-C1-C2	122.3(8)
C1-C2	1.383(13)	C2'-S'-S	96.3(3)
C1-C2'	1.394(13)	C2-S-S'	95.4(3)
C2-C3	1.414(12)	C1-C2-S	113.1(7)
C2'-C3'	1.396(12)	C3-C2-S	124.2(7)
C3-C4	1.421(13)	C1-C2'-S'	112.7(7)
C3'-C4'	1.416(13)	C3'-C2'-S'	125.3(7)
C4-C5	1.366(14)	C1-C2-C3	122.7(8)
C4'-C5'	1.348(14)	C1-C2'-C3'	122.0(8)
C2-S	1.753(9)	C2-C3-C3'	118.1(8)
C2'-S'	1.742(9)	C2'-C3'-C3	119.5(8)
S-S'	2.075(7)	C2-C3-C4	123.2(8)
H1-C4	0.93	C3'-C3-C4	118.7(8)
H1'-C4'	0.88	C3-C3'-C4'	116.7(8)
H2-C5	0.91	C3-C4-C5	121.3(9)
H2'-C5'	0.92	C3'-C4'-C5'	122.6(9)
		C4-C5-C5'	119.7(1.0)
		C4'-C5'-C5	120.9(1.0)

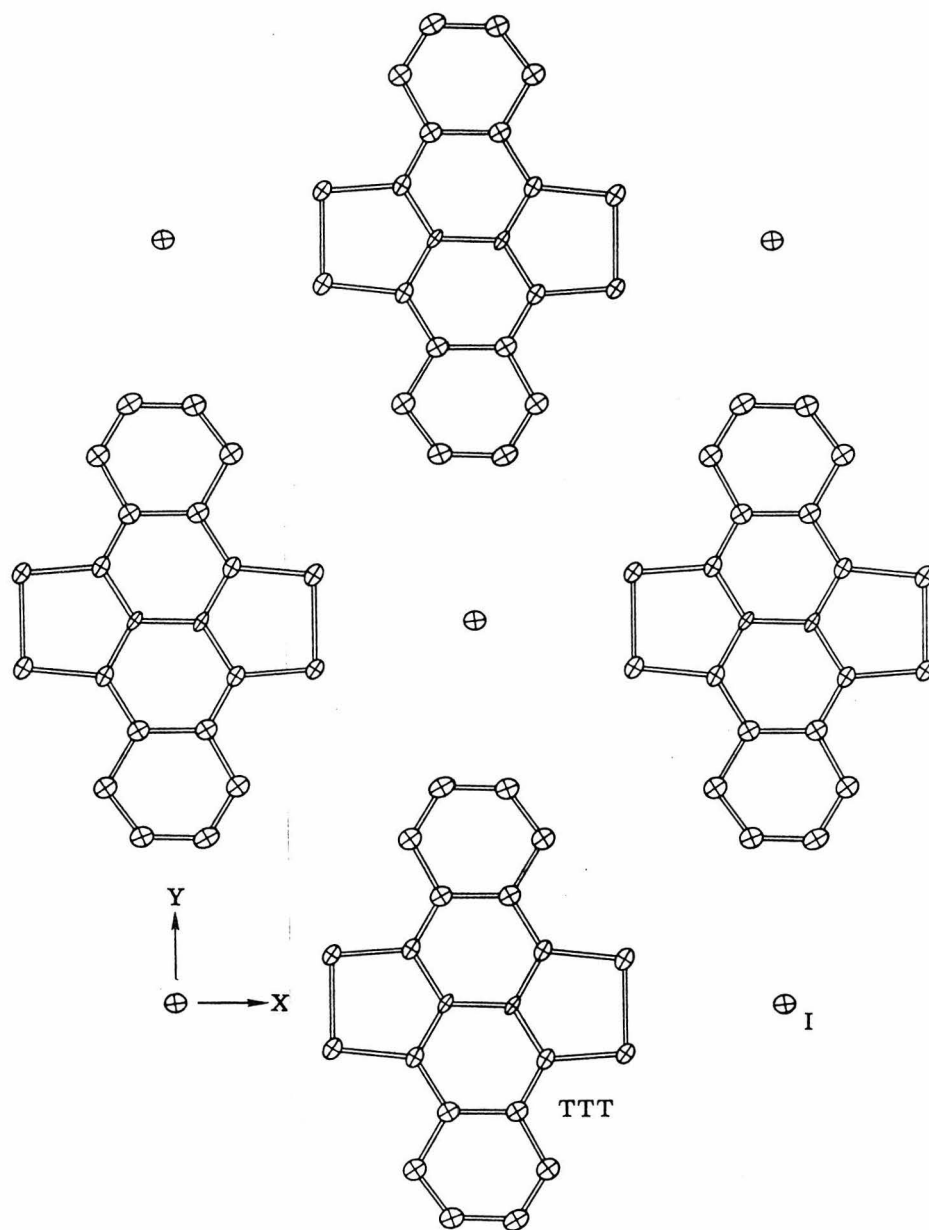
Direction cosines of the normal to the TTT best plane: 0.0162, 0.2834, 0.9469.

Interplanar distance 3.56 \AA .

*The numbers in the parentheses are the deviations. The hydrogen atoms were not refined.

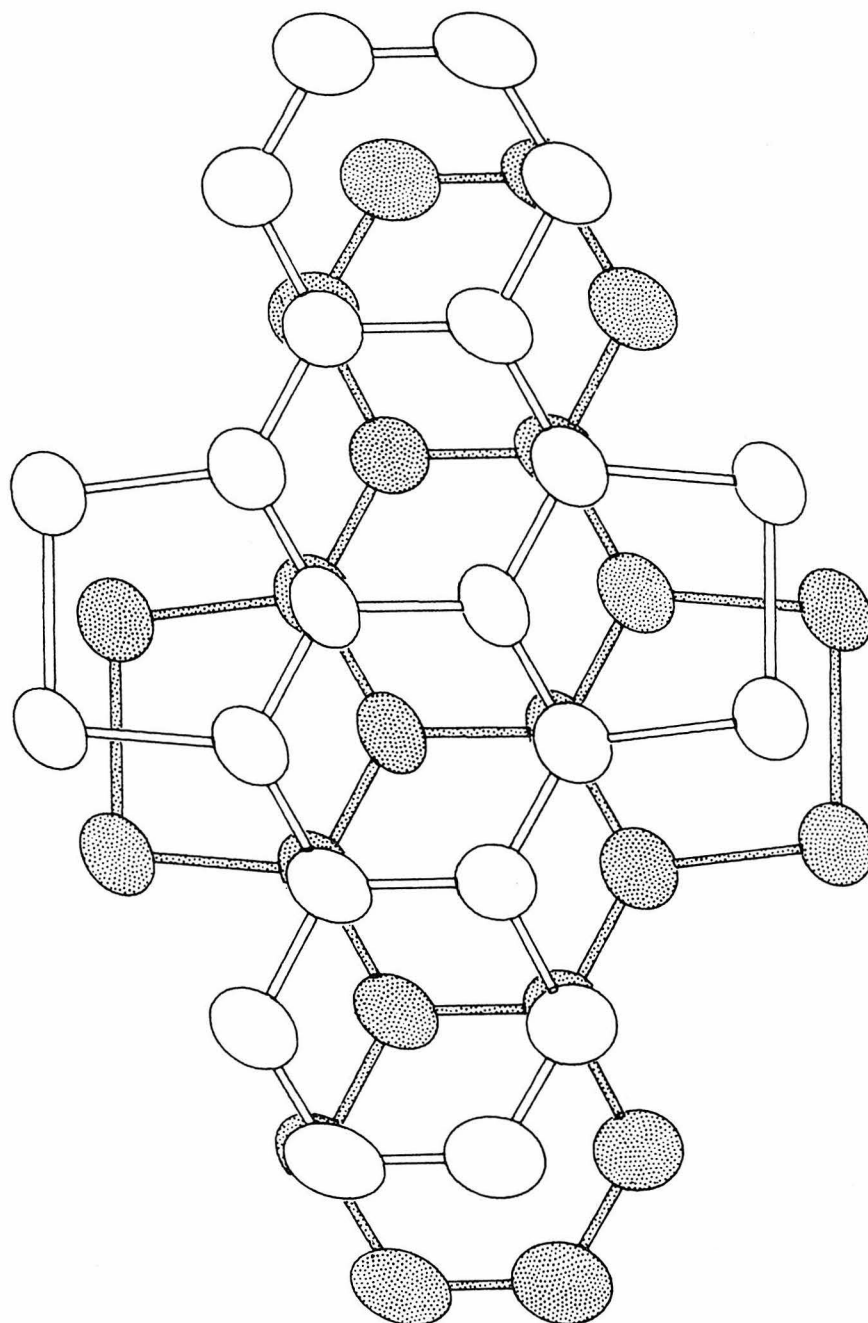
atoms) for the TTT molecule. The TTT molecular geometry is very similar to that of a TTT molecule in TTT_2I_3 (previous Chapter or neutral TTT (8)). However, the arrangement of cation stacks and anion channels is much different in TTTI than in the conducting TTT_2I_3 . An [001] projection of TTTI is shown in Figure 2. Whereas in conducting TTT_2I_3 the crystal structure suggests significant interchain coupling, each TTT stack in TTTI is essentially isolated by surrounding iodides from neighboring TTT stacks. There might be only very weak interchain coupling through S-H2 contacts ($\sim 3.10 \text{ \AA}$) or S'-H2' contacts ($\sim 3.04 \text{ \AA}$), although these distances are approximately equal to the sum of the van der Waals radii (3.05 \AA) (9). The overlap of adjacent TTT molecules in the same stack is also significantly different in TTTI from that of TTT_2I_3 . Two adjacent molecules in the same stack viewed perpendicular to the molecular plane are shown in Figure 3. Because there is more overlap in TTTI than in TTT_2I_3 and since the stoichiometry implies a half-filled band, one might expect either a good conductor or a dimerized semiconductor. The transport properties (Appendix B) do suggest semiconductor behavior, but the TTT molecules do not appear to be dimerized. However, the interpretation of transport properties in the presence of two incommensurate lattices is not well understood (10).

As mentioned before the actual crystal structure of this phase consists of two lattices. Shown in Figure 4 is an X-ray oscillation photograph of TTTI rotated about the needle axis, \vec{c} , and taken with nickel-filtered copper radiation. In the approximate position of a sixth layer line there are actually two distinct layer lines. Careful



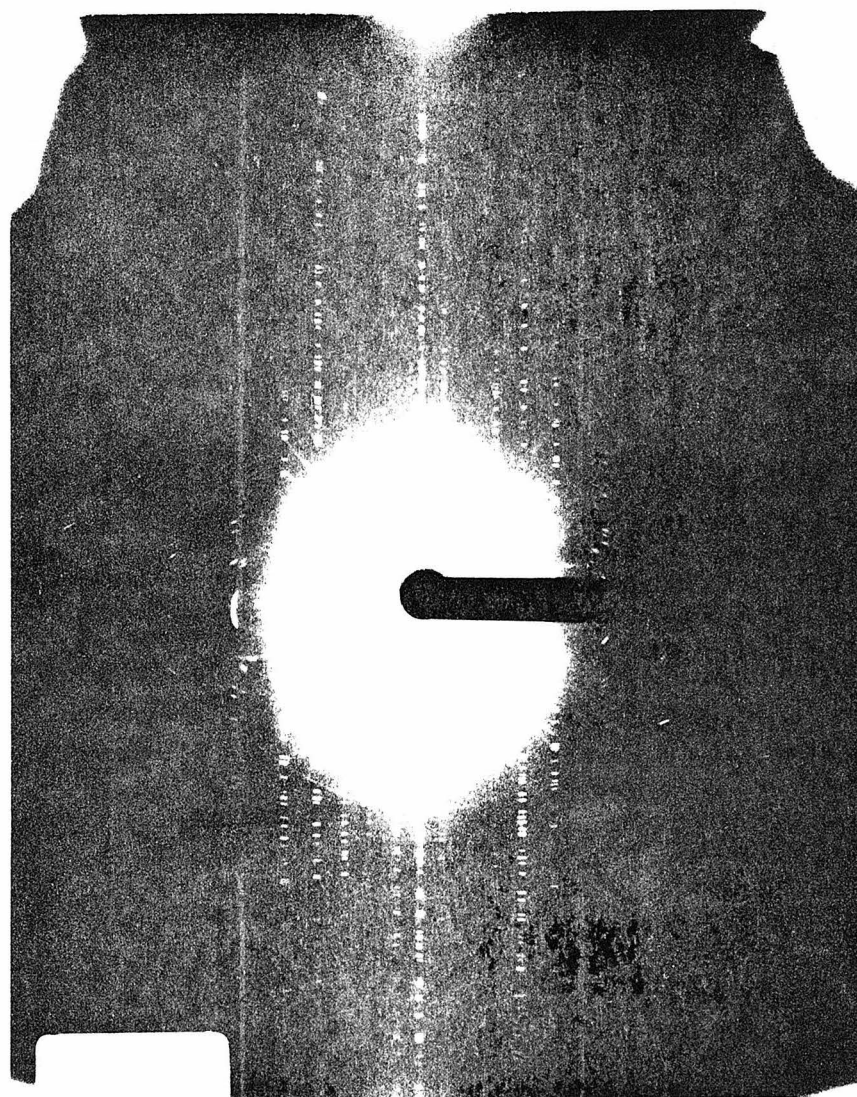
[001] Projection for TTTI

Figure 2



Semiconducting (Tetrathiotetracene)(Iodide), TTTI
Neighboring Tetrathiotetracene molecules, in the same cation stack,
viewed perpendicular to the molecular plane.
(Based on the refinement of Lattice 1, $c=3.64 \text{ \AA}$)

Figure 3



An X-Ray Oscillation Photograph of TTTI ($\text{CuK}\alpha$)

Figure 4

measurements of the layer line spacings and the presence of two "sixth" layer lines strongly suggest that a single lattice description of the TTTI structure is inadequate. The layer line spacings obtained from oscillation photographs of two different TTTI crystals are listed in Table IV. These measurements are not absolute, but the measurements for each set of layer line spacings are internally consistent. Although the spacings cannot be fit to a single lattice in the \vec{c} direction, they can be quite adequately fit by assuming the presence of two incommensurate repeat distances in the \vec{c} direction. Structures exhibiting two or more incommensurate lattices are rare, but have been reported (for example 11,12,13). Conceptually TTTI can be viewed as consisting of two structures modulating each other. C. K. Johnson has reported an elegant analysis of the TTF_7I_5 structure, which also consists of two incommensurate lattices (13). Following the indexing method suggested by Johnson, Figure 5 is a graph of the "selection rule" $\ell = 4w + 3t$ where ℓ represents the indexing of an approximate supercell and w and t represent indices of the two sublattices. This graph suggests there might be two layer lines of slightly different spacings for $\ell = 2$ and two layer lines of slightly different spacings for $\ell = 6$. Although two slightly separated layer lines at $\ell = 2$ were not observed, perhaps for lack of intensity, two slightly separated layer lines at $\ell = 6$ were definitely observed. The actual positions of the layer lines can be fit by assuming that planes of spacing 3.642 \AA are periodically displaced by an error occurring every 4.787 \AA . The period of displacement would occur every Q number of planes (11,14). For

Table IV

Layer Line Spacings of TTTI

<u>Crystal #1</u>	<u>Crystal #2</u>	<u>Interpretation^a</u>	<u>Calculated Spacings^b</u>
15.07 Å	15.57 Å	1, $\bar{1}$	15.26 Å
6.86	6.98	$\bar{1}$, 2	6.97
4.75	4.79	0, 1	4.78
3.62	3.65	1, 0	3.64
2.93	2.94	2, $\bar{1}$	2.94
2.458	2.465	3, 2	2.47
2.385	2.391	0, 2	2.39
2.064	2.069	1, 1	2.07
1.822	1.817	2, 0	1.82

^a _{w,t} w is the index for Lattice 1; t is the index for Lattice 2.

^b _d = $\pm c_1 / (|w| \pm |t|/Q)$, $c_1 = 3.642 \text{ Å}$ and $Q = 1.314$.

$\begin{array}{c} w \\ t \end{array}$	0	+1	+2	+3
0	0	4	8	12
+1	3	7	11	15
-1	-3	1	5	9
+2	6	10	14	18
-2	-6	-2	2	6

W indices for Lattice 1

t indices for Lattice 2

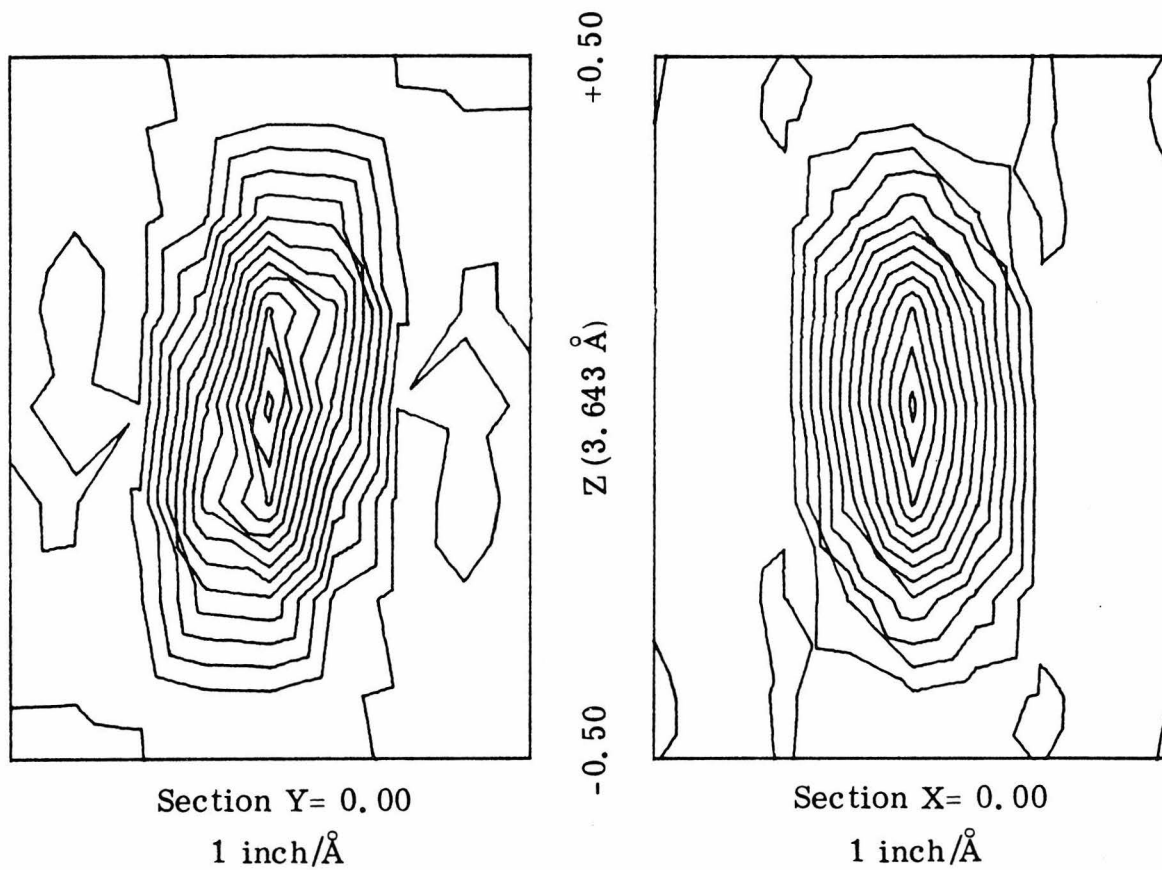
$$\ell = 4w + 3t$$

Calculations for the indexing of TTTI X-ray Oscillation Photographs

Figure 5

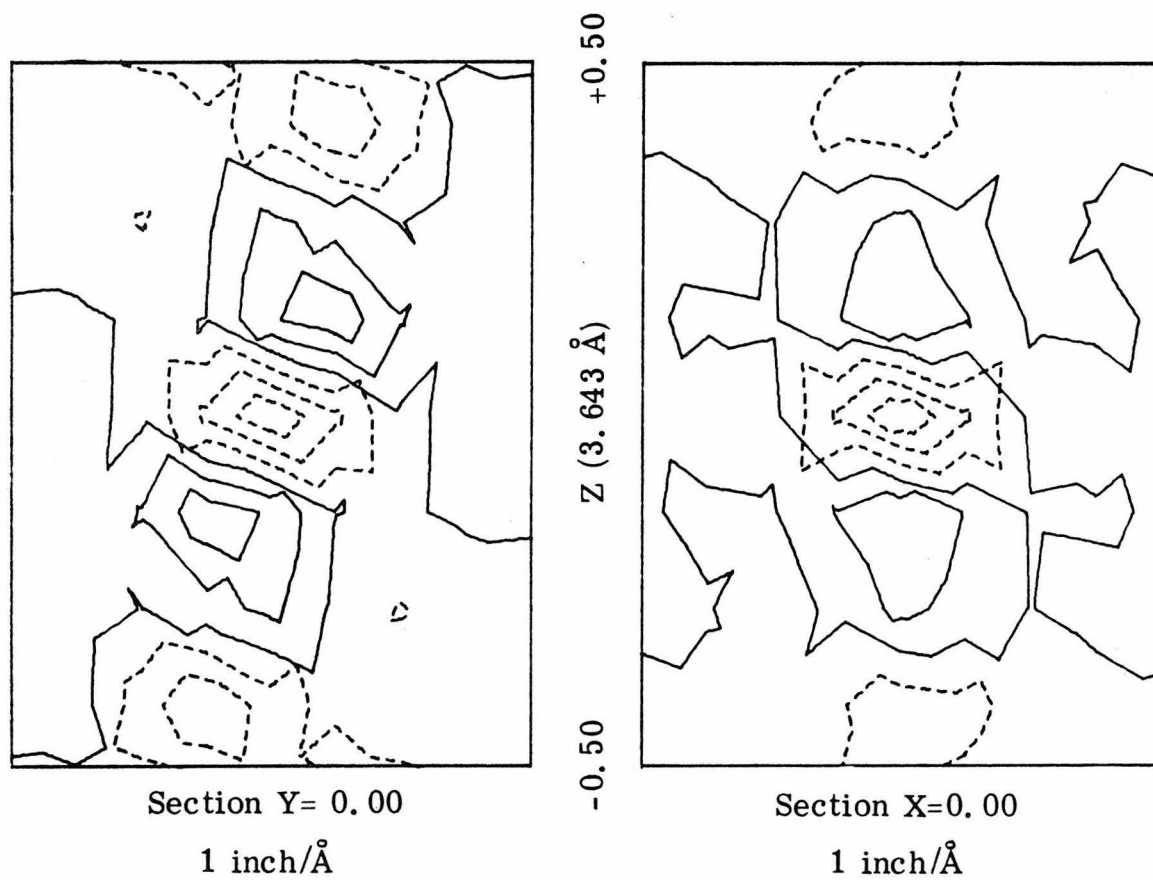
TTTI, $Q = 4.787/3.642 = 1.314$ planes. If Q equalled a whole number of planes (or a rational fraction), then the periodic error becomes part of the repeat pattern of an exact supercell. For example, if $Q = 4 \text{ \AA}/3 \text{ \AA} = 1.3333$, then the structure could equally well be described by a supercell with a repeat of 12 \AA . In TTTI Q is very far from being a rational fraction. If the periodic displacement is small, James has shown that additional layer lines will be observed at spacings $d = \frac{c_1}{|w| \pm |t|/Q}$ (11,14). James calls the extra lines "ghosts" or satellites, but they could equally well be described as the interference cross terms from the diffraction of two lattices. Shown in the last column of Table IV are the calculated d spacings for TTTI using the w and t indices found from the graph in Figure 5 and based on the values of $c_1 = 3.642 \text{ \AA}$, $c_2 = 4.787 \text{ \AA}$, $\lambda = 1.5418 \text{ \AA}$ (CuK_{α}). With the exception of the first layer line the agreement is quite good suggesting that TTTI does consist of two lattices incommensurate along \vec{c} , but commensurate in a^*, b^* .

Although only the structure of lattice 1 ($c=3.64 \text{ \AA}$) has been solved and refined, by analogy to other known channel structures, such as TTF_7I_5 (13), TTFBr_x (15), TMTSF(SCN)_x (16), one can postulate that lattice 2 ($c \sim 4.78 \text{ \AA}$) is due to the iodide atoms. This is reasonable considering that refinement of the $hk0$ projection of both lattices indicated the stoichiometry as TTTI (1:1) which, assuming complete charge-transfer implies I^- . Two times the van der Waals radius of I^- is $\sim 4.3 \text{ \AA}$ which is intermediate between the lattice spacings 3.64 \AA and 4.79 \AA . In the lattice 1 structure the iodide is at $x = 0$, $y = 0$, $z = 0$ (Figure 6) but the root-mean-square (rms) displacement along \vec{c} is



TTTI
Electron Density about 0, 0, 0
Contoured at $2e^-/\text{\AA}$ beginning at zero

Figure 6



TTTI

Difference Fourier Sections about 0, 0, 0

—— zero and positive 0.5 e⁻/Å contours

----- negative 0.5 e⁻/Å contours

Figure 7

approximately 0.69 \AA which is about 3.6 times as large as the rms displacement in the other two directions. Difference Fourier sections through the iodide (Figure 7) indicate that positioning the iodide at $0,0,0$ in lattice 1 is a reasonable model but the iodide does appear to be at $x = 0, y = 0, z \approx 0.197$ occasionally. D_{I-I} for $z = 0$ to $z = 1.197$ is about 4.4 \AA or the expected van der Waals distance between I^- atoms. This structural determination has shown that:

- (A) the crystals have golden reflection and look very similar to TTT_2I_3
- (B) this phase is apparently semiconducting
- (C) the stoichiometry of this semiconducting phase is $TTTI$ (1:1)
- (D) the structure consists of two lattices which are incommensurate along \vec{c} , the needle axis of the crystal
- (E) the two repeat distances along \vec{c} are $c_1 = 3.643(1) \text{ \AA}$ and $c_2 \approx 4.79 \text{ \AA}$.

In 1976 Buravov, et al., (1) reported crystallizing two phases of tetrathiotetracene-iodide. One phase was the highly conducting TTT_2I_3 ; the other was a semiconducting phase with the reported stoichiometry of TTT_4I_4 . Their crystal data for this TTT_4I_4 is listed in Table V. Except for an approximate quadrupling of c_1 , their unit cell is the same as $TTTI$. Although they reported solving the hko projection and were attempting the three-dimensional structure, the only structural information reported was "the existence of parallel stacks of NBDT $[TTT]$ and I along the c-axis is characteristic of the structure" (1). Either Buravov and coworkers did not notice the existence of two incommensurate

Table VComparison of the Unit Cell Parameters of TTT_{44}I ^a and TTTI

TTT_{44}I ^a	TTTI ^b
$a = 13.036(2) \text{ \AA}$	$a = 13.028(2) \text{ \AA}$
$b = 16.435(2)$	$b = 16.445(2)$
$c = 14.582(2)$	$c = 3.643(1)$
$\alpha = 90^{\circ}42'(3)$	$\alpha = 90.81(1)^{\circ}$
$\beta = 83^{\circ}49'(3)$	$\beta = 96.11(1)$
$\gamma = 88^{\circ}53'(3)$	$\gamma = 91.11(1)$
Space group, $\text{C}\bar{1}$	Space group, $\text{C}\bar{1}$

^aL. I. Buravov, G. I. Zvereva, V. F. Kaminskii, L. P. Rosenberg, M. L. Khidekel, R. P. Shibaeva, I. F. Schegolev, and E. B. Yagubskii, J. Chem. Soc. Chem. Comm., 720 (1976).

^bThis study (Lattice 1, see text).

lattices and used an approximate supercell or they were able to crystallize a semiconducting TTTI with an ordered, single lattice structure. If they did crystallize an ordered, semiconducting TTTI, it would be of interest to know if the TTT molecules were dimerized as would be expected for a static Peierls distortion arising from the half-filled band.

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Chapter 4

Structure of (Hexamethylenetetraselenofulvalene)(Iodide)_x

Hexamethylenetetraselenofulvalene-tetracyanoquinodimethane (HMTSF-TCNQ) is the best organic metal known (1). At 300° K its conductivity is $\sim 2000 \Omega^{-1} \text{ cm}^{-1}$ and increases with decreasing temperature to about 110° K. Between 45° and 75° K the conductivity goes through a broad maximum and then decreases with decreasing temperature. But the low temperature state appears to be semi-metallic. At 1° K the conductivity is $\sim 600 \Omega^{-1} \text{ cm}^{-1}$. Crystals of HMTSF-TCNQ remain conducting to 0.045° K (1). HMTSF-TCNQ is a two charge-carrier system. It would be of interest to synthesize a single-carrier conductor based on HMTSF for comparison with HMTSF-TCNQ.

Experimental

Single crystals of HMTSF-I_x were prepared by R. Williams at Caltech. HMTSF, supplied by Dr. D. Cowan and Dr. P. Shu of Johns Hopkins University, was chemically oxidized to HMTSF(ClO₄)_x by a very dilute solution of H₂O₂ and HClO₄ in acetonitrile. Solutions of HMTSF(ClO₄)_x in acetonitrile and tetrabutylammonium iodide in acetonitrile were allowed to slowly diffuse together. Single crystals of HMTSF-I_x grew near the interface.

Two crystalline modifications of HMTSF-I_x were obtained by diffusion. Both crystalline modifications formed dark needles. One modification

formed very poor quality crystals and had a repeat spacing along the needle axis of 10.24 Å. The second modification, the one reported here, formed nice, dark needles. Initial X-ray diffraction photographs showed the crystals to be triclinic with a repeat spacing along the needle axis, \vec{a} , of about 8.08 Å

A single crystal with the dimensions 0.396 mm x 0.058 mm x 0.066 mm was chosen for data collection. Based on a least-squares fit of thirty-eight 2θ values obtained with a Syntex P2₁ diffractometer the unit cell parameters were determined to be $a = 8.056(4)$ Å, $b = 12.740(4)$ Å, $c = 8.016(3)$ Å, and $\alpha = 81.72(4)^\circ$, $\beta = 67.73(5)^\circ$, $\gamma = 102.64(4)^\circ$ for $\lambda = \text{MoK}_{\alpha} = 0.71069$ Å (Table I). The numbers in parentheses represent the estimated standard deviation in the last decimal place. A complete data set to $2\theta = 60^\circ$ was collected on an automated Syntex P2₁ diffractometer with monochromatized molybdenum K_{α} radiation. Intensities were collected with 1° per minute 2θ - θ scans. The time spent collecting background was 50% of the amount of time spent scanning a reflection, or about one minute for each side of a scan. The integrated intensities were corrected for Lorentz and polarization effects. The weights assigned to the reflections and used in the refinement were based on counting statistics. No more than two observations contributed to a weight.

Initial selenium atom coordinates were found from a three-dimensional Patterson map. Preliminary structure factor-Fourier calculations were based on the noncentric space group P1. After most of the atom

Table ICrystal Data for HMTSF-I_x

$$a = 8.056(4) \text{ \AA}$$

$$b = 12.740(4)$$

$$c = 8.016(3)$$

$$\alpha = 81.72(4)^\circ$$

$$\beta = 67.73(5)$$

$$\gamma = 102.64(4)$$

$$V = 715.2(9) \text{ \AA}^3$$

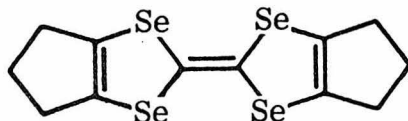
$$Z = 2$$

Space group $P\bar{1}$

Linear absorption coefficient (MoK $_{\alpha}$), $\mu = 111.938 \text{ cm}^{-1}$ (for the stoichiometry HMTSF₂I_{0.64}).

Stoichiometry \sim HMTSF₂I_{0.31}

Crystal dimensions 0.396 mm x 0.058 mm x 0.066 mm

HMTSF \equiv 

coordinates had been determined and crudely refined, the center of symmetry (inversion center) became obvious. All subsequent calculations were based on the space group $P\bar{1}$. The real parts of the atomic scattering factors of iodine and selenium after correction for anomalous dispersion ($f_0 + \Delta f'$) were used for structure factor calculations (2). There are two HMTSF molecules per unit cell, and they are related to each other by the inversion center. Preliminary least-squares refinement of the structure suggested that the stoichiometry was $\sim(\text{HMTSF})_2(\text{I})_{0.64}$ /unit cell. Absorption corrections based on this stoichiometry ($\mu = 111.938 \text{ cm}^{-1}$) were applied to the data set (3,4). Although the crystal used for data collection had six faces parallel to \vec{a} , two of the faces were too small to measure. So for the absorption corrections the crystal was approximated as being bounded by the major faces, the (001), (00 $\bar{1}$), (010) and (0 $\bar{1}$ 0) faces, and the ends by the (100) and ($\bar{1}$ 00) planes.

After several full-matrix least-squares refinement cycles the HMTSF molecule was refined but the iodine at 1/2,1/2,1/2 would not converge. Difference Fourier sections about 1/2,1/2,1/2 showed that a single iodine atom at 1/2,1/2,1/2 was a poor description of what was actually present. Further refinement of the structure was accomplished by adjusting the "solvent" atoms, based on the appearance of the difference Fourier sections, followed by a least-squares refinement of all of the HMTSF parameters and the population and temperature factors of the iodine. The "solvent" atoms were assumed

to be approximately equal to carbon in scattering power and were given carbon atomic scattering factors. An unsuccessful attempt was made to find the hydrogen atoms on a low-angle three-dimensional difference Fourier map. The hydrogens have not been included in the structure. The structure was judged to be refined when the R value stopped changing and the difference Fourier map looked reasonable. The final difference Fourier map showed a maximum range of $-2.3 \text{ e}/\text{\AA}^3$ to $+2.1 \text{ e}/\text{\AA}^3$. All shifts in the final least-squares cycle were less than $1/3 \sigma$. For the complete data set of 4209 reflections (3611 reflections had $F_o^2 > 0$) $R = 0.098$, with a goodness-of-fit of 1.65 (5). For the 2038 reflections with $F_o^2 > 3\sigma(F_o^2)$, $R = 0.051$. Listed in Table I are crystal data. Listed in Table II are refined atomic coordinates and thermal parameters.

Discussion

One of the primary objectives of a structure determination of a quasi-one-dimensional conductor which contains halide is to determine its stoichiometry. Because one can assume with some confidence that all of the halide present will be as X^- , knowing the amount of X^- in the structure will make the calculation of the number of charge carriers (holes) easy. This is valuable for interpretation of the transport properties. Unfortunately, stoichiometry is the most poorly defined parameter of this structure. The "refined" value of 0.3144 iodines per two HMTSF molecules is, obviously, dependent on the amount

Table IIa

Atomic Coordinates

	<u>X</u>	<u>Y</u>	<u>Z</u>
I	.5	.5	.5
Se1	-.36099(14)	.09124(8)	.20432(12)
Se2	-.18153(14)	.12862(8)	.33786(12)
Se3	-.14371(14)	.05500(8)	-.22328(12)
Se4	.05146(14)	.27062(7)	-.09138(12)
C1	-.19470(116)	.04781(67)	.02878(109)
C2	-.39847(120)	-.14949(66)	.01729(108)
C3	-.52680(143)	-.26156(73)	.04128(134)
C4	-.48679(155)	-.26479(74)	-.16303(131)
C5	-.35653(137)	-.14793(75)	-.29290(124)
C6	-.30915(133)	-.08978(69)	-.16280(118)
C7	-.12495(123)	.13476(71)	.08311(106)
C8	-.03743(122)	.27965(69)	.27695(118)
C9	-.01853(172)	.34484(83)	.41394(137)
C10	.08380(183)	.46475(74)	.28498(143)
C11	.17392(143)	.45430(66)	.07938(133)
C12	.06052(128)	.33674(65)	.09765(113)

Table IIb

Anisotropic Temperature Factors

	<u>U₁₁</u>	<u>U₂₂</u>	<u>U₃₃</u>	<u>U₁₂</u>	<u>U₁₃</u>	<u>U₂₃</u>
I	.0657(35)	.3266(110)	.0459(25)	-.0246(45)	-.0140(23)	-.0092(37)
Se1	.0347(6)	.0274(5)	.0218(4)	.0005(4)	-.0073(4)	-.0004(4)
Se2	.0378(6)	.0248(5)	.0228(4)	.0020(4)	-.0099(4)	-.0022(3)
Se3	.0341(6)	.0251(5)	.0228(4)	.0004(4)	-.0106(4)	-.0005(3)
Se4	.0359(6)	.0246(5)	.0239(4)	.0030(4)	-.0091(4)	.0000(3)
C1	.0255(47)	.0266(41)	.0308(41)	.0053(37)	-.0064(37)	-.0031(33)
C2	.0319(49)	.0260(40)	.0304(40)	.0060(37)	-.0156(38)	-.0055(32)
C3	.0491(63)	.0269(46)	.0474(54)	-.0054(45)	-.0178(49)	-.0068(40)
C4	.0681(75)	.0261(43)	.0431(50)	.0023(48)	-.0273(52)	.0010(37)
C5	.0425(57)	.0307(45)	.0384(47)	.0015(43)	-.0158(43)	-.0057(37)
C6	.0465(58)	.0275(42)	.0379(46)	.0105(41)	-.0230(43)	-.0072(35)
C7	.0315(50)	.0321(45)	.0265(40)	.0060(40)	-.0064(37)	-.0028(33)
C8	.0302(49)	.0309(43)	.0402(46)	.0057(38)	-.0182(40)	-.0101(36)
C9	.0837(85)	.0380(52)	.0452(54)	.0085(56)	-.0350(59)	-.0160(43)
C10	.1027(99)	.0235(44)	.0546(60)	-.0008(56)	-.0350(66)	-.0052(41)
C11	.0513(63)	.0151(38)	.0510(54)	-.0058(41)	-.0198(50)	.0011(36)
C12	.0389(54)	.0225(39)	.0332(41)	.0057(39)	-.0100(40)	-.0020(33)

Population

I .31443(453)

of "solvent" assumed to be present. By assuming that only iodide or disordered CH_3CN is present, it is reasonable to also put part of a carbon atom at $1/2, 1/2, 1/2$. For 0.39C with a B = 12.0 at $1/2, 1/2, 1/2$, there is also $\sim 0.3144\text{I}$ at $1/2, 1/2, 1/2$. But the actual amount of iodide present could differ from that value by as much as 30%. Another possibility is that the "solvent" is not always CH_3CN but could occasionally be ClO_4^- , which would affect any calculations of the amount of charge transferred. Listed in Table III are the atomic coordinates, populations, and thermal parameters used to model the electron density at and about $1/2, 1/2, 1/2$. The populations and temperature factors are not meaningful and are highly correlated. The temperature factors for the "solvent" atoms were chosen arbitrarily. An attempt was made to use electron microprobe analysis to determine the amount of iodine present in the crystals. Although selenium was found, no iodine was found. Because of the fairly unstable nature of the crystals in solution (6), it was felt that the lack of iodine could be attributed to decomposition of the crystals in the electron beam.

The HMTSF molecule looks quite normal. ORTEP drawings (7) of the HMTSF molecule are shown in Figure 1. There are no significant differences between the HMTSF geometry found in this structure determination and the HMTSF geometry in HMTSF-TCNQ(F_4) (8). Figure 2 is an ORTEP showing the arrangement of HMTSF molecules in one cation stack. The HMTSF molecules slip back and forth much the same as in TMTSF- $\text{Br}_{0.82}$ (tetramethyltetraselenofulvalene) (9). The orientation of

Table III

	<u>X</u>	<u>Y</u>	<u>Z</u>	<u>B</u>	<u>Populations</u>
SOLC	.5	.5	.5	12.0	.39
SOL1	.558	.521	.321	10.0	.33
SOL4	.38	.45	.400	9.0	.25
SOL2	.36	.466	.67	9.0	.33
SOL3	.52	.355	.55	10.0	.32
SOL5	.51	.478	.655	10.0	.17
SOL8	.301	.477	.6	10.0	.20

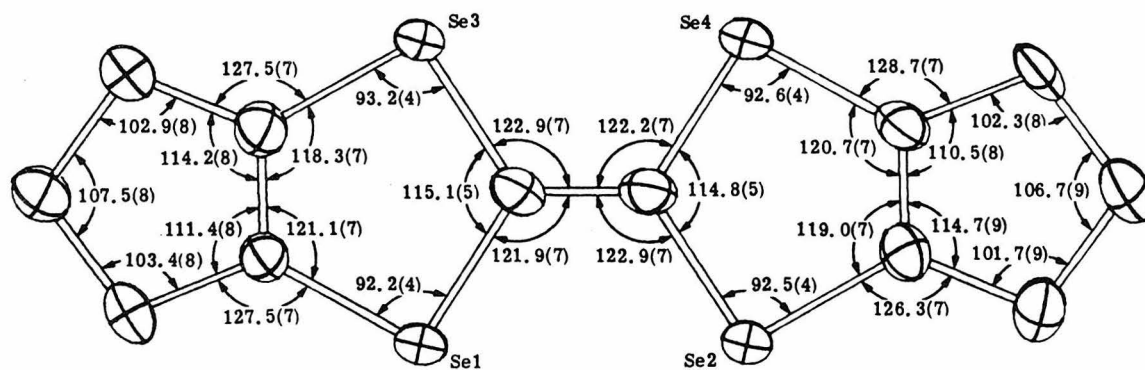
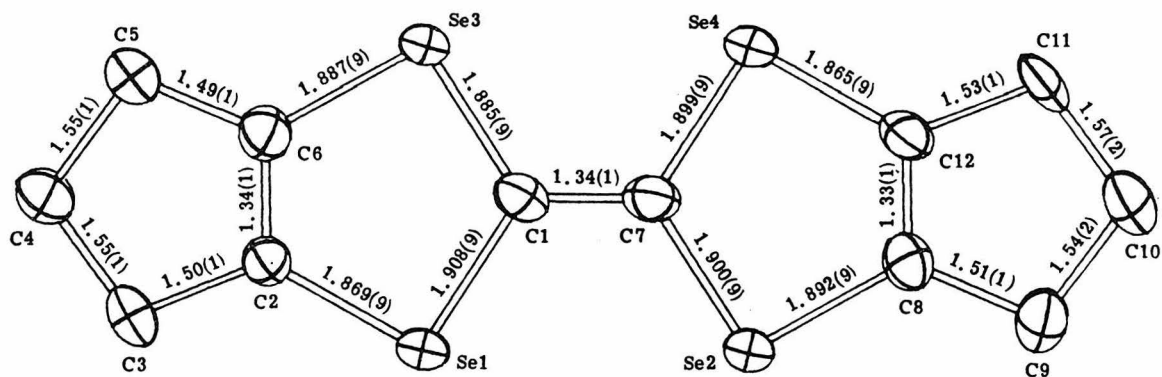
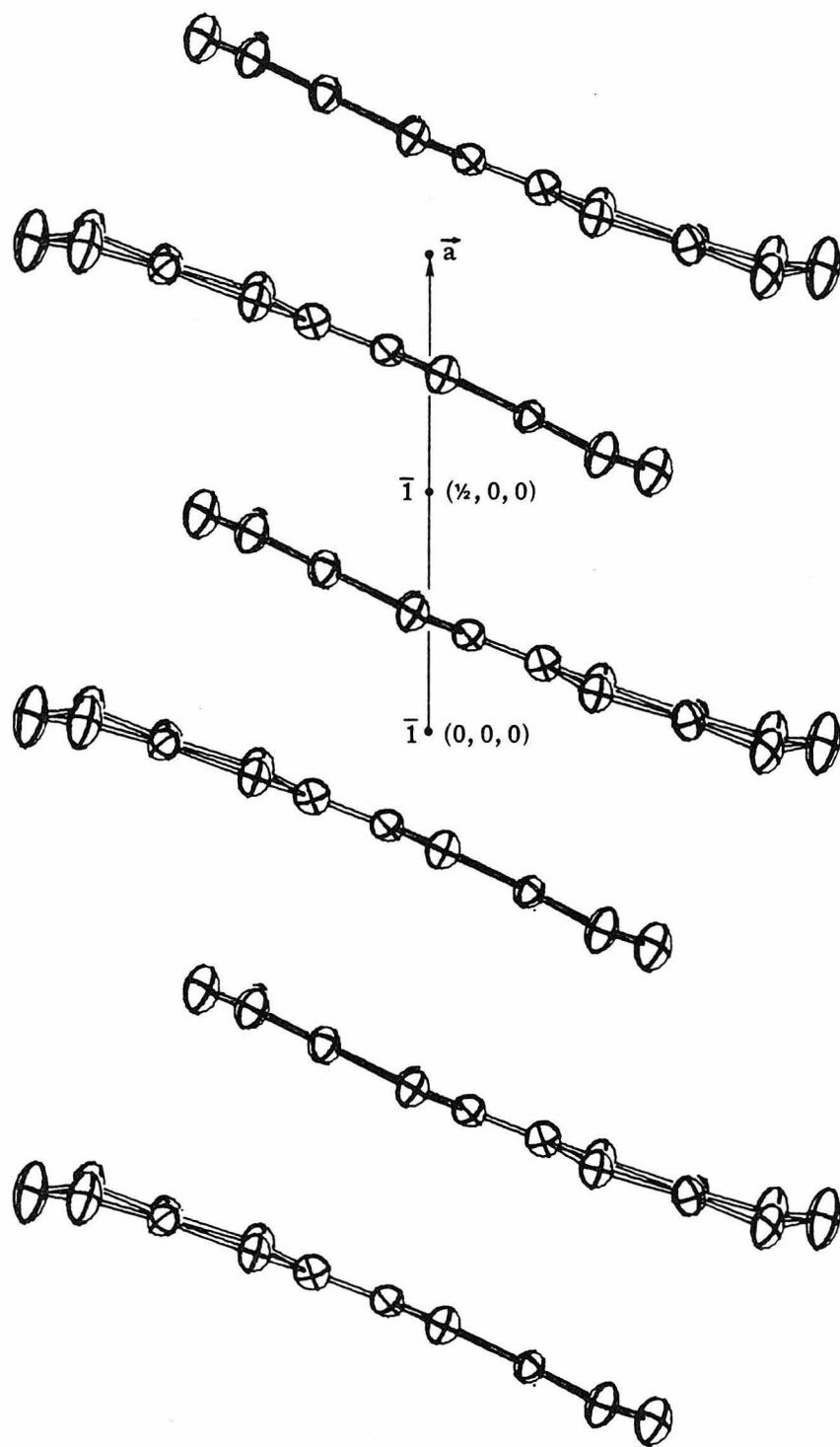
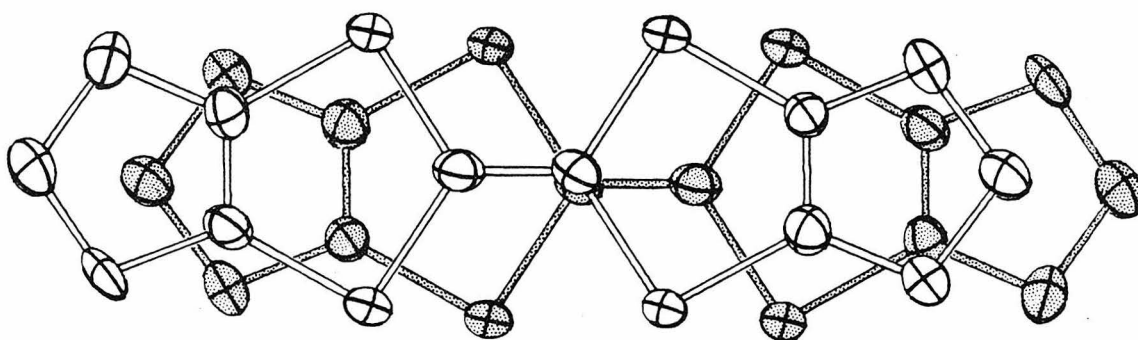


Figure 1

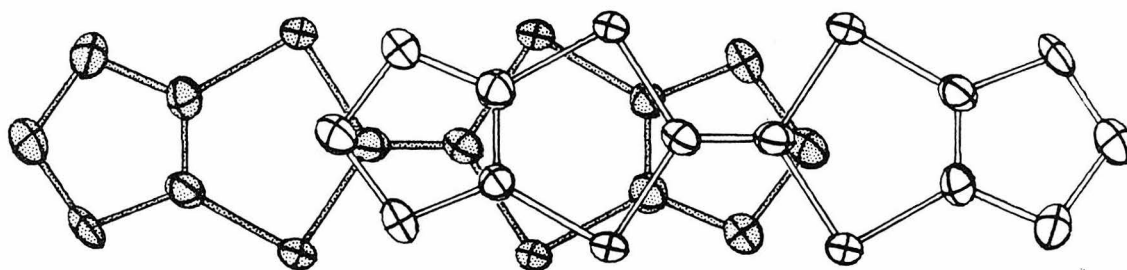


One cation stack along \vec{a} in HMTSF- I_x

Figure 2



Neighboring HMTSF molecules in the same cation stack viewed perpendicular to the molecular plane. The two molecules are related by the inversion center at $0, 0, 0$.



Neighboring HMTSF molecules in the same cation stack viewed perpendicular to the molecular planes. The two molecules are related by the inversion center at $\frac{1}{2}, 0, 0$.

Figure 3

neighboring HMTSF molecules, which are in the same cation stack and related by the inversion centers at 0,0,0 and 1/2,0,0, is shown in Figure 3. The amount of slippage of the HMTSF molecules related by the inversion center at 0,0,0 is the same as in TMTSF-Br_{0.82}, HMTSF-TCNQ, and other TMTSF, HMTSF, and related structures (for example, 9,10,11,12). However, the neighboring HMTSF molecules related by the inversion center at 1/2,0,0 show a different amount of slippage (See Figure 3).

The room temperature conductivity of HMTSF-I_x is $\sim 1.3\text{--}1.7 \Omega^{-1} \text{ cm}^{-1}$ (13). The temperature-dependence of the conductivity has not been measured. The apparent stoichiometry (1:0.157) certainly suggests that the band formed from overlapping HMTSF orbitals is only partially occupied and that HMTSF-I_x could be metallic. This is reasonable for a tight-binding model, but assumes that the HMTSF molecules are all evenly spaced. However, the HMTSF molecules are not required by the lattice symmetry to be evenly spaced. The results for three least-squares, best plane calculations are listed in Table IV. These results suggest that one cannot unequivocally state that the HMTSF molecules are, or are not, equally spaced. Also influencing the electronic properties of HMTSF-I_x is the presence of a random potential caused by the random occurrence of iodide and "solvent" at 1/2,1/2,1/2. To confirm that there was no superperiod and that the occurrence of iodide and "solvent" was, indeed, random, a very long exposure (92 hours) X-ray oscillation photograph was taken of HMTSF-I_x with nickel-filtered copper K_α radiation. The film was wrapped in aluminum foil to eliminate some

Table IVLeast-square, Best Plane Calculations^a and Interplanar Spacings for HMTSF-I_xPlane 1

Atoms	SD ^b	Deviations ^c
Se1	.001	-0.0374
Se2	.001	0.0292
Se3	.001	0.0352
Se4	.001	-0.0435
C1	.009	-0.0070
C7	.0091	0.0235

Direction cosines of the plane normal -0.9226, 0.5698, -0.1827

Interplanar spacings: $d_1 = 3.52$ $d_2 = 3.91 \text{ \AA}$ Plane 2

Atoms	SD ^b	Deviations ^c
Se1	.001	-0.0824
Se2	.001	-0.0194
Se3	.001	-0.0069
Se4	.001	-0.0892
C1	.009	-0.0516
C2	.009	0.0498
C6	.010	0.0842
C7	.0091	-0.0226
C8	.0094	0.0941
C12	.0092	0.0440

Direction cosines of the plane normal -0.9230, 0.5692, -0.1840

Interplanar spacings: $d_1 = 3.61$ $d_2 = 3.83 \text{ \AA}$ Plane 3

Atoms	SD ^b	Deviations ^c
Se1	.001	-0.1438
Se2	.001	-0.0745
Se3	.001	-0.0471
Se4	.001	-0.1231
C1	.009	-0.1005
C2	.009	-0.0085
C3	.010	-0.0913
C5	.010	0.1586
C6	.010	0.0348
C7	.0091	-0.0690
C8	.0094	0.0483
C9	.010	0.1770
C11	.010	0.0496
C12	.0092	0.0070

(Continued)

Table IV (Continued)Plane 3 (Continued)

Direction cosines of the plane normal -0.9220, 0.5724, -0.1879

Interplanar spacings: $d_1 = 3.70$ $d_2 = 3.72 \text{ \AA}$

^aV. Schomaker, J. Waser, R. E. Marsh and G. Bergman, Acta Cryst., 12, 600 (1959).

^bCoordinate standard deviations from full-matrix least-squares calculation.

^cDeviations, in \AA , from the calculated best plane. All atoms contributed to the best plane with weights of one.

of the longer wavelength fluorescence. There was no indication of any diffuse layer lines or of a supercell.

Figure 3 graphically shows that the overlap between adjacent HMTSF molecules is unique. What a HMTSF molecule "sees" above itself is not the same as what it "sees" underneath itself. This additional potential will cause additional band gaps, although not necessarily at the Fermi surface. Of course, in this HMTSF-I_x structure, since the stoichiometry is ill-defined, the energy of the Fermi surface is equally ill-defined. This structure is the first example of a new type of alternating cation overlap in quasi-one-dimensional conductors.

Because of the subtle interactions of the random potential caused by the disorder at 1/2,1/2,1/2, the band width, the amount of interchain coupling, and the potential from the alternating overlap patterns, a definitive statement about the cause of the apparently semiconducting behavior is difficult to make.

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goodness-of-fit = $[\sum w(|F_o|^2 - |F_c|^2)^2 / (N-P)]^{1/2}$
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Chapter 5

Summary

The three quasi-one-dimensional conductors studied here have shown a variety of interesting distortions and disorder. Understanding these distortions, disorder, and the overall structures of these materials is important for understanding their physical properties.

Prior to low-temperature structural studies of TTT_2I_3 it was not known whether or not the iodine atoms would order during the phase transitions or whether or not the TTT lattice would undergo a static distortion. For very slow cooling crystals of TTT_2I_3 (h.d.) do not appear to undergo a static distortion. Although all regions of reciprocal space cannot be conveniently sampled with a diffractometer, there was no indication of a major structural phase transition and the iodine chains never achieved long-range three-dimensional ordering. The lack of three-dimensional ordering can be rationalized by assuming the presence of iodides other than I_3^- . I_2 or I^- or other polyiodides will cause static disorder which will not, obviously, be eliminated at low temperatures. This is in contrast to $\text{TTFCl}_{.67}$ (1). Because the disorder of the chlorides in $\text{TTFCl}_{.67}$ is primarily dynamic disorder, as the temperature is lowered the chloride atoms gradually order and the TTF cations no longer "see" a random potential from the chlorides. With the decrease in the random, screening potential from the chlorides, electrons on the TTF molecules feel the effect of electrons on other

TTF molecules and couple. Coupling allows an overall decrease in the energy of the system. This electronic coupling gives rise to what is described as an electronically driven static Peierl's distortion (2,3,4). With the electronic coupling, a new band gap forms at the Fermi surface and the periodicity of the underlying lattice changes. That change in the lattice has been observed, crystallographically, in $\text{TTFCl}_{0.67}$ (1). That a similar effect is not observed in TTT_2I_3 (h.d.) suggests that the static disorder remaining in the iodide chains at low temperatures may cause a random potential sufficient to prevent strong electronic coupling (5). This is consistent with the observation that the phase transitions are most prominent in the crystals with the least disorder (Appendix A).

TTTI is an ordered, quasi-one-dimensional semiconductor which has not dimerized as would be expected for an exactly half-filled band. But since transport properties in crystals with two incommensurate, yet interacting, lattices are not understood, the interpretation of the graphs in Appendix B as being typical of a semiconductor could be incorrect (6).

HMTSF-I_x exhibits an interesting structure. There are primarily five features of the structure which should significantly affect its transport properties. (a) Although the stoichiometry could not be determined accurately from the X-ray data, there are indications that HMTSF-I_x is a non-stoichiometric complex. (b) There are no real "channels" in the structure. The random disorder of iodide and

"solvent" is localized about the coordinates $1/2, 1/2, 1/2$ in the unit cell. (c) The HMTSF molecules may or may not be evenly spaced.

(d) Each HMTSF molecule "sees" its two neighbors on either side in the same stack differently. (e) There is probably significant inter-chain coupling. The magnitude of the conductivity in HMTSF-I_x at room temperature suggests a semiconductor. This is not understood in light of its apparent non-stoichiometry and significant interchain coupling, but the other effects, (b) - (d), could be dominating and those three effects would favor semiconducting behavior.

These iodide-containing structures have shown three different types of iodide behavior in quasi-one-dimensional conductors. In TTT_2I_3 the iodide chains are at least as important, if not more important, than the TTT molecules in determining the observed physical properties. The retention of disorder in the iodide chains at low temperatures has important effects on the low-temperature transport properties. In TTTI the iodides are no longer disordered but are certainly still dominating the structure and causing a modulation of the TTT lattice. In HMTSF-I_x the iodide is probably of minor importance.

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Appendix A

PHYSICAL REVIEW B

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Effects of disorder on the transport properties of *bis*(tetrathiatetracene)triiodide

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Bis(tetrathiatetracene)triiodide [(TTT)₂I₃] is a quasi-one-dimensional organic metal consisting of segregated stacks of tetrathiatetracene (TTT) cation radicals and polyiodide chains. The TTT and iodine (I) sublattices are incommensurate with respect to each other, and the iodine lattice exhibits considerable disorder. In this paper, we report measurements of the conductivity and thermoelectric power of single crystals of (TTT)₂I₃ in which the degree of disorder is varied. The disorder is modified by crystallization processes and is characterized by x-ray studies. (TTT)₂I₃ exhibits metallic behavior at high temperatures (100°K < T < 300°K) even in the presence of considerable disorder. Below 100°K, (TTT)₂I₃ undergoes a broad metal-nonmetal transition. The effects of disorder on this transition and on the low-temperature transport properties are discussed. The present results are compared with data on (TTT)₂I₃ taken by other investigators.

INTRODUCTION

Quasi-one-dimensional organic metals have been the subject of intense interest during the past few years. This interest is due to the many exciting and unusual effects associated with the electronically driven instabilities possible in a one-dimensional electronic system. Most of the organic metals studied have been charge-transfer compounds such as tetrathiafulvalenium-tetracyanoquinodimethanide, TTF-TCNQ, and its derivatives.¹ These compounds have two types of charge carriers with electrons propagating along the acceptor (TCNQ) chain and holes along the donor (TTF) chain. They exhibit metallic behavior down to low temperatures (~60–100°K), but then undergo a metal-to-nonmetal transition that results in a non-metallic ground state at T = 0°K. The stabilization of the metallic state and an understanding of the roles of disorder, impurities, and interchain coupling on the metal-to-nonmetal transition have been important goals for many researchers in this field.¹

In this paper we describe the effects of disorder on the transport properties and metal-to-nonmetal transition in *bis*(tetrathiatetracene)triiodide, (TTT)₂I₃. This study is of particular interest for several reasons: (i) In contrast to TTF-TCNQ and its derivatives, (TTT)₂I₃ is a single-carrier system with holes propagating along the TTT cation radical stack.^{2–4} (ii) (TTT)₂I₃ has a high room-temperature conductivity and exhibits metallic behavior down to ~100°K where a broad metal-to-

nonmetal transition commences. Thus the transport properties are quite similar to those observed in the two-carrier systems. (iii) The iodine lattice in (TTT)₂I₃ exhibits disorder,⁵ and the degree of disorder can be varied somewhat systematically by crystallization processes,⁶ therefore, the effects of variable disorder in a given compound may be studied. (iv) Superconducting fluctuations at ~35°K have been reported⁷ in (TTT)₂I₃. It is suggested that the fluctuations are due to one-dimensional effects that become dominant when disorder disrupts the interchain coupling. Our results are compared with those found in other studies of (TTT)₂I₃.

SAMPLE PREPARATION AND CHARACTERIZATION

Tetrathiatetracene (TTT) (Fig. 1) was synthesized according to the procedures described by Perez-Albuerné.⁸ TTT was recrystallized twice from purified nitrobenzene and subsequently gradient sublimed three times in vacuum. Single crystals of (TTT)₂I₃ were obtained by recrystallization from nitrobenzene. TTT and iodine (I₂) were refluxed separately for approximately 1 h in purified nitrobenzene prior to mixing. Upon adding the iodine solution to the TTT solution, the resulting mixture was placed into a temperature bath at ~95°C. Depending upon the TTT concentration, the mole ratio X of I₂ to TTT, and the cooling rate, single crystals of (TTT)₂I₃ with various degrees of disorder in the iodine lattice could be obtained.

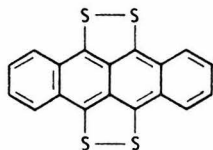


FIG. 1. Molecular structure of tetrathiatetracene (TTT).

Three different kinds of crystals were grown: (a) highly disordered (hd)—crystals with a highly disordered iodine lattice were grown using a TTT concentration of 3×10^{-3} moles/liter, a mole ratio, $X=1.0$, and a cooling rate of 3°C/h ; (b) medium disordered (md)—these crystals were grown with the same concentrations and mole ratio as in (a), but with a slightly faster cool-down rate; and (c) least disordered (ld)—crystals with the least-disordered iodine lattice were grown using a concentration of 5×10^{-4} moles/liter, $X=0.5$, and a cooling period of ~ 6 days.

The crystals were typically 3–6 mm long and exhibited a brilliant golden reflection. The crystal structures including the disorder of the iodine lattice were studied by x-ray diffraction. Oscillation and Weissenberg photographs were taken at room temperature and at approximately 120°K , and diffractometer data were obtained at various temperatures ranging from 300°K to $\sim 27^\circ\text{K}$ using a low-temperature diffractometer developed locally for this specific purpose.⁹ A description of some of the preliminary x-ray results is given in this paper. A detailed report on the crystallographic work will soon be reported.¹⁰

The room-temperature crystal structure, as determined from diffractometer studies, agrees with that reported by Smith and Luss⁵ and by Buravov *et al.*⁶ The TTT subcell is orthorhombic with space group $Cmca$, independent of the degree of disorder of the iodine lattice. Figure 2 shows a (010) projection of the crystal structure of $(\text{TTT})_2\text{I}_3$. The TTT molecules stack uniformly in a slipped configuration and form segregated chains lying parallel to the b axis (the axis of high conductivity). The iodide ions lie in channels situated between the TTT stacks, and also form chains parallel to the b axis. Figure 3 shows an oscillation photograph taken at room temperature of a highly-disordered (hd) crystal. It is seen that there are two different sets of layer lines. One set consists of sharp well defined spots (see lines 0, 2, and 4, Fig. 3) associated with the TTT lattice, while the other set is made up of diffuse lines (see 1 and 3, Fig. 3) associated with the iodine lattice. In particular, the diffuseness of the intense third-layer line clearly indicates the presence of disorder. Thus the iodine chains may be slipped relative to one another, along the direction

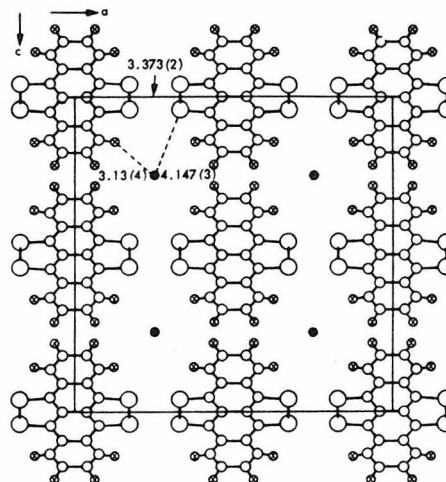


FIG. 2. (010) Projection of the crystal structure of $(\text{TTT})_2\text{I}_3$ (from Ref. 5). O—sulfur; ●—iodine; o—carbon; ⊙—hydrogen.

parallel to the stacking axis (b axis), such that there is a loss of horizontal registry in the iodine lattice. Oscillation photographs of a medium disordered crystal shows weak spots superimposed on the diffuse third-layer line. These spots are very faint and broad, but indicate that some ordering of the iodine lattice is present, e.g., the iodide chains may be more strongly coupled, resulting in the occurrence of some degree of horizontal registry. In the crystal designated as being least disordered (ld), sharper spots are observed on the diffuse third-layer line. Patterson maps calculated from diffractometer data using $(h3l)$ reflections

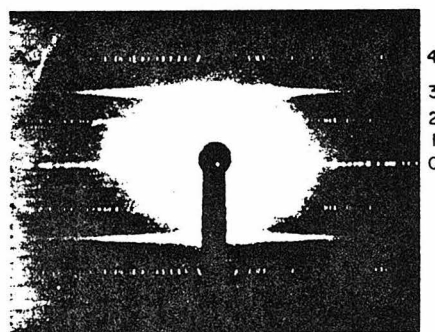


FIG. 3. X-ray oscillation photograph of a highly disordered crystal of $(\text{TTT})_2\text{I}_3$, taken at room temperature, using Ni filtered $\text{Cu K}\alpha$ radiation. The numbers label the layer lines.

alone from an ld crystal at room temperature and a hd crystal at 160 °K and 27 °K showed only contributions from the iodide chains. This indicates that the disorder is due solely to the iodide chains and there is little or no contribution to the third-layer line from the TTT subcell. We would like to point out that the ld crystals still exhibit considerable disorder in the iodine lattice. Thus it is believed that small changes in the degree of disorder are being observed.

The classification of the crystals into the three categories, namely, highly disordered, medium disordered, and least disordered, is arbitrary and based on the degree of diffuseness of the third-layer line observed in x-ray oscillation photographs. However, various crystals from the same batch, as well as other batches prepared under the same conditions, exhibit the same degree of diffuseness in the diffraction pattern, and hence we feel that the oscillation photographs provide a useful indicator to estimate the extent of disorder. These conclusions were confirmed by single-crystal diffractometer scans of peak profiles of third-layer spots. The peak profiles of third-layer spots of highly disordered $(TTT)_2I_3$ crystals are very weak, broad, and barely resolvable above the background at room temperature. For the least disordered crystal-

als, the peak profiles are sharper and well resolved on the third-layer line, indicating an increase in the order of the iodine lattice, in agreement with the data obtained from oscillation photographs.

EXPERIMENTAL RESULTS

The electrical conductivity σ along the needle b axis was measured by a standard four-probe technique using aquadag (graphite paint) for electrical contacts. Measurements were made using both dc and ac (16 Hz) techniques. The two types of measurements agreed quite well.

Figure 4 shows the temperature dependence of the normalized conductivity for crystals with different degrees of disorder. Values of the room-temperature conductivity are given in Table I. The conductivity increases with decreasing temperature, as expected for a metal. There is a broad maximum near 100 °K, below which the conductivity decreases with decreasing temperature. Similar temperature dependences for the conductivity have been observed in other quasi-one-dimensional systems.^{1,11} Both the peak value of the normalized conductivity σ_p/σ_{300} , and the temperature at which the peak value occurs, T_p° , decrease

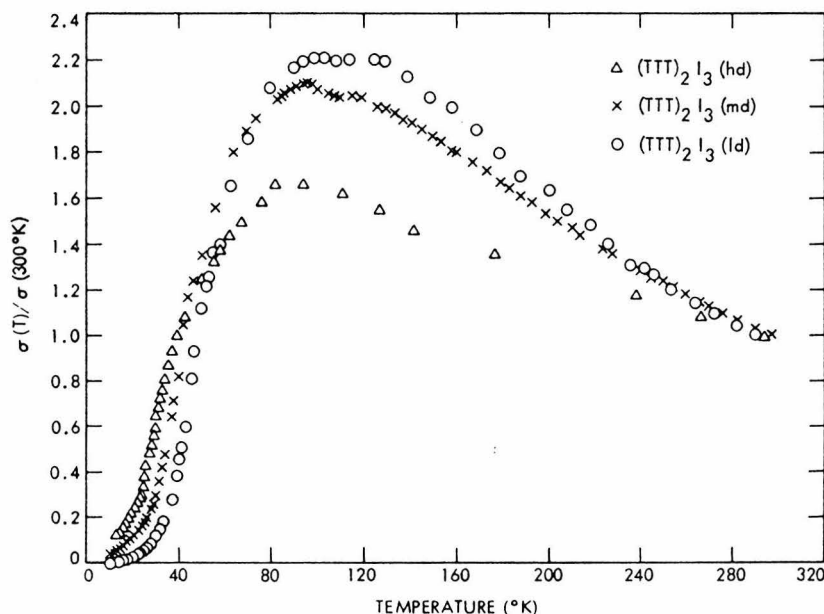


FIG. 4. Temperature dependence of the normalized conductivity of $(TTT)_2I_3$.

TABLE I. Transport parameters of $(\text{TTT})_2\text{I}_3$.

$(\text{TTT})_2\text{I}_3$	σ_{RT} ($\Omega^{-1}\text{cm}^{-1}$)	S_{RT} ($\mu\text{V}/^\circ\text{K}$)	$\sigma_p/\sigma_{\text{RT}}$	T_p^a ($^\circ\text{K}$)	T_{M-1}^a ($^\circ\text{K}$)
highly disordered	1000 ± 250	34 ± 0.2	1.66	82	24 ± 1
medium disordered	850 ± 250	40.7 ± 0.6	2.1	92	~ 30
least disordered	500 ± 250	42 ± 0.4	2.2	100	35 ± 1

^a Determined from $d \ln \rho / dT$.

with increasing disorder (see Table I). Figure 5 shows $\ln[\sigma(T)/\sigma(300^\circ\text{K})]$ versus reciprocal temperature $1/T$ for the three kinds of crystals. Note that crystals from the same batch agree quite well. The maximum rate of change in slope observed in ld crystals at $\sim 35^\circ\text{K}$ occurs at lower temperatures with increasing disorder. Measurements of the conductivity,^{2,3} ESR linewidth,³ and high-field magnetoresistance^{3,12} of hd crystals have attributed this change to the actual metal-to-nonmetal transition occurring at $T_{M-T} \sim 24^\circ\text{K}$. The other thing to note is that the magnitude of the conductivity at low temperatures e.g., $T \sim 13^\circ\text{K}$, increases with disorder.

The thermoelectric power was measured along the b axis by a technique described in detail elsewhere.¹³ The thermoelectric power (S) of several crystals with different degrees of disorder is shown in Fig. 6. The room-temperature values are small and positive for all the samples studied and are shown in Table I. As can be seen, the agreement between samples from the same batch is quite good, and the thermoelectric power (TEP) decreases with decreasing temperature down to $\sim 60^\circ\text{K}$ as expected for hole conduction along a metallic TTT chain. Note that the magnitude and temperature dependence of the TEP in the high-temperature range ($300^\circ\text{K} > T > 100^\circ\text{K}$) does not vary significantly with change in the degree of disorder. Below 60°K , the thermopower increases in a manner suggestive of nonmetallic behavior.

Another interesting feature is the variation of the peak value of the TEP at 21°K as a function of disorder. The peak TEP values scales with the degree of ordering and varies from $40 \pm 2 \mu\text{V}/^\circ\text{K}$ for the hd crystals to $255 \pm 15 \mu\text{V}/^\circ\text{K}$ for the ld samples (see Table II).

DISCUSSION

The crystals of $(\text{TTT})_2\text{I}_3$ grown under different crystallization conditions exhibit different degrees of disorder. A room-temperature crystal structure of the ld $(\text{TTT})_2\text{I}_3$ using third-level data indicates that the iodine is present in a polyiodide chain with less than full occupancy of all sites.¹⁰ Since there is no well-defined unit such as I_3^- , it seems unreasonable to determine stoichiometry

from repeat distances observed in oscillation photographs.

Temperature-dependent x-ray diffraction studies to 27°K of hd crystals have shown that the iodine chains do not completely order. An oscillation photograph at $120 \pm 10^\circ\text{K}$ of hd crystal was taken after rapidly cooling the crystal in a stream of cold N_2 gas. Beading along the diffuse third-layer line increased in intensity as compared to the room-temperature oscillation photograph. However, Weissenberg photograph ($h3l$) at $120 \pm 10^\circ\text{K}$ showed that there was a significant decrease in normal thermal motion, but the diffuse spots had the same size and shape as at room temperature, indicating residual disorder in the iodine chains and only short-range chain-chain coupling. Thus the apparent intensity gain of the diffuse spots is probably due to a loss of thermal motion, but also, perhaps, to some coalescing of the diffuse background.

Preliminary work on a low-temperature diffractometer⁹ to 27°K indicates that the iodine chains are disordered and minor structural changes may depend on the rate of cooling. Throughout the temperature range studied, the spots on the diffuse third-layer remain quite broad (in the 2θ and ω directions) as compared to normal Bragg reflections from the TTT lattice. The intensities of the broad spots do increase with decreasing temperature. There is no major structural change in the TTT lattice down to 27°K .

The effect of disorder on the conductivity and the TEP can be discussed best by considering three separate temperature ranges: high temperatures, $300^\circ\text{K} > T > 100^\circ\text{K}$, intermediate temperatures, $100^\circ\text{K} > T > 20^\circ\text{K}$, and low temperatures, $T < 20^\circ\text{K}$.

High-temperature region: $300^\circ\text{K} > T > 100^\circ\text{K}$

In this temperature range, the conductivity increases with decreasing temperature characteristic of a metal. The resistivity was fitted to

$$\rho(T) = \rho_0 + aT^\gamma, \quad (1)$$

with $\gamma = 2.20$ for crystals with the least disorder. γ decreases to 1.90 for the most highly disordered crystals. This is the trend one would expect since

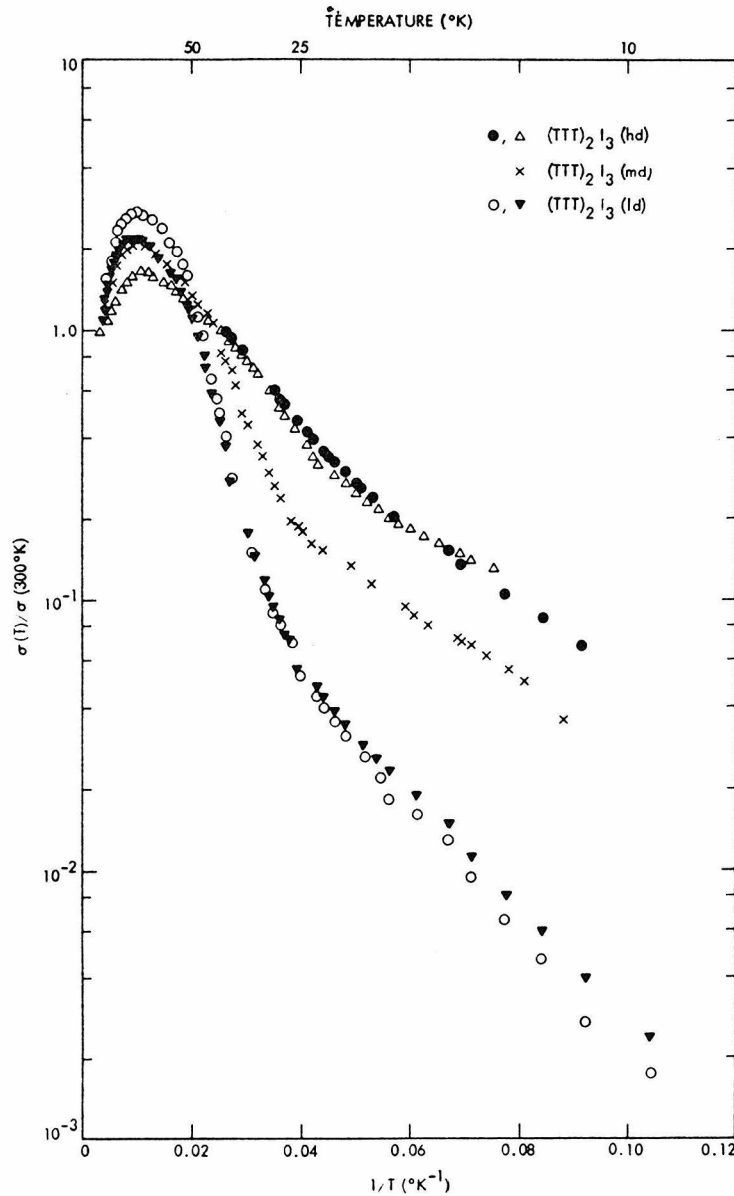
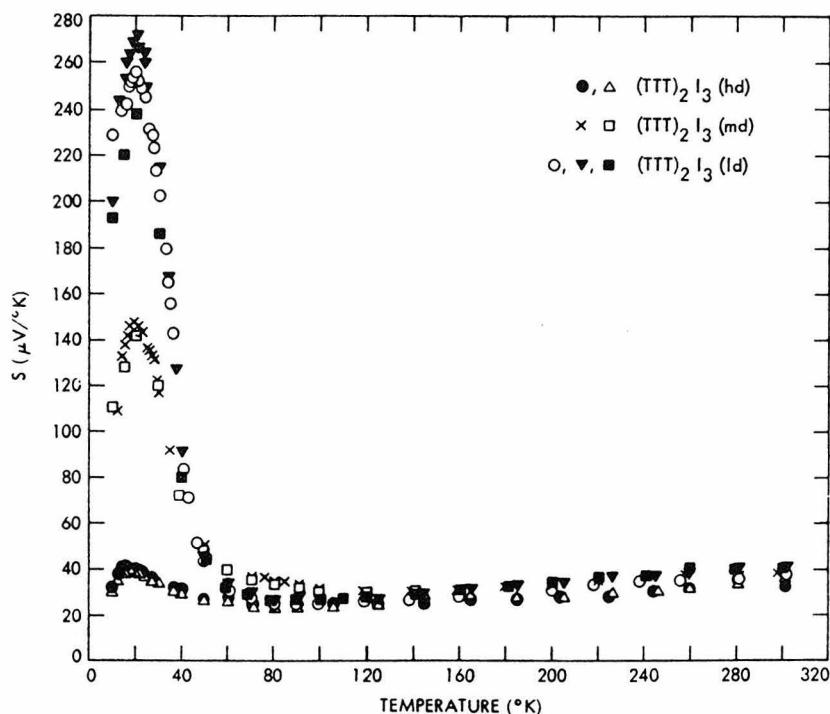


FIG. 5. $\ln[\sigma(T)/\sigma_{RT}]$ vs reciprocal temperature for $(TTT)_2I_3$.

increasing disorder should inhibit metallic transport. The metal-like temperature dependence plus the high value of the room-temperature conductivity suggests that electron transport occurs via de-

localized states and that the disorder is not large enough to cause localization at high temperatures. The peak value of the normalized conductivity decreases with increasing disorder. This is similar

FIG. 6. Thermoelectric power S of $(\text{TTT})_2\text{I}_3$.

to what has been observed in TTF-TCNQ, where the disorder was due to excessive thermal cycling¹⁴ or radiation damage.¹⁵ However, in contrast to TTF-TCNQ, the temperature of the conductivity

TABLE II. Comparison of peak TEP, S_p , and normalized low-temperature conductivities, $\sigma(13^\circ\text{K})/\sigma_{\text{RT}}$, of $(\text{TTT})_2\text{I}_3$.

$(\text{TTT})_2\text{I}_3$	S_p (21 °K) ($\mu\text{V}/^\circ\text{K}$)	$\sigma(13^\circ\text{K})/\sigma_{\text{RT}}$
JPL		
Least disordered	255	5×10^{-3}
Medium disordered	144	5×10^{-2}
Highly disordered	40	8×10^{-2}
Kodak	40	3.3×10^{-1}
Budapest ^a	20	~ 0.1
Chernogolovka (Cher.)		
Ordered ($T_p^\circ \sim 110^\circ\text{K}$)	270	... ^b
($T_p^\circ \sim 60^\circ\text{K}$)	110	$\sim 5 \times 10^{-2}$
Most disordered ($T_p^\circ \sim 35^\circ\text{K}$)	~ 30	... ^b

^a Unannealed.

^b Data unavailable.

maximum, T_p° , in $(\text{TTT})_2\text{I}_3$ does not increase with disorder but shifts to lower temperatures.

The TEP of $(\text{TTT})_2\text{I}_3$ at high temperatures may be described by a one-dimensional tight binding model. In this case, the thermoelectric power obtained from the Boltzmann equation is

$$S = -(\pi^2 k_B^2 T / 6 |e| t) [\cos \pi \frac{1}{2} \delta / (1 - \cos^2 \pi \frac{1}{2} \delta)], \quad (2)$$

where t is the transfer integral and δ is the number of conduction electrons per molecule. Although the TEP decreases linearly with decreasing temperature, characteristic of a metal, it does not extrapolate to zero at $T = 0^\circ\text{K}$. This indicates that energy-dependent scattering processes may be important. Nevertheless, the overall linear dependence of the TEP suggests that the room-temperature value and Eq. (2) may be used to obtain an approximate value of the electronic bandwidth, yielding $W = 4t = 0.7$ eV, assuming $\delta = \frac{1}{2}$. This value is consistent with the bandwidth obtained from magnetic susceptibility studies.² The temperature dependence of the TEP does not vary significantly with disorder throughout this tem-

perature range. This should be contrasted with the temperature dependence of the conductivity, which does change with increasing disorder. This is what one would expect since the thermoelectric power, unlike the conductivity, is a zero-current measurement and should be less sensitive to disorder.

In summary, $(\text{TTT})_2\text{I}_3$ is metallic throughout the temperature range $100^\circ\text{K} < T < 300^\circ\text{K}$, even in the presence of observable disorder. This suggests that the rms value of the disorder potential does not dominate the electronic bandwidth.

It is useful to compare the electronic properties of $(\text{TTT})_2\text{I}_3$ with those of the halides and pseudohalides of TTF.¹⁶ In both systems, electronic transport is by one carrier only; holes, which propagate along the cation radical chains. Also, there is disorder in the nonconducting anion chains in both types of compounds. While both systems have high room-temperature conductivities and exhibit metallic behavior at high temperatures, the TTF halides and pseudohalides undergo a metal-to-nonmetal transition at relatively high temperatures, i.e., $\sim 200^\circ\text{K}$, while $(\text{TTT})_2\text{I}_3$ remains metallic to below 100°K . We believe that the reason for this different behavior lies in the nature of the stacking of the cation radicals along the chains. In the TTF halides and pseudohalides, the TTF molecules stack in an eclipsed fashion which maximizes the wave-function overlap along the chains and leads to large electronic bandwidths of the order of 1–1.2 eV. In $(\text{TTT})_2\text{I}_3$, the TTT molecules are stacked in a slipped fashion which results in smaller bandwidths. Thus $(\text{TTT})_2\text{I}_3$ is another example of the empirical trend relating the stability of the metallic state and small electronic bandwidths.¹⁷ This indicates that the more stable metallic state observed in TTF-TCNQ and its derivatives is not critically dependent upon the presence of two different types of conducting chains, and that slipped stacking is favorable for obtaining a more stable metallic state, among other things. Naturally, the nature of the inter-chain coupling may vary considerably with the type of stacking, with slipped stacking being conducive for the development of significant transverse bandwidth.

Intermediate temperature range: $100^\circ\text{K} > T > 20^\circ\text{K}$

Below 100°K , the conductivity decreases with decreasing temperature, indicative of a broad transition from metallic to nonmetallic behavior. With increasing disorder, the change is further smeared out, with the conductivity falling less rapidly with decreasing temperature.

From previous ESR, high-field magnetoresis-

tance and conductivity studies^{2,3} of $\text{hd}(\text{TTT})_2\text{I}_3$, the metal-to-nonmetal transition temperature T_{M-I} (as determined from the temperature dependence of $d\ln\rho/dT$), is found to be at $\sim 24^\circ\text{K}$. Thus the region between T_P° and T_{M-I} represents a transition region between metallic and semiconducting states which has been broadened by the pronounced disorder. As the disorder diminishes, both T_P° and T_{M-I} increase with $T_{M-I} \sim 35^\circ\text{K}$ for the 1d crystals (see Table II).

The TEP of the least disordered sample attains a minimum value of $\sim 25 \mu\text{V}/^\circ\text{K}$ at $\sim 100^\circ\text{K}$, and then rises steeply to a value of $270 \mu\text{V}/^\circ\text{K}$ at 21°K . The TEP always reaches a maximum value at $\sim 21^\circ\text{K}$, regardless of the degree of disorder. However, the magnitude of the TEP at this temperature decreases with increasing disorder. This, again, indicates that disorder is smearing out the transition and creating states in the single-particle gap.

The TEP in the semiconducting region is very sensitive to the degree of disorder, in contrast to its behavior in the metallic state. Thus the magnitude of the peak value of the TEP at $\sim 21^\circ\text{K}$ may be used as a rough indicator of the degree of disorder in $(\text{TTT})_2\text{I}_3$ compounds. Similar relationships between the magnitude of the peak value of the TEP at low temperatures and the presence of disorder can be found in other quasi-one-dimensional organic metals.¹⁸

Low-temperature range: $T < 20^\circ\text{K}$

As seen in Fig. 5, $(\text{TTT})_2\text{I}_3$ exhibits a change of slope in the $\ln\sigma/\sigma_{RT}$ vs $1/T$ plot at about $\sim 24^\circ\text{K}$ in 1d crystals. With increasing disorder, the slope in the low-temperature side decreases significantly. Also, the change in slope is less evident in the hd crystals. We believe that these facts indicate that below $\sim 24^\circ\text{K}$, $(\text{TTT})_2\text{I}_3$ is a disordered semiconductor. Conductivity studies to very low temperatures ($\sim 100 \text{ mK}$) are in progress in order to further elucidate the conduction processes in the nonmetallic state. As the disorder increases, states are formed in the gap resulting in a reduced effective gap and an increased conductivity at low temperatures. Since the value of the conductivity at low temperatures, e.g., 13°K , appears to vary considerably with disorder, we feel that it (as well as the peak TEP) may serve as a rough indicator of the degree of disorder in the crystal (see Table II).

$(\text{TTT})_2\text{I}_3$ has been the subject of several previous studies.²⁻⁴ Since the results of Kaminski *et al.*⁴ (Chernogolovka, Lab) are quite unusual and different from the results of Isett *et al.*² (Kodal Lab), Mihaly *et al.*⁴ (Budapest Lab), Somoano *et al.*³ [Jet

Propulsion Laboratory (JPL)], and those discussed in this paper, it is useful to make a more detailed comparison. It has been shown that $(\text{TTT})_2\text{I}_3$ can form crystals with varying degree of disorder in the iodine lattice. Comparing the properties of samples prepared by different groups, therefore, involves an assignment of the relative amount of disorder. As has been suggested earlier in this paper, rough indicators of the disorder are the magnitudes of the low-temperature conductivity and the peak value of the TEP at $\sim 21^\circ\text{K}$.

The values of the normalized conductivity at $\sim 13^\circ\text{K}$ and the peak values of the TEP for samples from various laboratories are shown in Table II. These two indicators suggest that the Budapest samples are the most disordered. The Kodak samples are similar to our hd samples. X-ray data on the Kodak samples⁵ support this conclusion, since no diffuse spots are seen on the third-layer lines of oscillation photographs of these crystals. In addition, derivative analysis² of the Kodak conductivity data reveals a transition at $T_{M-I} \sim 21^\circ\text{K}$.

The degree of ordering of the Chernogolovka samples is obtained from the work of Kaminski *et al.*,⁶ and the samples are labeled by their values of T_p° . Comparison with these samples is difficult since the values of $\sigma(13^\circ\text{K})/\sigma_{RT}$ are known for only one sample. Nevertheless, a comparison does offer some insight. The most ordered sample from the Chernogolovka Lab gives a peak TEP value in good agreement with our ld sample. As the disorder increases, the conductivity peaks of the Chernogolovka samples sharpen up and both T_p° and T_{M-I} decrease. In addition, the conductivity continues to increase in a metallic fashion with increasing disorder to yield high values of σ_p/σ_{RT} . This behavior is quite different from that found in the JPL, Kodak, and Budapest samples. The dominant effect of disorder in the latter samples is to smear the transition from the metallic to the semiconducting state and to decrease T_p° and T_{M-I} . Disorder inhibits metallic transport but enhances the low-temperature conductivity in the semiconducting state by providing states near the Fermi level in the single-particle gap. It is important to stress that x-ray studies of the JPL samples at 27°K indicate that structural disorder is still present. The dominant effect of disorder in the Chernogolovka samples is to decrease T_p° and T_{M-I} only.

The reason for the difference between the properties of the Chernogolovka samples and those from other laboratories is presently unknown. However, we believe that a significant factor is the tendency of the iodine lattice to order at low temperatures in the Chernogolovka samples. This effect was observed in recent x-ray diffuse scattering studies of the highly disordered Chernogolovka

samples by Comés.¹⁹ It is not known whether the ordering of the iodine lattice occurs gradually or discontinuously, and the actual degree of ordering is unknown. Nevertheless, electron scattering by disorder should decrease at low temperatures, thus allowing large values of the conductivity (and, hence, σ_p/σ_{RT}) to be attained prior to T_{M-I} . An obvious question associated with this point is: why doesn't the iodine lattice of the JPL, Kodak, and Budapest samples show signs of significant ordering? At present, the answer is unknown. Slight differences in the exact nature of the disorder, the presence of impurities, subtle differences in the interchain coupling, and differences in the amount of excess iodine between the Chernogolovka Lab samples and ours could contribute to differences in the tendency to order. This occurrence of low-temperature ordering is at variance with the model suggested by Abrahams *et al.*⁷ They suggest that the disorder disrupts the interchain coupling making the system more one dimensional such that superconducting fluctuations at $\sim 35^\circ\text{K}$ occur. This suggestion may be premature. It should be stressed that the occurrence of ordering in the iodine lattice at low temperatures does not provide a single consistent explanation of the differences observed in $(\text{TTT})_2\text{I}_3$ in different laboratories. However, we feel that this feature will play a prominent role in any eventual model that consistently explains the overall behavior of $(\text{TTT})_2\text{I}_3$. Obviously, additional low-temperature structural studies would be quite valuable.

CONCLUSIONS

Crystals of $(\text{TTT})_2\text{I}_3$ have been grown using different crystallization parameters to yield samples with varying degrees of disorder in the iodine lattice. $(\text{TTT})_2\text{I}_3$ is metallic at high temperatures ($T > 100^\circ\text{K}$), even in the presence of observable disorder. A metal-to-nonmetal transition occurs at $T_{M-I} \sim 24\text{--}35^\circ\text{K}$. The effects of disorder are to: (i) reduce the maximum in the conductivity at high temperatures, (ii) broaden the metal-to-nonmetal transition over a large temperature range, (iii) decrease the metal-to-nonmetal transition temperature T_{M-I} , and (iv) enhance the conductivity in the nonmetallic state. In particular, the low-temperature conductivity and thermoelectric power are rough indicators of the degree of disorder. A comparison of measurements made on $(\text{TTT})_2\text{I}_3$ at different laboratories indicates that tendencies for the iodine lattice to order at low temperatures may lead to the very different behavior observed in the conductivity by different researchers.

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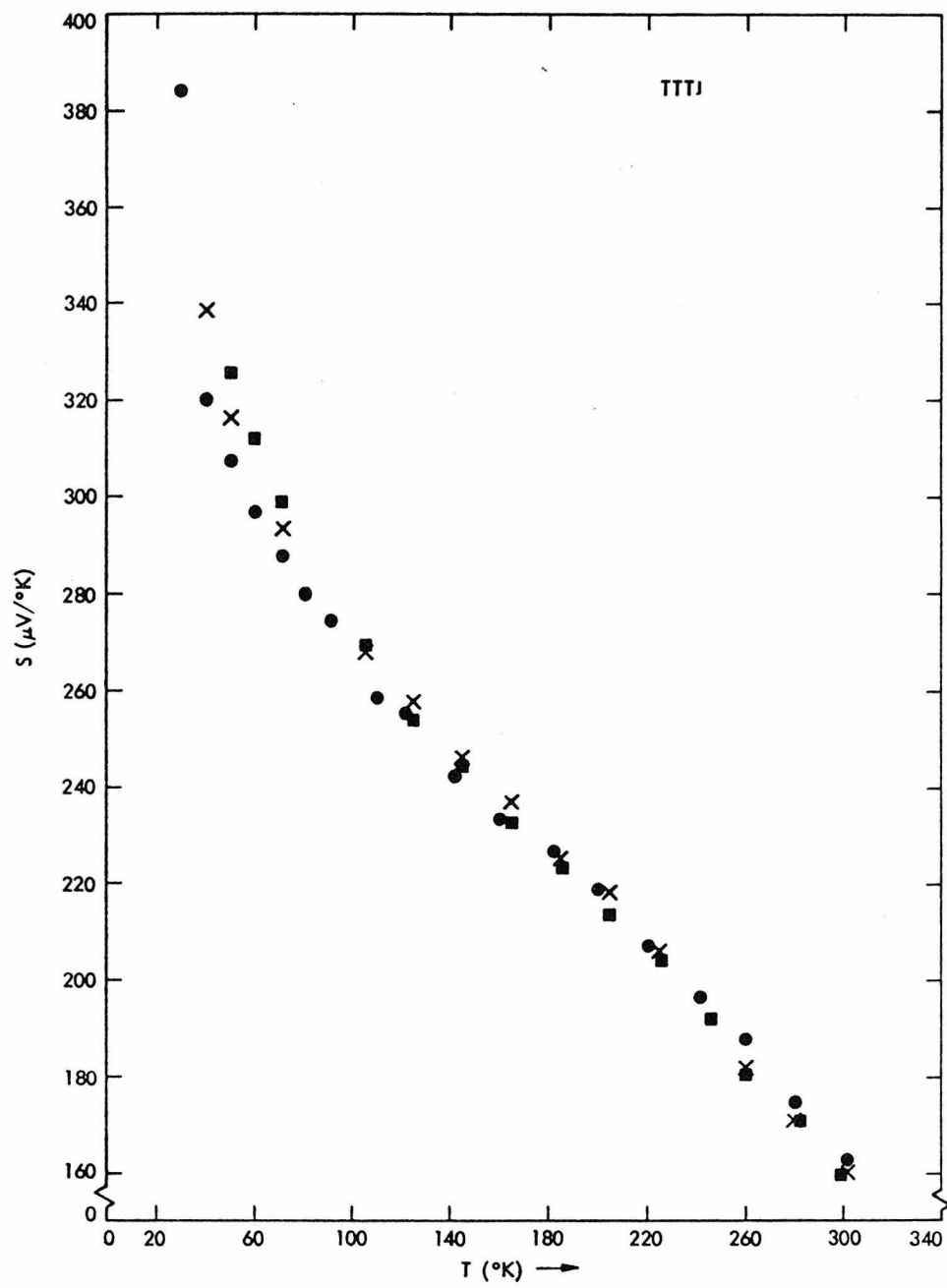
Technology, sponsored by the National Aeronautics and Space Administration, Contract NAS7-100. A portion of the x-ray crystallographic studies, especially the development of the low-temperature x-ray goniometer, was sponsored by the NSF under Grant No. DMR74-19029A1 (Contribution No. 5958). P. M. C., an A.P. Sloan Foundation Fellow, acknowledges research support by the NSF under Grant No. DMR76-83421.

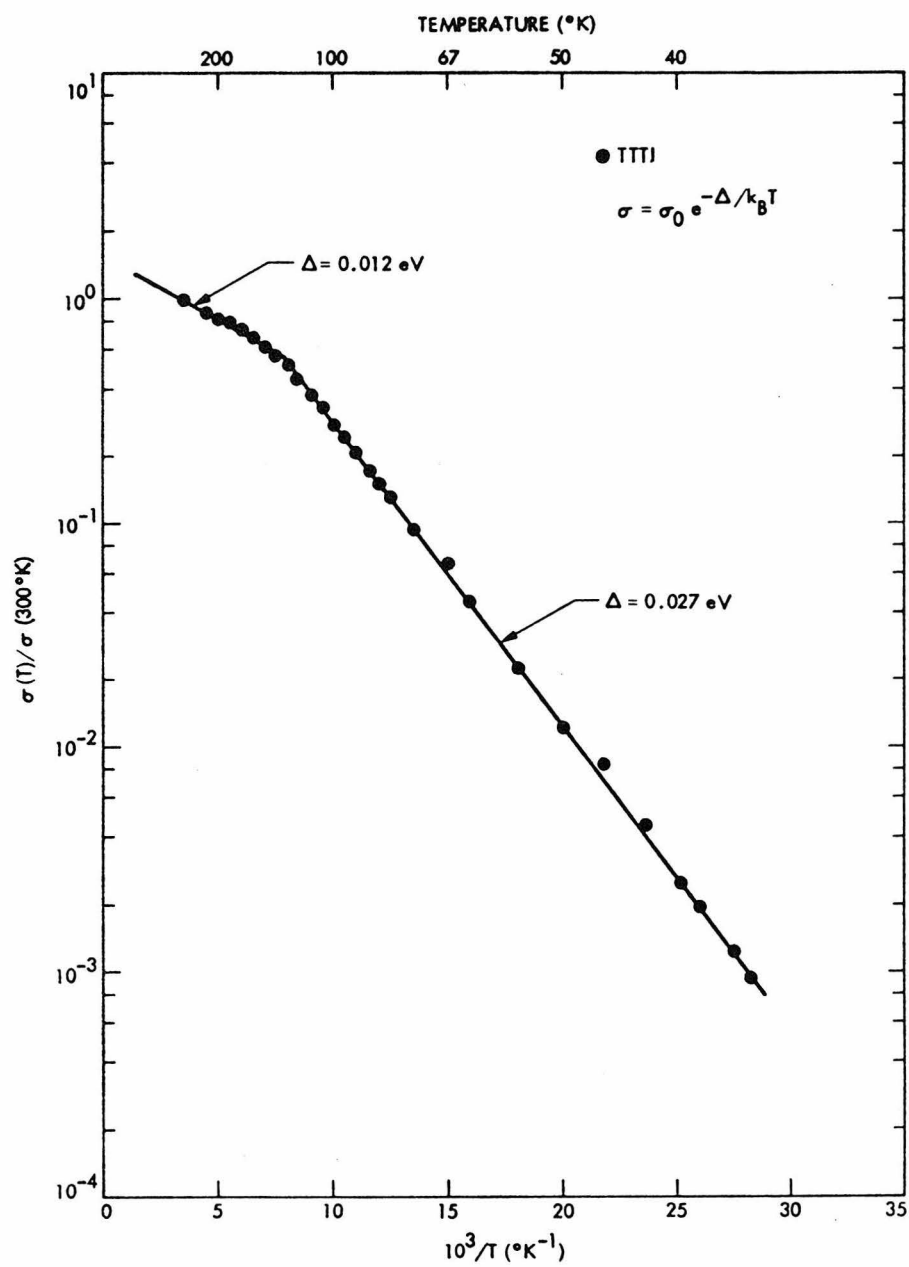
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Appendix B

Conductivity and Thermoelectric Power Measurements of
(Tetrathiotetracene)(Iodide)

Two graphs are shown for the conductivity and thermoelectric power of TTTI as functions of temperature. The measurements were made by R. B. Somoano and S. K. Khanna at the Jet Propulsion Laboratory.





Appendix C

Observed and Calculated Structure Factors

The following pages list the observed and calculated structure factors for the room-temperature, 164° K, 74° K, and 19° K (Cmca) structures of $\text{TTT}_{2/3}\text{I}_3$ (h.d.), for TTTI , and for HMTSF-I_x . For each hkl the numbers from left to right are $10|F_o|$, $10|F_c|$, and $10(F_o^2 - F_c^2)/\sigma(F_o^2)$.

TTT213 (H.D.)				ROOM TEMP.			COPPER			PAGE	1
0 0 L				8 0 L			8 672 655 19				
				0 332 287 97			10 664 640 26				
2	382	217	370	2	2784	3010	-118	12	321	327	-15
4	2850	2968	-50	4	1413	1333	72	14	314	331	-48
6	1705	1546	76	6	1766	1763	2	16	259	293	-119
8	1340	1207	127	8	1662	1636	21	18 0 L			
10	2337	2365	-14	10	629	542	143				
12	501	445	103	12	1527	1464	53	0	250	268	-46
14	1654	1671	-14	14	376	363	27	2	946	943	4
16	398	400	-3	16	776	782	-9	+	247	237	26
18	698	699	-1	18	301	302	-2	6	539	534	10
20	428	429	-2	20	252	272	-52	8	621	640	-36
22	291	287	10	10 0 L			10 172 173 -2				
2 0 L				0 2289 2492 -113			12 567 573 -12				
				2 1168 1149 21			14 68 92 -74				
0	1469	1074	213	4	1526	1505	17	20 0 L			
2	2062	2117	-27	6	1315	1210	99	0	168	160	21
4	3146	3560	-96	8	1042	946	111	2	874	868	8
6	2414	2375	13	10	982	918	76	+	187	187	0
8	1821	1716	77	12	1030	970	69	6	449	449	0
10	2091	2067	14	14	802	769	46	8	451	456	-11
12	518	469	97	16	+27	+11	33	10	78	31	77
14	1183	1192	-9	18	370	380	-23	22 0 L			
16	462	459	6	20	281	309	-81	0	262	265	-10
18	556	567	-20	12 0 L			2 334 333 2				
20	381	392	-24	0	1946	2116	-116	+	280	278	6
22	474	513	-95	2	385	314	156	6	285	283	6
4 0 L				4	1642	1691	-41	1 1 L			
0	2108	2043	29	6	1029	1010	23	1	295	324	-59
2	2838	3272	-177	8	801	737	90	2	338	336	3
4	1935	1927	3	10	1829	1751	58	3	1521	1533	-7
6	1999	1959	25	12	+75	393	1+1	4	210+	21+2	-17
8	1736	1689	36	14	1024	958	76	5	1172	1136	28
10	703	657	77	16	228	244	-40	6	914	912	2
12	1964	1948	10	18	422	460	-92	7	143	137	17
14	699	679	32	14 0 L			8 167 175 -24				
16	777	779	-3	0	1601	1706	-92	9	238	242	-12
18	414	417	-6	2	+35	399	79	10	+37	437	0
20	299	319	-49	4	1289	1321	-33	11	173	172	2
22	221	236	-42	6	764	773	-14	12	35	9	12
6 0 L				8	558	537	38	13	226	223	7
0	1541	1452	53	10	620	594	44	14	-24	10	-5
2	3162	3400	-84	12	644	656	-20	15	150	150	0
4	1002	897	130	14	768	755	19	16	10+	85	27
6	1890	1857	2+	16	135	166	-87	17	150	150	0
8	2051	2027	14	18	276	304	-91	18	453	453	0
10	1092	997	106	16 0 L			19 240 246 -14				
12	1359	1323	35	0	552	534	32	20	210	220	-23
14	258	223	72	2	846	870	-36	21	21	+3	-14
16	995	1002	-8	+	687	693	-10	22	13	5	1
18	444	428	33	6	663	658	8	23	-12	15	-6
20	313	331	-44				3 1 L				
22	194	212	-48								

TTT213 (H.D.)				ROOM TEMP.			COPPER		PAGE		2
1	1093	1154	-52	10	111	93	26		13	1	L
2	709	790	-101	11	162	165	-6				
3	382	438	-105	12	88	63	24	1	78	80	-2
4	360	427	-133	13	111	74	42	2	112	109	5
5	137	115	57	14	-31	13	-8	3	64	83	-24
6	1302	1360	-61	15	133	122	17	4	45	51	-4
7	477	437	89	16	52	65	-10	5	56	63	-6
8	1276	1258	17	17	100	107	-9	6	442	455	-27
9	462	474	-28	18	-13	30	-8	7	427	435	-17
10	449	427	50	19	88	103	-24	8	372	376	-8
11	444	446	-4	20	9	9	0	9	140	156	-30
12	206	202	10	21	39	2	20	10	49	54	-3
13	392	389	7					11	61	26	22
14	185	176	18		9	1	L	12	28	18	3
15	414	402	26					13	70	67	3
16	407	405	4	1	614	659	-87	14	142	128	26
17	226	237	-20	2	452	474	-50	15	296	297	-2
18	112	105	10	3	776	782	-9	16	290	302	-31
19	108	110	-3	4	1193	1206	-8	17	42	54	-15
20	170	161	20	5	413	417	-8	18	-17	17	-10
21	82	75	11	6	844	869	-36				
22	1+8	157	-23	7	42	3	13		15	1	L
				8	593	583	18				
				9	348	316	67	1	403	416	-30
	5	1	L	10	424	422	4	2	265	283	-47
				11	295	308	-28	3	2+0	257	-43
1	735	726	9	12	144	139	7	4	480	493	-28
2	319	346	-40	13	428	403	48	5	266	247	45
3	76	125	-92	14	116	110	8	6	349	352	-7
4	471	522	-123	15	126	135	-14	7	103	123	-36
5	161	141	51	16	153	144	16	8	215	214	2
6	202	243	-128	17	56	21	21	9	36	51	-11
7	351	351	0	18	366	360	13	10	315	294	48
8	172	208	-104	19	133	137	-8	11	35+	323	71
9	60	3	34	20	203	217	-40	12	24	28	-1
10	407	391	36	21	24	21	2	13	80	80	0
11	471	429	87					14	56	+3	16
12	57	53	3		11	1	L	15	59	31	34
13	113	133	-34					16	35	27	8
14	75	56	19	1	428	459	-7+	17	46	53	-14
15	25	18	2	2	278	305	-60				
16	40	34	3	3	299	311	-31		17	1	L
17	-41	+	-12	4	282	306	-60				
18	120	103	27	5	87	81	6	1	302	307	-13
19	97	97	0	6	717	724	-11	2	169	172	-7
20	128	125	5	7	278	302	-52	3	114	123	-17
21	24	46	-16	8	819	774	62	4	211	217	-15
22	53	52	1	9	432	388	83	5	57	31	22
				10	223	207	28	6	372	367	12
	7	1	L	11	90	120	-34	7	126	112	28
				12	164	145	28	8	320	315	12
1	309	315	-15	13	367	316	97	9	75	77	-3
2	92	118	-39	14	190	176	23	10	219	207	32
3	196	204	-18	15	302	286	34	11	241	221	56
4	147	145	4	16	308	290	39	12	64	59	7
5	251	228	32	17	245	230	35	13	127	137	-30
6	75	63	15	18	126	124	+	14	38	27	13
7	298	277	52	19	30	9	13	15	157	167	-38
8	159	172	-30	20	48	50	-3				
9	73	55	19						19	1	L

TTT2I3 (H.D.)				ROOM TEMP.		COPPER		PAGE		3	
				6	367	373	-17	15	203	182	34
1	34	14	12	7	317	331	-42	16	177	164	19
2	40	17	16	8	928	887	56	17	208	195	24
3	75	92	-31	9	170	145	65	18	50	76	-22
4	60	66	-9	10	280	264	46	19	185	176	19
5	87	90	-5	11	257	249	22	20	176	178	-4
6	105	106	-2	12	99	65	51	21	+6	48	-2
7	106	99	16	13	120	124	-7				
8	38	44	-6	14	68	60	8		8	2	L
9	-18	11	-6	15	153	158	-11				
10	40	40	0	16	176	156	45	0	78	75	3
11	25	9	10	17	223	213	25	1	382	397	-27
12	11	11	0	18	214	196	47	2	572	615	-63
				19	177	175	5	3	414	419	-8
	21	1	L	20	177	173	11	4	101	96	5
				21	111	115	-10	5	697	715	-22
1	199	198	3	22	22	57	-45	6	295	304	-16
2	139	143	-12					7	627	641	-18
3	201	194	23		4	2	L	8	588	574	19
4	282	277	15					9	759	754	5
5	-27	22	-20	0	444	456	-27	10	83	74	7
6	344	339	14	1	539	508	60	11	128	118	11
7	90	99	-28	2	904	977	-108	12	321	289	52
8	308	305	9	3	556	568	-26	13	82	37	-4
9	160	171	-43	4	115	48	71	14	113	144	-44
				5	699	699	0	15	300	306	-11
	0	2	L	6	307	329	-48	16	279	269	18
				7	140	132	14	17	149	155	-9
0	1064	1179	-150	8	282	278	8	18	134	120	24
1	321	307	38	9	381	361	35	19	194	194	0
2	238	224	40	10	254	238	29	20	86	87	-1
3	1202	1137	71	11	16	25	-2				
4	927	885	57	12	366	336	52		10	2	L
5	880	864	23	13	59	60	0				
6	364	343	53	14	194	185	15	0	463	494	-52
7	1742	1683	28	15	118	113	6	1	141	139	3
8	606	535	129	16	241	216	46	2	373	383	-18
9	1236	1192	45	17	101	99	2	3	89	70	19
10	653	634	33	18	151	127	38	4	173	176	-4
11	448	454	-13	19	233	234	-2	5	-43	19	-11
12	126	116	20	20	46	45	0	6	255	233	33
13	449	440	18	21	91	97	-10	7	446	420	39
14	865	855	14					8	78	38	21
15	474	482	-18		6	2	L	9	166	162	5
16	104	134	-56					10	333	309	38
17	358	361	-7	0	164	68	116	11	108	103	4
18	112	82	51	1	222	234	-22	12	127	125	2
19	187	174	35	2	497	517	-30	13	152	143	11
20	33	32	0	3	207	240	-63	14	317	313	7
21	222	231	-28	4	479	491	-19	15	108	110	-2
22	27	37	-12	5	971	1031	-59	16	88	48	36
				6	516	531	-23	17	87	91	-5
	2	2	L	7	643	656	-17	18	108	107	1
				8	139	141	-3	19	54	52	2
0	597	578	33	9	488	480	12				
1	43	0	19	10	154	137	23		12	2	L
2	86	5	88	11	163	155	11				
3	810	824	-22	12	420	402	28	0	451	476	-43
4	672	683	-21	13	164	176	-18	1	244	254	-20
5	594	583	22	14	108	124	-18	2	285	295	-19

TTT213 (H.O.)				ROOM TEMP.		COPPER		PAGE			4
3	920	956	-35	1	140	128	23	10	244	258	-25
4	692	713	-26	2	261	245	36	11	250	240	17
5	691	704	-16	3	190	178	25	12	163	167	-6
6	212	213	-1	4	80	67	18	13	349	334	26
7	633	621	15	5	402	375	50	14	198	201	-5
8	197	189	12	6	176	159	39	15	345	332	23
9	379	357	35	7	255	242	30	16	34	32	0
10	284	248	58	8	63	23	41	17	150	150	0
11	280	251	46	9	218	199	46	18	110	115	-7
12	39	28	4	10	50	46	5	19	-21	35	-15
13	90	91	-1	11	61	55	9	20	-19	9	-5
14	214	196	33	12	142	145	-9				
15	77	85	-9						5	3	L
16	39	11	14		20	2	L				
17	261	259	4					1	119	156	-38
18	22	34	-10	0	43	28	14	2	404	401	4
				1	199	181	21	3	319	349	-50
	14	2	L	2	273	253	47	4	108	110	-2
				3	294	285	21	5	193	205	-18
0	384	404	-36	4	74	74	0	6	65	72	-5
1	115	110	7	5	404	380	45	7	76	70	4
2	282	291	-16	6	103	86	38	8	183	172	15
3	135	124	16	7	305	281	56	9	168	154	18
4	66	27	25	8	154	149	15	10	34	74	-21
5	282	275	13	9	273	287	-33	11	205	186	28
6	97	94	3					12	-49	19	-14
7	566	545	30		1	3	L	13	203	200	5
8	127	115	16					14	93	104	-12
9	258	230	53	1	406	428	-40	15	212	208	7
10	236	230	12	2	167	165	3	16	62	52	7
11	215	193	40	3	380	358	40	17	76	88	-13
12	66	52	12	4	18	15	0	18	-18	15	-4
13	45	18	15	5	109	144	40	19	33	33	0
14	214	194	44	6	295	277	35				
15	107	99	15	7	-25	4	-4		7	3	L
16	25	25	0	8	29	37	-3				
17	80	121	-96	9	416	412	6	1	43	36	2
				10	372	377	-8	2	207	202	7
	16	2	L	11	841	825	16	3	165	147	18
				12	244	226	31	4	55	45	4
0	115	101	21	13	507	505	3	5	73	74	0
1	33	26	3	14	193	197	-7	6	72	36	19
2	105	89	22	15	149	145	6	7	33	47	-5
3	135	141	-10	16	174	180	-10	8	102	114	-11
4	225	218	14	17	50	97	-43	9	-41	14	-8
5	87	82	6	18	-18	16	-4	10	140	156	-20
6	146	139	12	19	26	8	6	11	101	135	-52
7	45	34	6	20	30	7	10	12	127	110	18
8	225	216	19					13	26	5	3
9	95	89	8		3	3	L	14	110	146	-45
10	61	44	16					15	80	91	-11
11	44	37	5	1	1175	1212	-28	16	-27	26	-10
12	125	121	8	2	96	106	-10	17	50	31	12
13	41	22	16	3	676	714	-49	18	27	1	7
14	31	10	14	4	312	308	7	19	-20	5	-6
15	80	96	-40	5	302	303	-1				
				6	194	201	-12		9	3	L
	18	2	L	7	100	95	5				
				8	192	196	-6	1	628	652	-31
0	46	48	-1	9	290	264	44	2	160	171	-15

TTT213 (H.D.)				ROOM TEMP.		COPPER		PAGE		5	
3	540	563	-33	4	105	92	18	2	162	167	-6
4	135	155	-22	5	167	152	28	3	178	181	-4
5	258	245	18	6	86	78	10	4	301	273	47
6	214	216	-2	7	47	64	-15	5	86	57	19
7	61	55	2	8	101	91	14	6	409	400	13
8	81	71	6	9	208	196	26	7	-9	12	-1
9	409	406	4	10	11	13	0	8	52	61	-5
10	60	100	-28	11	334	307	55	9	194	188	9
11	625	623	2	12	7	1	0	10	197	199	-3
12	48	58	-5	13	251	250	2	11	103	109	-6
13	510	500	14					12	138	123	20
14	22	28	-1		17	3	L	13	-35	37	-15
15	263	250	25					14	129	132	-4
16	142	138	7	1	367	358	18	15	97	103	-8
17	133	154	-44	2	21	27	-2	16	93	111	-27
18	2+	46	-21	3	279	267	27	17	43	61	-19
				4	119	113	11				
	11	3	L	5	145	136	19		4	4	L
				6	99	87	23				
1	787	823	-40	7	-20	27	-12	0	233	243	-12
2	155	160	-7	8	114	96	37	1	79	9	16
3	413	416	-5	9	61	68	-11	2	151	150	0
4	134	157	-30	10	87	80	15	3	150	146	4
5	134	149	-19	11	87	98	-32	4	215	222	-9
6	179	169	14					5	304	308	-6
7	19	20	0		19	3	L	6	230	222	11
8	92	94	-1					7	200	185	20
9	201	177	35	1	91	96	-11	8	87	101	-11
10	147	131	19	2	39	46	-8	9	107	109	-1
11	178	150	38	3	29	9	11	10	136	134	2
12	91	68	20	4	-18	10	-6	11	70	55	10
13	243	211	58	5	26	6	11	12	102	94	9
14	67	88	-2+	6	+1	52	-17	13	4+	23	9
15	175	166	19	7	8	4	1	14	121	115	8
16	33	37	-3					15	-38	18	-13
17	97	102	-11		0	+	L	16	55	56	0
								17	99	99	0
	13	3	L	0	454	494	-60				
				1	56	61	-2		6	+	L
1	510	522	-18	2	666	602	75				
2	270	286	-31	3	281	246	55	0	443	473	-46
3	156	156	0	4	192	166	37	1	151	160	-10
4	41	19	9	5	720	642	87	2	412	401	15
5	45	30	7	6	460	440	30	3	204	205	-1
6	47	41	3	7	760	713	51	+	151	163	-13
7	38	34	1	8	273	269	6	5	35	53	-6
8	65	42	16	9	261	225	60	6	51	4	12
9	73	68	+	10	280	273	12	7	153	133	21
10	30	16	4	11	161	148	19	8	142	118	26
11	68	77	-9	12	360	361	-1	9	308	305	+
12	20	32	-5	13	127	140	-18	10	281	275	9
13	41	44	-2	14	239	238	1	11	161	164	-4
14	-15	22	-8	15	215	217	-3	12	278	272	10
15	25	45	-20	16	326	326	0	13	6	8	0
				17	253	288	-82	14	171	161	18
	15	3	L					15	+1	+0	0
				2	4	L		16	114	110	7
1	232	219	25								
2	242	246	-8	0	375	364	12		8	+	L
3	290	276	28	1	37	45	-2				

TTT2I3 (H.O.)				ROCM TEMP.			COPPER		PAGE		6
0	300	331	-54	5	40	48	-7	6	267	256	15
1	41	33	3	6	168	173	-10	7	94	60	20
2	315	325	-17	7	87	76	16	8	106	99	5
3	156	144	14	8	93	89	7	9	98	113	-14
4	132	142	-9	9	147	141	15	10	44	11	10
5	365	374	-13	10	121	126	-13	11	89	99	-11
6	88	120	-26	11	77	79	-4	12	117	112	7
7	369	368	1					13	84	77	10
8	149	156	-8		16	4	L				
9	149	133	17					7	5	L	
10	182	158	32	0	23	16	3				
11	99	87	11	1	26	29	-2	1	75	80	-3
12	202	202	0	2	13	11	0	2	86	106	-17
13	88	85	3	3	-27	14	-12	3	-26	58	-16
14	164	156	14	4	56	67	-16	4	73	53	10
15	114	118	-11	5	26	1	10	5	9	24	-1
16	97	134	-127	6	135	129	15	6	153	143	10
				7	54	36	27	7	60	35	9
	10	4	L	8	29	34	-6	8	29	6	3
								9	54	36	8
0	69	79	-8		1	5	L	10	23	23	0
1	59	52	4					11	63	69	-6
2	100	99	1	1	241	230	14	12	49	65	-20
3	-21	39	-11	2	528	468	73				
4	78	76	1	3	249	218	37		9	5	L
5	44	39	2	4	257	215	55				
6	219	232	-22	5	30	19	2	1	107	123	-19
7	123	137	-17	6	183	164	23	2	285	311	-49
8	52	59	-4	7	37	16	5	3	115	122	-9
9	73	63	7	8	216	205	16	4	307	312	-8
10	33	33	0	9	100	91	9	5	28	15	3
11	43	37	3	10	-41	40	-17	6	312	305	12
12	67	70	-3	11	34	18	5	7	32	40	-3
13	25	16	3	12	-18	38	-11	8	270	268	4
14	31	17	9	13	5	17	-2	9	45	70	-25
				14	-29	13	-10	10	118	115	6
	12	4	L								
					3	5	L		11	5	L
0	361	372	-20								
1	-22	5	-3	1	67	67	0	1	-6	35	-10
2	299	292	13	2	58	41	5	2	63	57	5
3	255	251	8	3	71	66	2	3	32	47	-9
4	316	323	-13	4	355	352	4	4	199	195	8
5	456	457	-1	5	63	79	-8	5	55	48	6
6	242	238	7	6	422	387	47	6	215	206	20
7	328	322	11	7	-25	28	-5	7	36	6	14
8	131	128	5	8	368	349	29	8	257	260	-7
9	183	173	20	9	76	62	10				
10	223	224	-2	10	301	298	5		13	5	L
11	105	108	-5	11	16	26	-2				
12	182	186	-10	12	20	2	3	1	27	33	-4
13	43	36	9	13	78	76	2	2	30	23	4
								3	17	3	3
	14	4	L					4	51	66	-22
					5	5	L	5	7	36	-18
0	244	237	15	1	-40	32	-9				
1	84	84	0	2	70	61	3		0	6	L
2	238	226	27	3	-53	1	-8				
3	48	55	-6	4	161	162	-1	0	252	316	-81
4	20	6	3	5	26	4	2	1	133	147	-13

TTT2I3 (H.D.)				ROOM TEMP.		COPPER		PAGE			7
2	428	406	28	3	159	183	-23	4	34	44	-3
3	-45	65	-21	4	47	21	7	5	44	56	-5
4	239	269	-46	5	60	56	2				
5	134	166	-45	6	53	67	-10		6	6	L
6	51	17	14								
					4	6	L	0	143	160	-10
	2	6	L					1	97	97	0
				0	107	111	-1	2	132	131	0
0	-131	79	-15	1	-31	42	-4	3	117	155	-49
1	113	160	-10	2	-6	51	-5				
2	-69	32	-16	3	-21	57	-11				

TTT213 (F.D.)				164K		MOLYBDENUM			PAGE			1
				32	210	198	3	22	261	219	11	
	0	0	L	34	-36	34	-1	24	505	548	-19	
				36	26	103	-6	26	333	328	1	
4	3224	3230	-1					28	312	296	4	
6	1829	1810	9		6	0	L	30	225	182	9	
8	1514	1484	17					32	203	148	10	
10	2857	2862	-1	0	1799	1769	21	34	-29	10	0	
12	653	643	7	2	3603	3731	-50					
14	2277	2287	-4	4	1175	1174	1		12	0	L	
16	639	659	-12	6	2220	2212	4					
18	1170	1180	-6	8	2472	2476	-2	0	2562	2630	-25	
20	833	822	6	10	1343	1313	25	2	656	599	40	
22	333	327	2	12	1809	1835	-17	4	2197	2211	-5	
24	1068	1061	3	14	433	398	26	6	1439	1446	-3	
26	271	276	-1	16	1566	1570	-2	8	1152	1133	11	
28	510	493	7	18	754	750	3	10	2432	2476	-16	
30	171	103	10	20	657	658	0	12	374	394	-9	
32	-141	113	-16	22	560	552	6	14	1489	1519	-16	
34	127	75	5	24	268	282	-5	16	436	461	-13	
36	-53	39	-2	26	617	609	5	18	868	891	-14	
				28	188	104	14	20	659	678	-10	
	2	0	L	30	250	211	13	22	294	324	-9	
				32	-97	62	-9	24	588	544	21	
2	2383	2327	31	34	213	195	5	26	154	65	11	
4	3848	3847	0	36	-100	119	-16	28	267	311	-12	
6	2691	2675	8					30	139	92	5	
8	2064	2036	18		8	0	L	32	-177	50	-17	
10	2560	2542	9					34	356	335	6	
12	702	729	-28	0	155	117	19					
14	1698	1695	2	2	3425	3466	-16		14	0	L	
16	744	743	0	4	1662	1651	8					
18	986	977	8	6	2136	2149	-8	0	2380	2389	-3	
20	762	765	-2	8	2071	2087	-10	2	701	698	8	
22	906	921	-12	10	828	827	1	4	1886	1891	-2	
24	412	380	18	12	2073	2062	6	6	1184	1180	2	
26	118	83	5	14	568	585	-14	8	922	828	-3	
28	300	299	0	16	1275	1281	-5	10	1027	1023	2	
30	130	173	-9	18	579	597	-15	12	1024	1046	-13	
32	170	172	0	20	554	552	1	14	1368	1388	-11	
34	250	224	8	22	966	971	-4	16	344	343	0	
36	-70	60	-5	24	120	66	8	18	626	644	-10	
				26	412	395	9	20	493	474	8	
	4	0	L	28	168	122	10	22	202	231	-6	
				30	300	273	11	24	529	537	-3	
0	2350	2340	5	32	193	195	0	26	-85	134	-13	
2	3485	3526	-16	34	78	80	0	28	364	328	11	
4	2228	2222	3					30	128	148	-2	
6	2321	2296	14		10	0	L	32	220	234	-2	
8	2024	2021	1									
10	915	925	-11	0	2933	2933	0		16	0	L	
12	2507	2484	12	2	1569	1587	-10					
14	1041	1045	-3	4	1865	1880	-7	0	1004	946	34	
16	1224	1234	-8	6	1596	1600	-2	2	1269	1307	-22	
18	751	745	5	8	1344	1345	0	4	1164	1165	0	
20	613	626	-11	10	1233	1259	-15	6	1099	1109	-6	
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3 272 296 -6				7 264 292 -17				18 177 144 8	
C + L				8 112 135 -6				19 185 146 10	
				9 146 118 8				10 + L	
0 628 633 -3				10 204 183 9					
1 131 70 10				11 112 92 4				0 89 79 1	
2 724 738 -10				12 175 152 8				1 -74 66 -8	
3 353 345 4				13 -80 23 -7				2 163 125 9	
4 232 215 5				14 180 188 -3				3 -158 23 -19	
5 929 934 -3				15 -51 53 -5				4 51 101 -5	
6 520 517 1				16 -32 61 -4				5 -23 57 -2	
7 871 899 -19				17 222 191 11				6 296 337 -17	
8 248 284 -14				18 173 124 13				7 108 129 -3	
9 377 354 12				19 190 133 15				8 68 46 1	
10 412 408 2				20 -37 6 -1				9 -106 97 -14	
11 230 202 9								10 136 41 12	
12 566 543 14				6 4 L				11 -98 33 -7	
13 168 168 0				0 569 608 -38				12 -61 106 -10	
14 318 362 -19				1 204 208 -1				13 -42 15 -1	
15 339 374 -15				2 564 563 0				14 -163 22 -17	
16 488 571 -45				3 333 320 8				15 -238 100 -39	
17 515 515 0				4 197 248 -25				16 135 194 -10	
18 306 321 -5				5 93 83 2				17 108 37 6	
19 282 253 8				6 -143 11 -23				18 90 176 -12	
20 -119 196 -28				7 277 249 16				12 4 L	
				8 175 194 -7					
				9 434 438 -3				0 552 571 -11	

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6	382	398	-7	9	239	228	2	4	296	277	10
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8	169	195	-6	11	42	133	-9	6	+54	438	11
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14	303	341	-13	2	329	312	5	12	222	180	15
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16	-162	26	-15	4	182	221	-8	14	79	77	0
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7	86	152	-10	7	-51	26	-3	6	224	230	-2
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13	186	0	21	13	68	18	3	12	-47	39	-8
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3	162	174	-2					1	148	17	14

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2	91	108	-2								
3	135	34	10		0	6	L		6	6	L
4	405	385	8								
5	203	69	23	0	512	510	1	0	358	353	2
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9	241	176	15	4	488	489	0	4	106	122	-3
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12	117	52	6	7	296	231	22	7	250	228	9
13	160	163	2	8	144	101	6	8	-88	26	-7
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1	157	54	13								
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6	318	338	-5								

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				32	510	490	10	24	898	910	-9
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25	313	324	-5	25	117	101	3				
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3	343	353	-3	13	227	50	28				
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7	176	221	-10	8	123	221	-14	16	308	284	16
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3	361	371	-4	15	400	366	11	12	430	398	21
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16	85	70	1	22	88	110	-3	7	107	110	0
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4	69	83	-1					5	90	133	-9
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13	-136	13	-14	2	71	33	2	6	224	176	8
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				14	228	248	-4	5	213	95	31
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1	84	209	-20					9	213	136	21
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3	174	198	-5	1	97	13	4	11	204	149	14
4	-142	89	-16	2	395	394	0	12	126	2	12
5	157	60	13	3	308	307	0	13	223	168	15
6	334	373	-13	4	288	284	1	14	260	284	-8
7	219	219	0	5	629	613	7	15	118	118	0
8	-48	162	-15	6	138	111	3	16	314	318	-1
9	444	403	18	7	529	532	-1	17	84	+5	4
10	305	308	0	8	276	162	24	18	481	479	1
11	266	173	22	9	223	225	0	19	120	121	0
12	423	418	1	10	310	319	-2	20	310	265	21
13	-158	25	-12	11	243	158	16				
14	297	289	2	12	314	354	-10		5	5	L
15	-21	34	-1								
16	466	455	6		22	4	L	1	122	42	10
17	75	39	3					2	196	167	8
18	303	285	7	0	-34	95	-4	3	159	36	20
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	16	4	L	2	139	133	0	5	166	25	22
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TTT2I3 (H.O.)				MOLYBDENUM			74K	PAGE			10
9	124	149	-5	3	200	20	22	7	34	10	0
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11	222	183	11	5	-120	78	-10	9	172	34	12
12	238	231	2	6	466	501	-13	10	299	327	-7
13	169	93	13	7	173	8	15	11	153	43	9
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16	295	287	3	10	628	615	6				
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19	-112	37	-11	13	287	222	15	3	-159	34	-11
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	7	5	L	16	210	187	6	6	150	39	9
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2	172	159	3	13	5	L					
3	-123	62	-14					0	6	L	
4	186	126	14	1	94	53	2				
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7	74	14	4	4	128	125	0	2	885	899	-7
8	116	36	9	5	102	45	3	3	-89	102	-10
9	208	83	27	6	172	20	15	4	63+	632	1
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11	135	152	-3	8	242	239	0	6	110	91	2
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16	92	128	-6	13	301	246	12	11	164	82	10
17	92	13	6	14	262	293	-6	12	548	511	16
18	115	112	0	15	80	75	0	13	-124	71	-9
19	-88	60	-9	16	96	45	5	14	335	351	-4
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12	216	150	15	12	322	242	20	9	-104	47	-9
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14	163	101	10	14	157	50	9	11	77	29	3
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16	344	352	-4		17	5	L	13	142	33	14
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TTT2I3 (H.O.)				MOLYBDENUM				74K	PAGE				11
1	78	57	2										
2	217	138	23		10	6	L			1	7	L	
3	-89	114	-16										
4	52	124	-9	0	-58	77	-6		1	210	245	-9	
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6	169	65	20	2	177	218	-10		3	-109	91	-12	
7	85	43	4	3	102	82	2		4	-148	35	-14	
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9	140	84	9	5	127	11	11		6	366	368	0	
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13	146	131	2	9	184	160	5						
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15	-41	18	-2	11	-52	28	-2						
16	116	127	-2	12	91	59	3		1	473	457	8	
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6	141	22	14	9	398	387	3		7	144	161	-3	
7	190	185	1										
8	189	175	3		16	6	L			9	7	L	
9	-40	103	-8										
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14	310	306	1	4	130	33	7						

TTT2I3 (H.O.)				MOLYBDENUM			19K	PAGE			1
				24	874	896	-13	0	3205	3160	13
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8	1683	1654	15	34	380	357	7	10	1580	1568	6
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14	2924	2933	-2	40	565	538	12	16	1290	1299	-5
16	1033	1042	-5					18	1241	1259	-10
18	1798	1797	0		6	0	L	20	1172	1192	-11
20	1483	1516	-17					22	694	714	-10
22	-119	72	-13	0	1970	1911	28	24	1165	1183	-9
24	1974	1989	-6	2	3823	3881	-14	26	868	903	-18
26	690	705	-7	4	1320	1308	7	28	890	891	0
28	1257	1300	-20	6	2443	2417	10	30	776	728	24
30	562	579	-7	8	2811	2815	-1	32	732	702	14
32	619	621	0	10	1627	1587	29	34	388	392	-1
34	680	651	13	12	2348	2360	-4	36	527	527	0
36	391	398	-2	14	700	678	15	38	593	559	15
38	690	680	4	16	2237	2245	-3				
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10	2977	2935	12	34	703	718	-7	12	250	213	14
12	990	1006	-16	36	753	700	25	14	2121	2121	0
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6	2523	2472	19	30	903	893	5	10	1541	1542	0
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16	1782	1793	-5	40	564	531	14	20	1013	1048	-18
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22	1192	1215	-12					26	524	518	2

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28	1006	1022	-8	14	434	465	-13	10	1332	1336	-1
30	594	604	-4	16	1349	1352	-1	12	316	303	4
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34	206	272	-14	20	722	725	-1	16	415	443	-11
36	407	406	0	22	1247	1260	-6	18	832	830	1
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12	1663	1701	-18	4	1205	1237	-17	8	927	905	11
14	656	633	13	6	939	973	-19	10	728	708	9
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18	467	479	-4	7	136	144	-3	11	573	567	4
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1	351	315	35	5	146	150	-2	9	405	384	11
2	342	301	37	6	290	309	-21	10	642	636	3
3	1538	1518	11	7	467	439	33	11	531	524	4
4	2293	2255	15	8	334	349	-15	12	173	132	9
5	1249	1211	26	9	121	91	10	13	542	574	-20
6	1007	982	28	10	527	529	-1	14	198	191	2

TTT2I3 (H.O.)				MOLYBDENUM			19K	PAGE			+
15	331	333	-1	19	-141	51	-14	23	280	184	26
16	382	399	-12	20	-118	9	-8				
17	76	48	3	21	57	33	1	19	1	L	
18	811	829	-11	22	120	72	5				
19	99	179	-15	23	-69	33	-3	1	-107	50	-13
20	582	592	-5	24	-54	51	-3	2	94	74	3
21	174	126	9	25	-149	93	-16	3	243	277	-17
22	297	316	-6					4	244	289	-16
23	172	236	-15	15	1	L		5	105	146	-7
24	91	170	-11					6	346	370	-11
25	358	344	5	1	555	547	4	7	200	224	-7
				2	362	358	1	8	231	205	8
11	1	L		3	377	372	2	9	159	124	7
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1	610	591	20	5	327	341	-6	11	-164	11	-18
2	467	466	0	6	471	476	-3	12	175	12	21
3	419	422	-1	7	242	266	-9	13	132	101	4
4	392	424	-18	8	325	330	-2	14	159	51	15
5	149	81	13	9	141	150	-2	15	157	274	-27
6	1127	1145	-11	10	542	538	3	16	278	255	6
7	478	492	-8	11	525	530	-4	17	142	19	12
8	1129	1134	-3	12	93	36	8	18	198	211	-3
9	495	479	9	13	315	294	8	19	132	113	2
10	342	328	6	14	-89	1	-6	20	123	105	2
11	312	292	8	15	69	57	1	21	45	36	0
12	167	222	-16	16	49	74	-2	22	-180	36	-18
13	534	534	0	17	-255	16	-41				
14	344	349	-3	18	547	535	6	21	1	L	
15	632	644	-10	19	55	25	1				
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17	372	405	-16	21	168	174	-1	2	208	253	-13
18	306	351	-18	22	221	191	7	3	383	353	13
19	90	1	6	23	189	169	4	4	435	461	-14
20	361	332	12	24	126	186	-10	5	-103	53	-8
21	-66	98	-8	25	237	211	6	6	708	699	5
22	407	401	2					7	311	281	11
23	267	206	17	17	1	L		8	615	627	-6
24	109	60	5					9	309	308	0
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13	1	L		3	227	222	2	12	116	93	3
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2	-85	95	-12	6	686	687	0	15	350	377	-10
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6	519	547	-17	10	371	383	-6	19	75	98	-2
7	568	571	-1	11	448	432	8	20	190	217	-5
8	489	501	-7	12	171	149	5				
9	186	217	-9	13	345	383	-16	23	1	L	
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11	157	108	11	15	336	350	-5	1	414	394	8
12	75	47	3	16	414	392	10	2	328	312	6
13	148	132	4	17	188	239	-11	3	457	447	4
14	219	229	-4	18	264	228	10	4	729	726	1
15	443	464	-11	19	277	163	30	5	328	322	2
16	530	541	-6	20	364	374	-3	6	585	555	15
17	101	158	-10	21	78	131	-6	7	-126	84	-13
18	90	44	4	22	273	292	-5	8	380	393	-5

TTT213 (H.D.)				MOLYBDENUM				19K	PAGE		5
9	197	231	-8	9	1498	1457	23	11	62	66	0
10	352	374	-8	10	1045	1006	27	12	647	624	15
11	328	322	2	11	691	718	-19	13	187	192	-1
12	217	104	24	12	156	139	3	14	383	391	-4
13	335	330	1	13	801	788	8	15	268	309	-16
14	144	72	9	14	1418	1430	-6	16	522	503	16
15	-208	135	-31	15	633	640	-4	17	73	91	-3
16	-146	170	-26	16	-110	21	-13	18	396	387	6
17	-53	34	-2	17	845	844	0	19	627	603	21
18	571	535	17	18	282	280	1	20	46	64	-2
				19	489	497	-4	21	205	175	10
25	1	L		20	-98	184	-26	22	347	287	32
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2	70	47	1	23	188	142	9	25	278	253	10
3	81	134	-6	24	452	443	3				
4	192	174	3	25	-157	181	-29		6	2	L
5	198	220	-5								
6	-86	73	-7		2	2	L	0	140	7	34
7	131	25	10					1	350	353	-3
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9	97	56	3	1	-126	44	-27	3	156	163	-3
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11	190	109	14	3	1040	1019	16	5	1397	1398	0
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13	117	124	0	5	586	583	3	7	760	767	-5
14	-123	68	-11	6	568	579	-13	8	357	352	2
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2	141	105	5	12	195	181	4	14	303	263	17
3	35	82	-3	13	126	93	5	15	524	512	10
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5	198	148	10	15	119	223	-26	17	422	445	-17
6	191	185	1	16	336	324	8	18	34	39	0
7	-110	65	-9	17	588	589	0	19	644	640	3
8	287	308	-6	18	512	491	17	20	458	450	5
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11	227	202	6	21	452	467	-10	23	527	519	5
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29	1	L		23	-159	35	-21	25	161	88	15
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2	160	227	-14								
3	260	294	-9		4	2	L	0	165	88	29
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4	1056	1064	-6	6	420	431	-12	8	951	945	4
5	881	894	-11	7	103	67	9	9	1212	1204	4
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7	2096	2058	23	9	595	581	10	11	192	89	22
8	648	590	63	10	433	401	18	12	670	669	0

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13	-148	79	-21	15	72	115	-6	18	240	214	6
14	128	160	-10	16	-163	59	-20	19	-186	27	-20
15	650	675	-22	17	743	749	-3	20	422	390	12
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17	302	289	5	19	717	721	-2	22	202	257	-13
18	354	368	-6	20	642	667	-13	23	227	146	17
19	656	681	-14	21	803	798	2				
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24	189	82	17					2	540	538	1
25	190	121	13					3	187	168	6
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5	218	177	12	7	1065	1095	-19	12	477	497	-9
6	422	372	27	8	239	199	19	13	288	287	0
7	585	573	7	9	380	378	1	14	347	275	25
8	249	157	27	10	605	600	4	15	337	305	12
9	170	69	19	11	488	497	-7	16	292	321	-9
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11	271	254	6	13	138	179	-9	18	199	100	18
12	377	371	3	14	648	648	0	19	459	468	-4
13	303	342	-24	15	221	206	4	20	231	241	-2
14	628	640	-11	16	282	178	32	21	167	99	9
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21	92	106	-1	23	193	230	-8	2	731	737	-3
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7	1011	1031	-13	10	306	280	10	17	305	300	1
8	371	388	-8	11	187	198	-3	18	285	277	2
9	443	484	-22	12	301	275	10	19	743	719	12
10	574	592	-11	13	120	72	6	20	192	51	19
11	542	542	0	14	199	143	13				
12	65	12	4	15	288	264	8	22	2	L	
13	246	258	-5	16	276	277	0				
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1	-67	110	-10	5	120	57	6	4	148	108	8
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3	210	240	-8		1	3	L	6	155	71	15
4	381	382	0					7	151	97	9
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6	368	324	17	2	228	174	22	9	369	317	23
7	90	78	1	3	445	417	18	10	99	139	-7
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16	266	275	-2	13	843	831	8	19	-58	74	-8
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5	284	201	24	24	73	93	-2	3	264	160	34
6	277	241	10					4	181	70	22
7	710	712	-1		3	3	L	5	120	75	6
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13	360	372	-4	5	368	363	2	11	205	144	16
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15	277	215	15	7	270	155	38	13	108	18	11
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4	279	239	11	3	501	488	8	9	690	719	-20

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10	144	143	0	14	134	14	12				
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19	113	27	7	23	388	402	-5	7	58	10	2
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8	90	118	-4	13	622	662	-21	2	204	183	4
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25	3	L		9	342	344	-1	13	136	88	12
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8	154	167	-2	18	204	212	-3	22	228	226	0
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	27	3	L	23	142	95	9				
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2	-43	5	-1		4	4	L	2	499	546	-29
3	283	218	17					3	352	363	-5
4	133	84	5	0	338	377	-18	4	279	230	20
				1	76	1	4	5	713	758	-30
	0	4	L	2	227	194	11	6	109	178	-22
				3	228	211	6	7	660	683	-21
0	730	749	-13	4	294	292	0	8	252	219	17
1	126	63	10	5	463	460	1	9	326	309	10
2	852	878	-18	6	336	352	-7	10	345	354	-5
3	502	499	1	7	362	342	10	11	210	186	7
4	219	238	-6	8	148	143	1	12	486	481	3
5	1122	1146	-22	9	164	151	4	13	97	77	2
6	576	570	4	10	232	230	1	14	395	369	12
7	1061	1078	-11	11	38	94	-8	15	288	319	-11
8	307	300	3	12	206	199	3	16	365	401	-16
9	435	444	-5	13	101	32	10	17	528	510	9
10	520	519	0	14	210	257	-22	18	268	202	19
11	258	263	-2	15	124	132	-2	19	335	314	7
12	754	743	10	16	100	68	5	20	100	127	-3
13	31	127	-16	17	277	273	1	21	200	179	4
14	463	495	-18	18	105	137	-6	22	138	179	-7
15	486	493	-4	19	215	208	2	23	80	186	-14
16	743	756	-8	20	40	20	1	24	290	217	20
17	744	769	-15	21	-89	89	-13				
18	400	425	-11	22	-111	128	-24		10	4	L
19	423	449	-12	23	96	98	0				
20	276	265	3	24	158	119	8	0	30	53	-2
21	242	265	-6					1	63	53	1
22	179	231	-11		6	4	L	2	131	176	-15
23	210	260	-12					3	110	52	11
24	325	326	0	0	686	717	-21	4	147	110	11
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1	235	146	28	5	187	173	4	9	-101	87	-13
2	259	270	-4	6	110	6	11	10	-76	40	-5
3	250	273	-9	7	375	350	13	11	139	140	14
4	426	434	-4	8	279	252	15	12	141	137	0
5	147	130	3	9	583	589	-5	13	117	9	10
6	549	579	-19	10	500	504	-3	14	174	16	22
7	111	151	-8	11	308	281	16	15	-134	62	-14
8	180	178	0	12	566	581	-13	16	234	297	-18

TTT213 (H.D.)				MOLYBDENUM			19K	PAGE			10
17	154	33	15					14	378	362	5
18	270	236	10	0	182	111	14	15	309	284	7
19	37	21	0	1	-25	20	0				
20	-230	113	-34	2	-126	27	-11		22	4	L
21	-142	65	-13	3	148	36	15				
22	34	25	0	4	172	148	5	0	107	101	0
23	-150	18	-12	5	76	78	0	1	-198	53	-24
				6	313	308	1	2	93	152	-8
	12	4	L	7	43	9	1	3	-23	34	0
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0	728	729	0	9	146	8	15	5	59	21	1
1	53	42	1	10	94	43	4	6	255	210	12
2	447	461	-10	11	69	25	2	7	-71	43	-4
3	437	456	-14	12	-131	4	-10	8	-62	55	-4
4	547	550	-2	13	116	17	8	9	101	102	0
5	908	897	10	14	-179	68	-21	10	137	42	10
6	521	529	-4	15	190	8	23	11	-98	43	-6
7	739	725	8	16	-66	174	-18	12	167	105	10
8	213	264	-16	17	49	28	0	13	-121	9	-8
9	373	352	10	18	164	134	5				
10	486	488	-1	19	76	23	3		24	4	L
11	244	213	10								
12	469	454	7		18	4	L	0	274	251	6
13	122	51	8					1	-21	36	-1
14	482	512	-14	0	415	463	-20	2	272	291	-5
15	341	305	14	1	157	158	0	3	179	183	0
16	-143	28	-13	2	358	351	2	4	171	19	17
17	610	576	17	3	287	254	11	5	352	368	-5
18	197	194	0	4	259	254	1	6	264	283	-5
19	433	422	4	5	168	139	5	7	420	374	17
20	110	5	7	6	85	131	-6	8	-143	137	-21
21	202	203	0	7	227	191	9				
22	234	241	-1	8	154	168	-2		1	5	L
				9	332	333	0				
	14	4	L	10	315	295	6	1	484	483	0
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1	172	209	-10	13	-116	52	-9	4	373	349	15
2	549	563	-8	14	340	298	14	5	100	30	9
3	218	255	-11	15	-134	19	-10	6	397	376	14
4	98	103	0	16	27	102	-5	7	157	15	26
5	134	80	9	17	-152	14	-13	8	492	467	18
6	394	400	-3	18	166	2	16	9	123	41	13
7	264	282	-6					10	129	113	3
8	143	175	-7		20	4	L	11	30	67	-3
9	458	435	11					12	216	132	27
10	318	328	-4	0	491	510	-8	13	122	18	13
11	185	205	-4	1	126	35	9	14	110	74	5
12	455	463	-3	2	416	438	-9	15	131	14	14
13	113	49	6	3	344	324	7	16	317	261	25
14	319	319	0	4	275	318	-13	17	172	85	18
15	178	33	19	5	734	702	17	18	266	207	21
16	524	528	-1	6	-129	128	-18	19	87	88	0
17	69	4	2	7	599	607	-4	20	555	543	7
18	320	328	-2	8	245	184	16	21	252	286	-13
19	100	193	-15	9	275	280	-1	22	300	313	-5
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21	259	298	-11	11	-20	167	-16				
				12	396	407	-4		3	5	L
	16	4	L	13	88	59	2				

TTT2I3 (H.D.)				MOLYBDENUM			19K	PAGE			11
1	193	57	36	11	191	158	7				
2	165	158	2	12	180	123	10	1	156	41	14
3	37	9	1	13	148	69	10	2	-191	31	-21
4	781	781	0	14	247	65	35	3	142	33	11
5	162	99	17	15	177	23	18	4	48	129	-8
6	843	836	6	16	237	140	22	5	-59	43	-3
7	109	1	12	17	154	16	13	6	-123	32	-9
8	778	785	-6	18	142	123	2	7	76	26	2
9	234	158	30	19	145	61	9	8	285	247	11
10	701	692	7	20	164	71	11	9	246	293	-13
11	211	186	9	21	160	169	-1	10	383	411	-10
12	185	5	33					11	342	383	-14
13	275	198	32		9	5	L	12	290	276	4
14	326	318	4					13	352	269	27
15	197	132	18	1	320	314	2	14	336	325	3
16	390	373	9	2	654	639	9	15	80	63	1
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18	574	564	6	4	649	650	0	17	154	35	13
19	151	117	7	5	121	18	10	18	-177	89	-19
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21	115	155	-7	7	60	15	2				
22	290	232	21	8	704	695	5		15	5	L
				9	-64	24	-2				
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				11	25	35	0	2	415	391	9
1	125	38	14	12	199	170	6	3	80	136	-6
2	211	184	10	13	136	41	10	4	449	449	0
3	136	52	16	14	170	126	7	5	132	22	9
4	420	406	10	15	80	69	0	6	606	631	-12
5	61	30	2	16	425	422	1	7	146	20	12
6	625	616	7	17	181	104	13	8	424	374	19
7	-125	21	-15	18	487	450	16	9	112	164	-7
8	375	331	28	19	-46	10	-1	10	-37	54	-2
9	167	143	7	20	539	560	-8	11	262	182	19
10	155	75	18	21	198	137	11	12	343	277	21
11	225	187	14					13	121	93	3
12	300	254	22		11	5	L	14	187	50	18
13	156	93	13					15	128	40	8
14	-38	63	-4	1	-90	22	-5	16	370	373	0
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16	347	332	7	3	167	1	18				
17	139	68	11	4	596	593	1		17	5	L
18	332	302	14	5	63	85	-2				
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21	107	69	5	8	690	682	4	3	145	14	11
22	221	209	3	9	228	247	-4	4	478	493	-6
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	7	5	L	11	391	326	25	6	597	578	9
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3	142	57	12	15	-100	112	-12	10	417	380	14
4	163	139	5	16	218	219	0	11	122	46	7
5	-72	36	-4	17	96	13	5	12	111	51	5
6	246	277	-10	18	393	413	-7	13	150	71	9
7	156	11	17	19	211	78	21	14	174	138	6
8	72	33	2	20	285	309	-6	15	128	73	5
9	156	88	11								
10	115	8	9		13	5	L		19	5	L

TTT213 (H.D.)				MOLYBDENUM			19K	PAGE			12
				4	146	3	23	18	181	241	-18
1	141	153	-1	5	118	204	-27				
2	196	227	-6	6	99	165	-16		8	6	L
3	-63	56	-3	7	169	125	12				
4	170	114	9	8	238	210	11	0	404	481	-31
5	-16	9	0	9	-128	57	-18	1	-41	87	-5
6	58	57	0	10	272	283	-5	2	581	597	-8
7	184	1	18	11	-144	32	-19	3	57	18	2
8	248	207	10	12	248	259	-4	4	450	438	5
9	-123	92	-12	13	-231	34	-43	5	264	206	16
10	160	188	-4	14	393	385	4	6	-58	17	-2
11	197	171	5	15	103	38	8	7	200	215	-3
12	-150	44	-13	16	69	137	-11	8	263	205	17
13	270	100	35	17	112	137	-5	9	196	113	16
				18	152	136	3	10	312	318	-2
	21	5	L	19	260	259	0	11	142	68	9
								12	414	412	0
1	153	110	5	4	6	L		13	58	14	1
2	219	232	-3					14	254	373	-35
3	-168	51	-15	0	306	295	5	15	131	105	3
4	375	372	1	1	125	58	12	16	349	284	23
5	161	48	13	2	117	158	-10	17	81	140	-7
6	386	367	6	3	134	123	2				
7	-90	1	-4	4	114	140	-6		10	6	L
8	480	459	8	5	146	133	3				
9	234	156	15	6	-44	62	-5	0	-133	67	-12
				7	-73	43	-6	1	123	76	6
	23	5	L	8	307	278	14	2	257	227	9
				9	30	88	-6	3	-182	85	-24
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	0	6	L	11	153	6	22	5	93	12	5
				12	247	222	10	6	143	51	11
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1	195	198	0	15	-99	23	-8	9	275	183	27
2	975	989	-8	16	129	148	-4	10	90	3	4
3	141	109	6	17	-33	97	-9	11	-139	31	-12
4	691	701	-5	18	37	1	1	12	151	77	10
5	290	296	-2	19	160	96	13	13	199	167	7
6	65	85	-2					14	100	108	-1
7	386	354	14	6	6	L		15	-169	118	-24
8	186	200	-3					16	220	28	28
9	197	238	-11	0	599	602	-2				
10	456	434	10	1	218	199	6		12	6	L
11	-104	102	-13	2	464	445	13				
12	623	606	9	3	324	361	-21	0	593	617	-12
13	-111	65	-9	4	178	189	-3	1	147	87	8
14	457	431	11	5	323	295	14	2	431	414	7
15	126	195	-13	6	159	111	12	3	128	220	-17
16	428	435	-2	7	304	347	-23	4	392	330	26
17	173	197	-4	8	97	10	9	5	232	273	-11
18	-91	36	-5	9	381	387	-3	6	78	87	0
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2	-20	56	-3	16	254	248	2	13	283	248	10
3	416	387	20	17	164	184	-5	14	704	720	-8

TTT2I3 (H.D.)				MOLYBDENUM			19K	PAGE			13
15	-49	28	-1	4	-86	54	-8	4	112	56	5
				5	254	257	-1	5	20	57	-1
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3	151	136	2	11	278	292	-5	11	-119	17	-7
4	223	219	1	12	95	83	1	12	-107	16	-6
5	94	200	-18	13	119	179	-13				
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7	361	341	7								
8	127	163	-6		3	7	L	1	489	463	10
9	404	434	-11					2	-105	140	-16
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12	340	303	12	3	343	360	-8	5	198	258	-13
13	346	306	13	4	374	367	3	6	417	360	22
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	16	6	L	6	133	192	-14	8	346	358	-4
				7	167	234	-19	9	267	242	6
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2	204	83	22	10	223	249	-8				
3	154	104	7	11	331	384	-24	11	7	L	
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5	81	39	2	13	399	409	-5	1	445	410	14
6	228	48	31	14	-46	24	-2	2	-44	85	-5
7	-159	32	-15					3	79	166	-11
8	-114	134	-16		5	7	L	4	401	412	-3
9	149	89	8					5	155	106	7
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4	147	96	7	10	-87	85	-11	3	74	144	-7
5	182	176	1	11	145	199	-13	4	390	339	17
6	120	91	3	12	70	67	0	5	314	253	16
7	108	182	-12	13	300	277	9	6	-69	155	-14
	1	7	L		7	7	L		0	8	L
1	297	281	7	1	56	78	-1	0	-71	107	-8
2	109	117	-1	2	105	12	6				
3	91	120	-5	3	231	204	6				

TTTT				LATTICE 1 (C=3.6+Å)			COPPER RAD.			PAGE	1
				8	226	230	-17	5	334	354	-76
	-16	K	0	10	235	224	43	7	206	226	-76
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2	84	86	-13	16	10	2	7	13	204	206	-9
4	53	55	-10					15	166	163	15
					-9	K	0	17	99	95	23
	-15	K	0					19	91	87	28
				1	100	103	-17				
1	136	123	75	3	188	185	15		-4	K	0
3	92	84	+6	5	215	216	-4				
5	32	30	5	7	441	444	-8	0	607	638	-68
7	45	35	46	9	198	199	-4	2	706	751	-86
				11	184	171	50	4	324	334	-39
	-14	K	0	13	84	83	4	6	212	216	-21
0	23	11	16	15	60	61	-4	8	98	83	70
2	35	33	5	17	59	55	28	10	57	52	13
4	69	67	9					12	179	185	-29
6	161	153	+2		-8	K	0	14	208	203	23
8	182	163	88	0	395	403	-26	16	249	241	35
10	118	113	33	2	431	431	0	18	165	157	45
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	-13	K	0	6	51	32	58		-3	K	0
1	+7	52	-16	8	91	89	8				
3	74	73	4	10	45	30	26	1	516	537	-55
5	135	123	62	12	132	130	8	3	371	391	-71
7	136	129	36	14	168	155	65	5	404	426	-72
9	157	132	112	16	203	192	56	7	64	75	-51
11	89	80	53	18	137	129	56	9	303	302	3
					-7	K	0	11	203	213	-49
	-12	K	0					13	202	203	-4
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6	34	27	15	7	257	271	-59		-2	K	0
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14	72	77	-29	15	169	164	26	4	410	447	-121
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	-11	K	0		-6	K	0	8	667	687	-39
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3	194	186	37	2	116	113	20	12	214	218	-19
5	94	92	9	4	260	266	-27	14	163	168	-26
7	138	131	34	6	451	474	-65	16	26	15	17
9	44	35	19	8	462	481	-52	18	54	52	9
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13	123	111	70	12	193	197	-18		-1	K	0
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6	174	172	10	3	287	299	-53	13	142	139	15
								15	109	113	-20

		TTTT		LATTICE 1 (C=3.64A)		COPPER RAD.		PAGE		2
17	150	145	27	4	338	353	-57	9	K	0
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14	438	430	22							
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9	345	367	-80					16	35	43 -50
11	261	269	-35		6	K	0			
13	127	130	-15					11	K	0
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7	61	50	41		8	K	0	3	67	62 21
9	242	245	-13					5	119	110 46
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TTTT				LATTICE 1 (C=3.64A)				COPPER RAD.				PAGE	3
8	127	113	73		-12	K	1		-3	175	177	-9	3
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				-12	71	78	-34	1	136	132	18		
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	-16	K	1					-14	131	135	-19		
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-2	98	105	-41					-10	78	79	-2		
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	-15	K	1	-7	87	84	7	-2	280	275	19		
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		TTTT	LATTICE 1 (C=3.64A)				CGPPER RAD.			PAGE	4
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TTTT LATTICE 1 (C=3.64A) CGPPER RAD. PAGE 5											
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TTTT				LATTICE 1 (C=3.64A)				COPPER RAD.				PAGE		6	
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								5				69		72	
								7				19		13	
								9				64		43	
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								13				26		25	
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TTTT LATTICE 1 (C=3.64A) CCPPER RAD. PAGE 7											
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	-7	K	2					14	24	30	-9
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-8	137	152	-49	-11	24	56	-52	-6	137	108	111
-6	172	203	-140	-9	2+	5	8	-+	212	201	46
-4	239	276	-170	-7	49	73	-60	-2	-3	24	-15
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0	183	172	55	-3	1+	1	5	2	69	88	-63
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-13	-3	30	-23	-1+	86	67	76	-5	1+9	166	-78
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17	17	10	10	16	44	33	50	-8	31	43	-20
								-6	66	80	-47
	4	K	2		7	K	2	-+	67	55	35
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-16	32	46	-43	-15	21	26	-11	0	77	49	80
-1+	92	99	-30	-13	1+	33	-29	2	9+	65	97

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TTTT				LATTICE 1 (C=3.64A)				CCPPER RAD.		PAGE	12
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	3 K	4		-2	-35	12	-30				
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-5	-37	25	-20	4	37	39	-3	2	23	26	-3

HMTSF-I				MOLYBDENUM RAD.				PAGE			
0 K -9				0 K -4				8 -62 17 -13			
								9 39 95 -3			
								10 451 408 57			
1	429	+38	-10	1	508	495	22	11	52+	537	-18
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0 K -8				7	456	429	36				
				8	650	666	-21	0 K 0			
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7	-74	17	-11					6	766	782	-23
0 K -7				1	219	181	55	7	166	166	0
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HMTSF-I				MOLYBDENUM RAD.				PAGE			
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-7	200	193	6	-16	75	95	-7	9	352	359	-6	
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-1	201	228	-31	-12	-18	32	-3	14	130	138	-4	
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-13	239	263	-24	12	158	175	-13	5	1159	1176	-13	
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5	188	16+	18	-5	352	324	33	-1+	+18	+55	-29
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-13	170	182	-7	12	-70	13	-8	3	302	295	9
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14	138	73	21								
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				-6	56	79	-4				
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	4 K	-5		-15	204	213	-5				
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0	1+3	178	-1+	-13	-13	25	-1	-13	264	270	-4
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-17	89	32	10	10	82	20	11	5	223	247	-21
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-2	45	20	+	-7	150	105	23	-10	10+	122	-6
-1	173	158	11	-6	500	477	24	-9	162	175	-7
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5	123	142	-4	0	498	517	-24	-3	36	30	0
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7	122	99	9	2	553	558	-5	-1	281	296	-15
8	587	593	-5	3	575	562	14	0	435	439	-4
9	610	605	4	4	50	38	2	1	586	604	-20
10	217	244	-20	5	171	162	6	2	328	358	-34
11	-62	24	-6	6	161	153	4	3	-45	27	-6
12	-69	76	-1+	7	595	597	-2	+	+5	12	-5
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	5 K	2		9	154	163	-4	6	605	612	-7
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-15	256	295	-25	12	110	112	0	9	62	47	3
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-7	18	53	-5	-12	73	103	-8	-13	-35	91	-12
-6	168	153	10	-11	445	451	-5	-12	288	303	-9
-5	207	192	12	-10	354	337	14	-11	239	250	-6
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3	577	530	+8	-2	72	53	6	-3	115	113	0
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5	-82	42	-17	0	57	19	8	-1	15	17	0
6	-54	14	-6	1	68	35	9	0	-62	50	-14
7	321	292	35	2	580	587	-8	1	40+	411	-8
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9	-13	49	-4	4	71	80	-3	3	102	110	-3

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4	132	152	-12	8	235	242	-5	-11	73	0	7	
5	405	405	0	9	432	418	12	-10	50	14	3	
6	222	204	14	10	276	268	5	-9	-42	71	-9	
7	50	33	3	11	32	63	-4	-8	-55	56	-8	
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11	425	431	-5					-4	59	41	2	
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	5 K	7		-5	25	13	0					
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-5	30	79	0	4	449	437	10	-7	84	52	7	
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	5 K	8		4	80	25	9	-8	314	329	-13	
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-13	120	93	9	-8	121	119	1	-7	530	530	0	
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	6 K	-2		-15	71	70	0	-16	-60	0	-4	
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-11	-47	77	-15	-8	92	31	17	-9	-44	22	-6	
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-8	142	144	-1	-5	137	158	-15	-6	407	385	21	
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	6 K	-1						12	-80	34	-10	
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-9	79	29	12	-8	86	52	10	-11	97	62	10	

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-10	152	166	-7	-11	103	109	-1	-8	115	101	4
-9	50	26	3	-10	131	106	10	-7	68	85	-4
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-12	208	208	0	-11	-76	37	-9	-5	313	330	-13
-11	52	29	3	-10	151	138	5	-4	163	134	-8
-10	76	15	9	-9	-59	2	-5	-3	134	116	8
-9	254	230	18	-8	-71	55	-13	-2	29	9	1
-8	44	5	5	-7	97	82	4	-1	327	355	-24
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3	241	243	-1	-2	139	139	0	-10	99	82	6
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-5	53	71	-5	-2	324	349	-27	0	6	18	0
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-4	80	45	5	3	-93	5	-14	-3	-44	25	-3
-3	90	104	-6	4	-30	51	-4	-2	-101	30	-14
-2	252	259	-5					-1	49	64	-2
-1	110	107	1		10 K	8		0	40	25	1
0	96	29	14								
1	117	66	1+	-5	201	15+	22		11 K	5	
2	95	97	-1	-4	99	141	-12				
3	87	15	11	-3	71	58	2	-7	-28	55	-4
+	80	0	9	-2	-59	75	-13	-6	-86	11	-10
				-1	-52	35	-6	-5	-27	32	-2
	10 K	6		0	1+3	139	3	-+	58	8+	-5
				1	16+	200	-17	-3	72	12	8
-9	73	57	2	2	48	86	-7	-2	54	53	0
-8	95	26	11					-1	-112	37	-20
-7	61	80	-3		11 K	2		0	-44	75	-10
-6	-74	19	-8								
-5	101	1+	15	-7	-73	+5	-10		11 K	6	
-4	203	197	3	-6	-76	39	-10				
-3	210	212	-1	-5	113	109	1	-6	75	37	5
-2	96	89	2	-+	23	22	0	-5	0	52	-3
-1	14	53	-14	-3	-109	44	-16	-4	-109	12	-17
0	-35	74	-10					-3	43	4	2
1	146	134	3		11 K	3		-2	21	23	0
2	230	233	-1					-1	160	108	19
3	168	155	5	-8	70	25	7	0	117	143	-9